NBS CIRCULAR 536

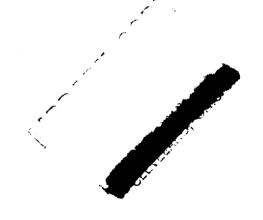
Thermal Conductivity of Metals and Alloys at Low Temperatures

A Review of the Literature

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Thermal Conductivity of Metals and Alloys at Low Temperatures

A Review of the Literature

Robert L. Powell and William A. Blanpied



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National Bureau of Standards Circular 556

Issued September 1, 1954

Preface

Accurate data on the thermal conductivity of materials of construction at low temperatures are essential in the design of cyrogenic equipment. Such data on pure metals also have important applications in basic physics.

This Circular is issued to satisfy the need for a complete and authoritative compilation of the useful data on thermal conductivity at low temperatures given in the widely scattered and extensive literature on the subject. Although the Circular is not primarily a critical compilation, the text indicates a method that might be used in choosing between conflicting data.

It will be noted that there are wide unexplored regions; much experimental work remains to be done. It is hoped, therefore, that this Circular will stimulate additional measurements and indicate the areas in which data are most needed.

A. V. ASTIN, Director.

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Thermal Conductivity of Metals and Alloys at Low Temperatures

A Review of the Literature[†]

Robert L. Powell and William A. Blanpied*

An extensive compilation is given of the measured values of thermal conductivity for metals and alloys from room temperature down to approximately 0° K. The more extensive and important data are plotted in 48 graphs. The tables and graphs for the metallic elements and alloys are essentially complete for literature reference from 1900 to early 1954. For comparison, several graphs and tables are given for some representative dielectrics.

1. Introduction

1.1. Scope and Arrangement

The thermal conductivity values of three types of solids are presented: (1) metallic elements, (2) alloys, and (3) dielectrics. Very little discussion is presented on the qualitative theories or significance of the various experiments. Recent articles, as indicated under each material, usually will contain comments on these aspects of conductivity. Under "metallic elements," the materials are arranged by periodic groups, beginning with the alkali metals. Under "alloys," the materials are arranged in this same manner by major component. In group 3, several dielectrics are included for comparison. A list of the figures and tables is given in section 2.1.

The professional abstract and leading research journals were searched for references dating from 1900 to the spring of 1954. It is felt that the compilation is complete for the metals and essentially complete for the alloys, but only a few representative references are given for the dielectrics. Conductivity values were collected for the temperature range approximately 0° to 300° K. Many of the references contain information for room temperature only, and conductivity values from these are given in the tables only.

The letters at the left end of the curves are a code to the names of the authors. The symbols at the right end of the curves indicate the material tested. Conductivity values in the graphs and tables are given in units of watts

†This work was supported by funds from the U. S. Atomic Energy Commission.

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per centimeter degree Kelvin (except for a few, which are in milliwatts per centimeter degree Kelvin). A table of conversion factors is given. Arrows on the bottom of the graphs indicate the normal boiling points of helium, hydrogen, and nitrogen, and the melting point of ice, respectively. A bibliography (nearly all dated after 1900) is included at the end of the Circular. It is shown in the tables when other properties of the samples have been measured, such as electrical resistance, thermal electromotive force, and specific heat, which are symbolized by R, emf, and C_p, respectively.

The units in the tables and graphs are usually watts per centimeter degree Kelvin. These may be converted to other systems of units by use of the following factors:

To convert to—	Multiply by—
Cal/cm deg K	0.239
Btu/ft hr deg F	57.8
Btu in/ft² hr deg F	693

The preparation of this Circular required the assistance and cooperation of many. Foremost among them was Charles A. Meizner, who plotted most of the graphs and analyzed some of the original research papers. The cooperation of the many authors and manufacturers who supplied reprints of their articles and manuals for use in this study is acknowledged.

2. Figures and Tables

2.1. List
METALLIC ELEMENTS

	Figu	res	Tables
Material	Number	Page	(page)
Aluminum	3, 3a	8, 9	8
Antimony	20	36	36
Beryllium	2	6	6
Bismuth	20	36	36, 37
Cadmium	15, 14a	27, 26	26
Carbon (graphite)	17	30	30
Cerium	21	37	37
Cobalt	8, 9	16, 17	17
Copper	11, 11a	20, 21	20, 21
Gallium	16	29	29
Germanium	18, 19a	31, 35	31
Gold	13, 12a	24, 23	24
Indium	16	29	29
Iridium	10, 10a	18, 19	19
Iron	8, 9	16, 17	16
Lanthanum			9
Lead	19, 19a	34, 35	34
Lithium	1	5	5
Magnesium	2, 2a	6, 7	6, 7
Manganese	7	14	15
Mercury	15, 15a	27, 28	27
Molybdenum	6, 6a	12, 13	12
Nickel	8, 9	16, 17	17
Niobium	5	11	11
Paladium	10, 10a	18, 19	18
Platinum	10, 10a	18, 19	19
Potassium	1	5	5
Rhodium	10, 10a	18, 19	18
Silicon			30
Silver	12, 12a	22, 23	22
Sodium	1	5	5
Tantalum	5	11	11
Tellurium	21	37	37
Thallium	16	29	29
Tin	18, 18a, 18b	31, 32, 33	32
Titanium	4	10	10
Tungsten	6, 6a	12, 13	12, 13
Uranium	21	37	37
Vanadium	5	11	11
Zinc	14, 14a	25, 26	25
Zirconium	4	10	10

ALLOYS

35.4.4.	Figu	Tables	
Material	Number	Page	(page)
Alkali metal			38
Aluminum	22	40	39, 40,
Antimony	29	54	55
Beryllium		!	38
Bismuth	29, 29a	54, 55	55
Cadmium	29	54	53
Chromium			42
Copper	27, 29a	49, 55	48, 49, 50, 51
Copper-nickel	28, 29a	51, 55	51, 52
Gold	26	47	52, 53
Indium	29, 29a	54, 55	53
Iron:		1	
Carbon steel	23, 24	43, 44	42
Deoxidized steels_	24	44	45
Silicon steels			43
Corrosion resisting steels	23, 24	43, 44	43, 44, 45
Lead	29, 29a	54, 55	54
Magnesium	2a	7	38, 39
Mercury			53
Nickel	25	46	45, 46
Palladium	26	47	47
Platinum	26	47	47, 48
Silver	26	47	52
Thallium	29, 29a	54, 55	53
Tin	18a, b	32, 33	53
Titanium	29	54	41
Tungsten			41
Zinc]	53

DIELECTRICS

Beryllia	32	60	60
Diamond	30, 30a	57, 58	57
Disordered di- electrics	33, 33a	62, 63	62
Ionic crystals	32, 32a	60, 61	60
Miscellaneous			56
Quartz	31, 30a	59, 58	59
Sapphire	30, 30a, 32	57, 58, 60	57
i	•	ł	

2.2. Metallic Elements

The variations of the thermal conductivities of metallic elements with temperature are given in figures 1 to 21. The main figures (those without a or b) have the higher temperature curves; usually the temperature range 4° to 300° K. When there is sufficient data in the liquid-helium range, there is a supplementary graph for the range from approximately 0° to 5° or 10° K. The graphs are arranged by periodic groups, beginning with the alkali metals. A summary table is included for each graph, giving for each element a list of references to research papers on the thermal conductivity of the element. The first column contains the chemical symbol and the property or composition identification on the curves if the data for the author reference are plotted on the graph.

Not all the available data are plotted on graphs. If measurements were made at only 1 or 2 temperatures, representative conductivity values are usually given in the "Remarks" column in the corresponding table. When several authors report values that are nearly identical, the report that was published first is usually represented on the graph. There are exceptions to this when the results of a later author are more accurate or more extensive. In most graphs, where there are more than one curve for a given element, the graph showing the highest conductivity is considered most likely to be representative of that of the pure material. The higher values are associated with the more pure material, adequate annealing, and large crystal size.

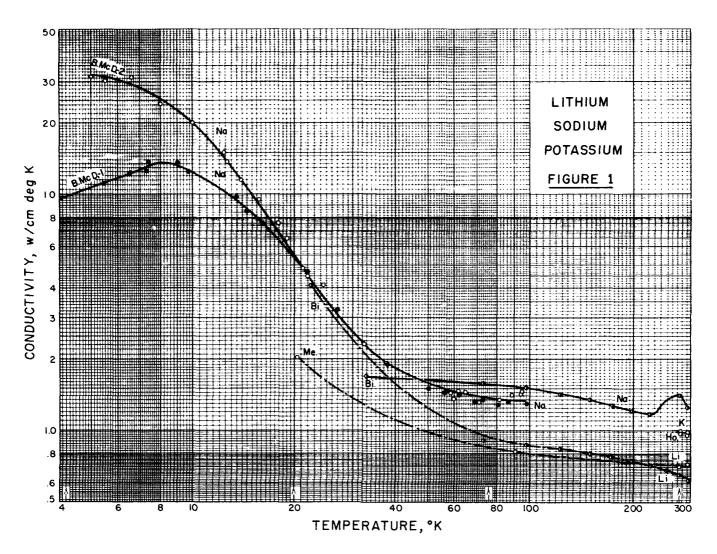
In the commonly accepted theory for the conductivity of metals, there are two mechanisms for the conduction of heat. In pure metals nearly all of the energy transfer is by electrons. In dielectrics, there is also a transport of energy

by the lattice vibrations. However, the relative contribution of this latter mechanism is insignificant except for alloys, impure metals, and the semimetals like bismuth. The transfer of energy by electrons is impeded by several scattering mechanisms. At temperatures above about 20° K the main scattering agent is the metallic lattice itself. Below that temperature the scattering due to impurity centers and lattice defects becomes increasingly more important. In the temperature range from several degrees to about 40° K, the conductivity of a pure metal may be expressed closely by the equation

 $1/k = \alpha T^2 + \beta T^{-1}.$

The term αT^2 is characteristic of the lattice of the metal being investigated; the term βT^{-1} represents the scattering due to impurities. The latter term is related to the residual electrical resistance. Experimental values for α and β are given in the tables when the authors include these values in their research reports.

Several physical and chemical properties of the sample affect the conductivity directly. As the purity of the material is increased, the conductivity maximum rises and is shifted toward lower temperatures. The thermal resistance caused by impurities is not additive—small changes in purity can cause very large changes in the conductivity near the maximum. At higher temperatures, however, the effect is not as important. Cold-working and hardening reduce the conductivity and for that reason, other things being equal, the annealed samples will have a higher conductivity. For some single crystals the conductivity depends upon the direction of heat flow. For several metals that are anisotropic, curves are given for various crystal orientations.

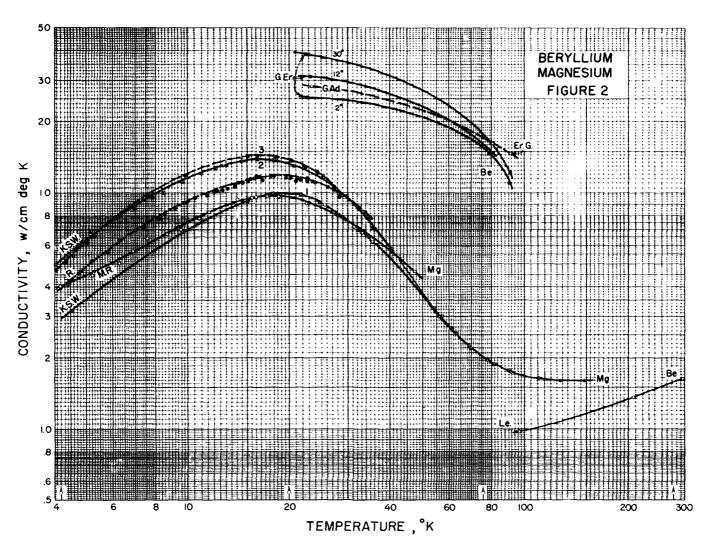


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Curve	Sample source and analysis	Remarks	Reference
Me	Kahlbaum;"very pure".	Cold-worked; handled in CO2 at- mosphere; R.	W. Meissner (1920).
Bi		Extruded and mounted in glass tubes; cycled thermally; R.	C. C. Bidwell (1926b, 1925, 1926a).
		POTASSIUM	
Но	Eimer and Amend; "very pure", free of Fe, Ca, Mg, Al, trace of Na, by chemical analysis.	Melted in vacuum; cast in glass under vacuum; R.	J. W. Hornbeck (1913).

SODIUM

Curve	Sample source and analysis	Remarks	Reference
	Eimer and Amend; "very pure" free of Fe, Ca, Mg, Al, and K by chemical analysis.	Melted in vacuum; cast in glass under vacuum; obtained k = 1.34 at 5.7°C, 1.33 at 21.0°C; R.	J. W. Hornbeck (1913).
Bi		Extruded and mounted in glass tubes; cycled thermally; R.	C. C. Bidwell (1926b, 1925, 1926a).
B.McD. 2.	British Thom- son-Houston; 0.01 to 0.1% Ca and Al.	Melted, cast in vacuum; cast into soft glass; R.	R. Berman and D. K. C. Mac- Donald (1951).
B.McD. 2	Philips; trace of Ag.	Melted, cast in vacuum; cast into soft glass; R.	R. Berman and D. K. C. Mac- Donald (1951).

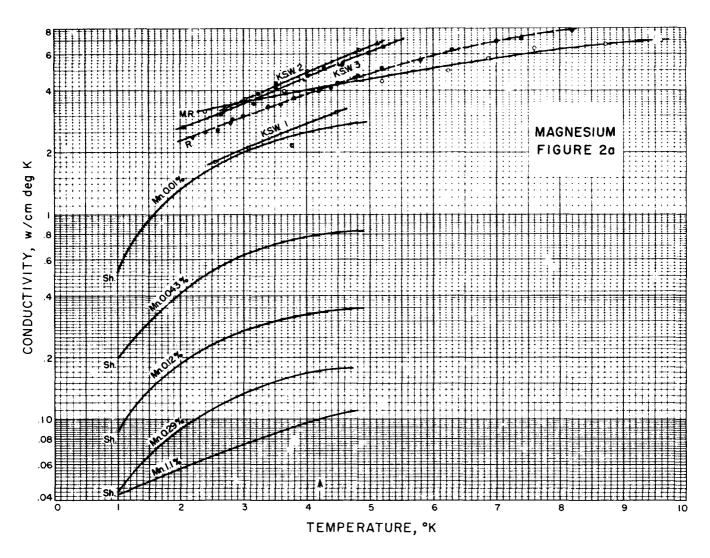


BERYLLIUM

Curve	Sample source and analysis	Remarks	Reference
Le	Beryllium Co. Am.; comm. pure; traces of Al, Mn, Cr, Fe, Si, and Mg: total im- purity ½%.	Physical imperfections noted; R, Cp, and emf.	E. J. Lewis (1929).
G. Ad	Degussa Co.; "high purity".	Residual resistance 1% of R272; single crystal with heat flow parallel to hexagonal axis; studied effect of magnetic field on R and k.	E. Grüneisen and H. Adenstedt (1938).
G. Er	do	Same; except rod axis perpendicu- lar to hexagonal axis; binary lat- eral axis inclined to rod axis by 2°, 12° and 30°; showed anisot- ropy.	E. Grüneisen and HD. Erfling (1940).
Er. G	do	Same as G. Ad	HD. Erfling and E. Grüneiser (1942).

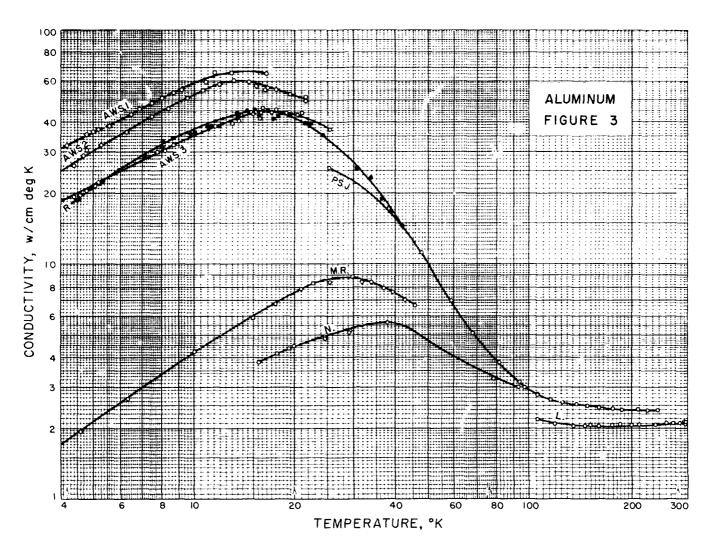
MAGNESIUM

Curve	Sample source and analysis	Remarks	Reference
	"Pure"	k=1.57 at0 °C; R	L. Lorens (1881a).
•••••	do	k=1.72 at 0°C, 2.0 at 80°K; R	J. Staebler (1929).
	do	k=1.72 at 0°C, 1.87 at 90°K; R	W. Mannchen (1931).
	do	k=1.60 at 18°C; R	R. Kikuchi (1932).
M. R	Johnson, Mat- they; 99.95% pure.	Equation $\alpha = 10.6 \times 10^{-5}$, $\beta = 1.25$.	K. Mendelssohn and H. M. Rosenberg. (1952a).
• • • • • • • • • • • • • • • • • • • •	do	Equation $\alpha = 8.6 \times 10^{-6}$, $\beta = 1.05$.	H. M. Rosenberg (1954a).
s	Dow; manga- nese imprui- ties as marked on graph.	R	E. G. Sharkoff (1952, 1953ab).



MAGNESIUM (Cont'd)

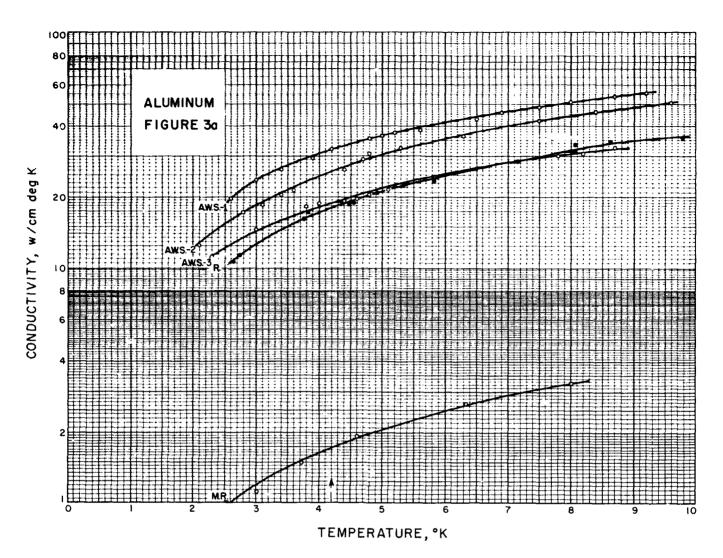
Curve	Sample source and analysis	Remarks	Reference
K. S. W. 1	Johnson, Mat- they; 99.98% purity; 013% Fe, .0023% Mn, .0013% Pb, trace of Si, Cu, Ag, Ca, Na.	Cold-drawn.	W. R. G. Kemp, A. K. Sreedhar, and G. K. White (1953).
K. 2	do	Annealed in vacuum 3 hr at 350°C.	Do.
K. S. W. 3	do	Same treatment as number 2	Do.
R	Johnson, Mat- they; 99.95% pure; .03% Mn, .0075% Fe, .004% Al.	Annealed 6 hr in vacuum at 500°C; equation $\alpha = 8.5 \times 10^{-5}$, $\beta = 1.05$.	H. M. Rosenberg (1954b).



ALUMINUM

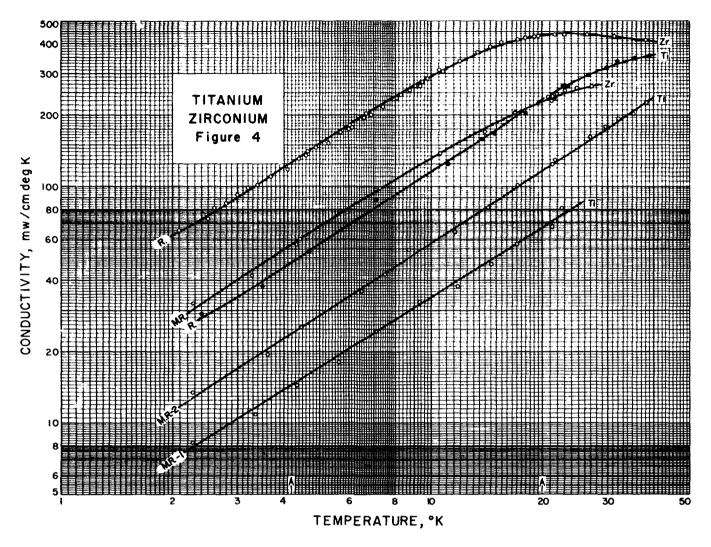
ALUMINUM (Cont'd)

Curve	Sample source and analysis	Remarks	Reference	Curve	Sample source and analysis	Remarks	Reference
	"Pure"	Low value of $k=1.43$ at 0°C; R	L. Lorens (1881a).	A. W. S. 1	Alcoa; 99.996% pure, .001%	Single crystal; residual electrical resistance of 1.19×10-4 R273;	R. A. Andrews, R. T. Webber,
*******	0.5% Fe, 0.4% Cu.	k=2.01 at 18°C; R, Cp	W. Jaeger and H. Diesselhorst (1900).		Mg, .001% Si, .0006% Fe, .0004% Cu, .004% Na.	$\alpha = 2.7 \times 10^{-6}, \ \beta = 7.04.$	and D. A. Spohr (1951).
L	Johnson, Mat- they; 99% pure.	Lathe turned from larger sample, density of 2.70.	C. H. Lees (1908).	A. W. S. 2	do	Single crystal, $R = 1.48 \times 10^{-3}$ R_{273} ; $\alpha = 2.72 \times 10^{-5}$, $\beta = 6.06$.	Do.
	Commercial	k=1.93 at 0°C, 1.90 at 85°K, 1.59 at 21.4°K.	R. Schott (1916).	A. W. S. 3.	Johnson, Mat- they; 99.995% pure; .002%	Polycrystalline rod; residual resistance of 2.14×10^{-3} R ₂₇₃ ; $\alpha = 2.72 \times 10^{-5}$, $\beta = 4.05$.	Do.
		Measured the effect of torsion on the thermal and electrical con- ductivity.	J. E. Calthrop (1926).		Mg, .001% Si, .0005% Fe, .0005% Cu, trace of Na.	,	
	5 samples rang- ing from pure to technical.	Measured at 20° and 80°K; results for purest samples lie just below curve of P. S. J.; studied effect of grain size and crystal bound- aries.	E. Grüneisen and E. Goens (1927); E. Grüneisen (1927).	P. S. J	Alcos; 99.99% pure.	Cold-drawn	R. W. Powers, D. Schwarts, and H. L. Johnston (1951).
•••••	"Pure"	k=2.26 at 0°C, 2.55 at 89°K; R	J. Staebler (1929).	M. R	Johnson, Mat- they;99.994% pure.	Annealed polycrystal; $\alpha = 2.2 \times 10^{-5}$, $\beta = 2.3$.	K. Mendelssobn and H. M. Rosenberg
••••	do	k=2.26 at 0°C, 2.56 at 80°K; also measured R.	W. Mannchen (1931).		pare.		(1952a).
•••••	Approx. 99.7% pure; technic- ally pure.	Two samples gave values at 0°C of $k=2.26$.	A. Eucken and H. Warrentrup (1935).			Polycrystalline; superconducting state; representative values were .07 at 0.8°K, .015 at 0.65°K, .007 at 0.37°K.	K. Mendelssohn and C. A. Ren- ton (1953).
N	Hadfield's	Brinnel hardness of 17	J. de Nobel (1951).	R.		$\alpha = 3.2 \times 10^{-5}, \ \beta = 0.23$	H. M. Rosen' erg (1954a)



LANTHANUM

Curve	Sample source and analysis	Remarks	Reference
		k=T/740 between about 2° and 20°K.	H. M. Rosenberg (1954a).

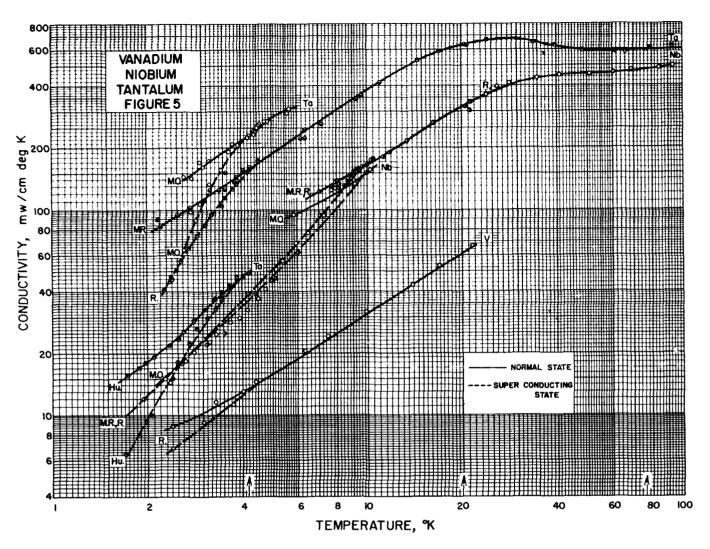


TITANIUM

Curve	Sample source and analysis	Remarks	Reference
	Comm. pure	Abstract only, k=0.20 at 273°K, 0.21 at 195°K, 0.18 at 90°K, 0.12 at 20°K.	C. J. Rigney and L. I. Bochstah- ler (1951).
M. R. 1	Assoc. Elect. Ind. Res. Lab., Eng- land; 99.9% pure.	Unannealed; <i>B</i> = 290	K. Mendelssohr and H. M. Rosenberg (1952b).
M. R. 2	Same source; 99.99% pure.	Annealed; B=170	Do.
B.		Single crystal; conductivity constant from 50° to 100°K; α= 454×10 ⁻¹ , β=82.	H. M. Rosenberg (1954a).

ZIRCONIUM

Curve	Sample source and analysis	Remarks	Reference
M. R	Johnson, Mat- they; 98% pure.	Annealed; $\alpha = 130 \times 10^{-6}$, $\beta = 76$	K. Mendelssohn and H. M. Rosenberg (1952b).
R		α=125×10 ⁻⁵ , β=34	H. M. Rosenberg (1954a).



VANADIUM

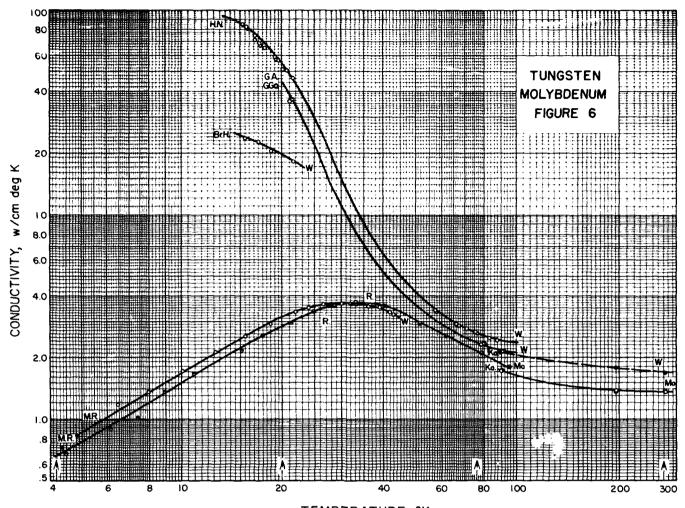
Curve	Sample source and analysis	Remarks	Reference
R		In both normal and superconducting states.	H. M. Rosenberg (1954a).

NIOBIUM

Curve	Sample source and analysis	Remarks	Reference
м. о	Hilger; "high purity".	In both normal and superconduct- ing states; studied effect of mag- netic field.	K. Mendelssohn and J. L. Olsen (1950a).
м. в	Johnson, Mat- they; 99.99% pure.	In both normal and superconduct- ing states; up to 22°K.	K. Mendelssohn and H. M. Rosenberg (1952b).
R	• • • • • • • • • • • • • • • • • • • •	Continuation to temperatures above 22°K.	H. M. Rosenberg (1954a).
•••••	Same as M.R.	Superconducting state below 1°K	K. Mendelssohn and C. A. Ren- ton (1953).

TANTALUM

Carve	Sample source and analysis	Remarks	Reference
• • • • • • • • • • • • • • • • • • • •		k=0.54 at 17°C	T. Barratt and R. M. Winte (1925).
•••••	Fansteel, 99.9% pure.	k=0.36 at 0°C	M. Cox (1943).
		Measured ratio of conductivity in superconducting and normal states.	C. V. Heer and J. G. Daunt (1949).
· · · · · · · · · · · · · · · · · · ·		do	J. K. Hulm (1949).
M. O	99.95% pure	Measured in both normal and su- perconducting states.	K. Mendelmoh and J. L. Olse (1950a).
Hu	Hilger; 0.1% impurities.	Polycrystalline; impurities in solid solution; measured effect of mag- netic field; both normal and su- perconducting states.	J. K. Hulm (1950).
M. R	Johnson, Mat- they; 99.98% purity.	Measured both normal and super- conducting states; $\beta = 27$.	K. Mendelssoh and H. M. Rosenberg (1952b).
	do	Measured superconducting state below 1°K.	K. Mendelmohr and C. A. Ren ton (1953).
R	do	Continued M. R. curve to higher temperatures; $\alpha = 79 \times 10^{-5}$, $\beta = 25$.	H. M. Rosenber (1954a).



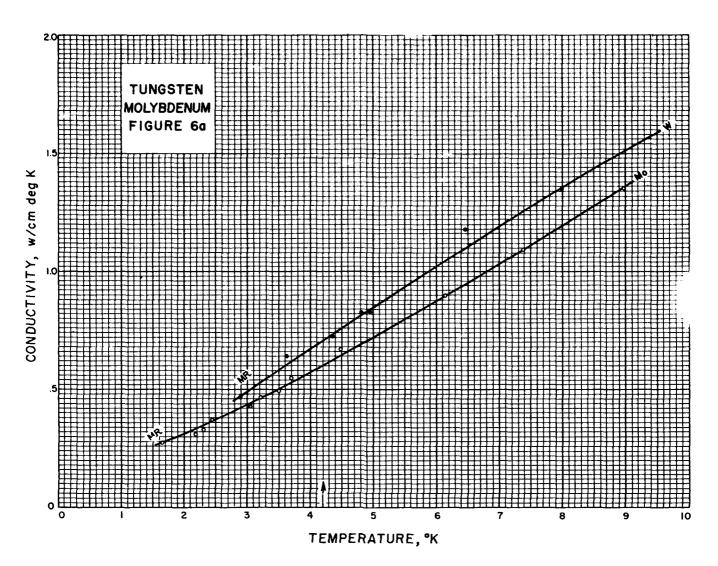
TEMPERATURE, °K

MOLYBDENUM

Curve	Sample source and analysis	Remarks	Reference
		k=1.45 at 17°C	T. Barratt and R. M. Winter (1925).
	Philips: "very pure".	Annealed at 900°C; k = 1.44 at 0°C; R.	W. G. Kannaluik (1931).
	Gen. Elec	Annealed at 220°C; k = 1.32 at 0°C; R.	Do.
Ka	.05% Bi, Cd; .01% Al, Ge, Sn, Ti, V, W; .001% Co, Cu, Pt, Rh; trace of C.	R	W. G. Kannaluik (1933).
M. R	Johnson, Mat- they; 99.95% pure.	α=7.5×10 ⁻⁵ , β=6.7	K. Mendelssohn and H. M. Rosenberg (1952b).
R.	do	Continued above work to 100°K	H. M. Rosenberg (1954a).

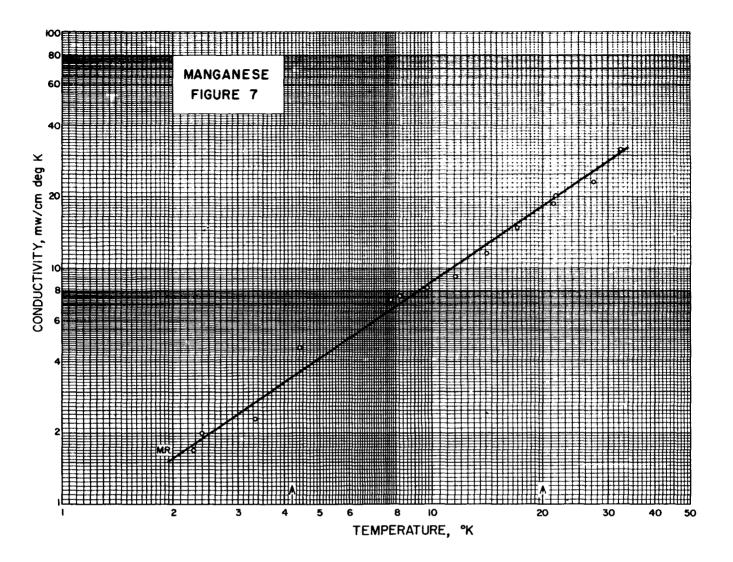
TUNGSTEN

Curve	Sample source and analysis	Remarks	Reference
	Heraeus	k=1.6 at 0°C	S. Weber (1917).
•••••		k=2 at 17°C	T. Barratt and R. M. Winter (1925).
	Osram; impure	Single crystal; k=1.83 at 83°K, 1.80 at 21°K.	E. Grüneisen and E. Goens (1927).
G. Go	Philips; "very pure".	Single crystal	Do.
	Gen. Elec	One sample annealed at 220°C, $k=1.64$ at 08°C; another sample annealed at 1300°C, $k=1.66$ at 18°C.	W. G. Kannaluik (1931).
Ka	Phillips	Single crystals, only higher values plotted.	W. G. Kannaluik (1933).
Br. H	Philips		H. Bremmer and W. J. de Haas (1936).



TUNGSTEN (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
•••••	Gen. Elec	k=1.66 at 0°C	I. Langmuir and J. B. Taylor (1936).
	• • • • • • • • • • • • • • • • • • • •	At 78°, 194°, 273°K, approx. same results as Kannaluk (1933).	W. C. Michela and M. Cox (1936).
G. A	Same as G. Go. above.	Studied effect of magnetic field and anisotropy.	E. Grüneisen and H. Adenstedt (1937).
H. N	Phillips	Single crystal; residual resistance of 4×10 ⁻⁴ R ₂₇₃ ; measured effect of magnetic field on k and R.	W.J. de Hass and J. de Nobel (1938).
G. A	Same as G. Go	Single crystals; graph results are for a sample with rod axis parallel to (0.10) crystal axis; for another crystal with rod axis parallel to (100) axis, $k=22.2$ at 21°K ; R.	E. Grüneisen and H. Adenstedt (1938).
•••••	Gen. Elec	k=1.93 at 77°K, 1.87 at 90°K, and 1.69 at 0°C.	M. Coz (1943)
	Same as H. N. above.	Extended the measurements to higher magnetic fields.	J. de Nobel (1949).
M. R	Johnson, Mat- they; 99.99% pure.	Annealed; $\alpha = 10.2 \times 10^{-5}$, $\beta = 5.9$.	K. Mendelssohn and H. M. Rosenberg (1952b).
R		$\alpha = 9.3 \times 10^{-5}$, $\beta = 5.8 \dots$	H. M. Rosenberg (1954a).

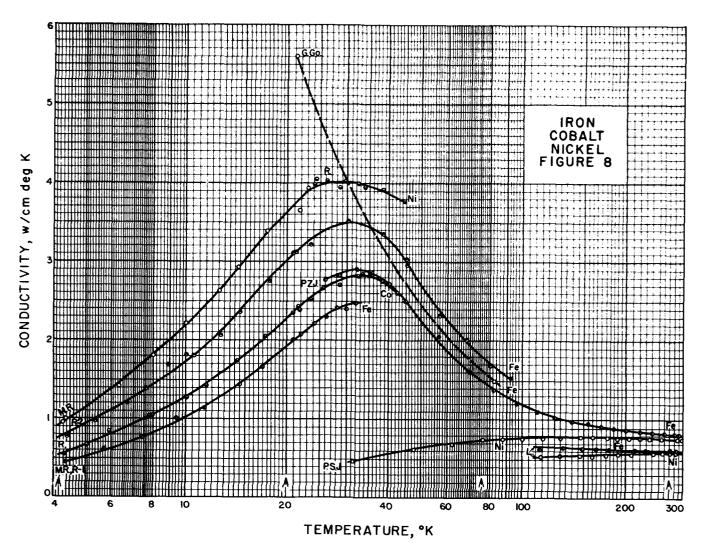


MANGANESE

Curve	Sample source and analysis	Remarks	Reference
		k=0.05 at 83°K for the B phase	H. Reddemann (1935).
M. R	Johnson, Mat- they; 99.99% pure.	Annealed; β = 1200	K. Mendelmohn and H. M. Rosenberg (1952b).

SUPPLEMENTARY DATA

Curve	Sample source and analysis	Remarks	Reference
			İ

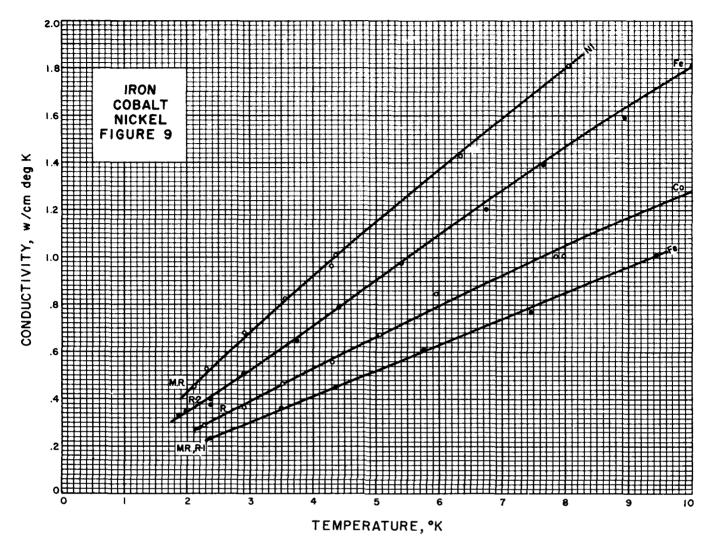


IRON

Curve	Sample source and analysis	Remarks	Reference
	"Pure"	k=0.70 at 0°C	L. Lorens (1881a).
	"Pure", .1% C, .06% Mn, .02% Si, .05% Cu, .03% P, .03% P, .02% .02% 8.	k=0.72 at 18°C	E. Grüneisen (1900).
	0.1% C+metals	Also measured R, C_D , emf; $k=0.67$ at 18°C.	W. Jaeger and H. Diesselhorst (1900).
•••••	Krupp; .1% C, 2% Si, .1% Mn.	Also measured R, C_p , emf; $k=0.60$ at 18°C.	Do.
L	99.42% pure; .1% C, .15% Ma, .13% Si.	Wrought iron	C. H. Lees (1908).
		Electrolytic; two rods with average grain sizes of 1×10^{-1} and 6×10^{-2} cm; $k=0.94$ and 0.90 , respectively, at 0° C; $k=1.84$ and 1.83 at 80° K.	A. Eucken and K. Dittrich (1927).
	Heraeus	Electrolytic; average grain size 2× 10 ⁻² cm; k=0.82 at 0°C and 1.17 at 80°K.	Do.
G. Go	"Double re- fined.	Tempered; electrolytic	E. Grüneisen and E. Goens (1927).

IRON (Cont'd)

Curve	Sample source and analysis	Remarks	Reference E. Grüneisen and E. Goens (1927).	
	"Technically pure".	Two samples untempered; electrolytic; $k=1.36$ and 0.91 at 83°K, 3.01 and 0.5 at 21°K.		
• • • • • • • • • • • • • • • • • • • •	"Pure"	Electrolytic; k=0.77 at 16°C	R. Kikuchi (1932).	
	Armeo; .01% C, .02% Mn, .006% P, .026% S, .06% Ca, .02% Si.	k=0.7 at 0°C, 0.72 at 195°K, 0.94 at 90°K	W. G. Kannaluik (1933).	
•••••	"Pure"	Between 3° and 20°K, the values fall just below the curve marked M. R.	J. Karweil and K. Schäfer (1939).	
• • • • • • • • • • • • • • • • • • • •	Hadfield; 99 93% pure.	Forged; k=0.9 at 90°K, maximum of 1.3 at 52°K, 0.5 at 15°K.	J. de Nobel (1951).	
P. Z. J	Johnson, Mat- they; 99.99% pure.		R. W. Powers, J. B. Ziegler and H. L. Johnston (1951a).	
M. R	Johnson, Mat- they; 99.99% pure.	α=18×10 ⁻⁵ , β=9.5	K. Mendelssohr and H. M. Rosenberg (1952b).	
R		$\alpha = 10.2 \times 10^{-5}, \beta = 9.6$	H. M. Rosenber (1954a).	

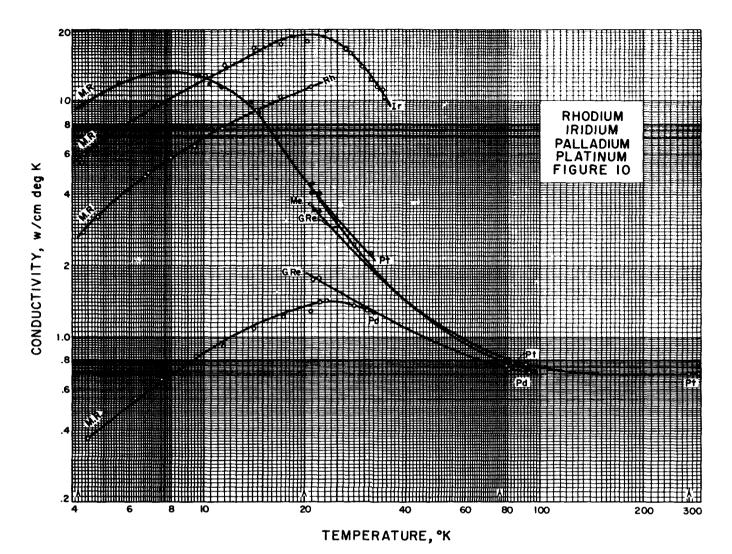


WITH A THE WAY	
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Curve	Sample source and analysis	Remarks	Reference
	Basse and Selve; impure.	k=0.60 at 18°C; R, Cp, emf	W. Jaeger and H. Diemelhorst (1900).
L	Johnson, Mat- they; 99% pure.	Lathe turned; density 8.80	C. H. Lees (1908).
	Heraeus	Drawn rod; k=0.84 at 0°C, 1.11 at 80°K.	A. Eucken and K. Dittrich (1927).
P 9. J	Int. Nickel; comm. pure.		R. W. Powers, D. Schwarts, and H. L. Johnston (1951).
	99.4% pure	Forged; approx. same curve as P. S. J. from 93° to 25°K; at 15° K, k=0.18.	J. de Nobel (1951).
M. R	Johnson, Mat- they;99.997% pure.	Annealed; $\alpha = 22 \times 10^{-5}$, $\beta = 4.4$; measured up to 22°K.	K. Mendelssohn and H. M. Rosenberg (1952b).
R	do	Annealed; $\alpha = 10.4 \times 10^{-5}$, $\beta = 4.6$; measured up to 40°K.	H. M. Rosenberg (1954a).

COBALT

Curve	Sample source and analysis	Remarks	Reference	
R.		$\alpha = 10.5 \times 10^{-6}, \beta = 7.9$	H. M. Rosenberg (1954a).	

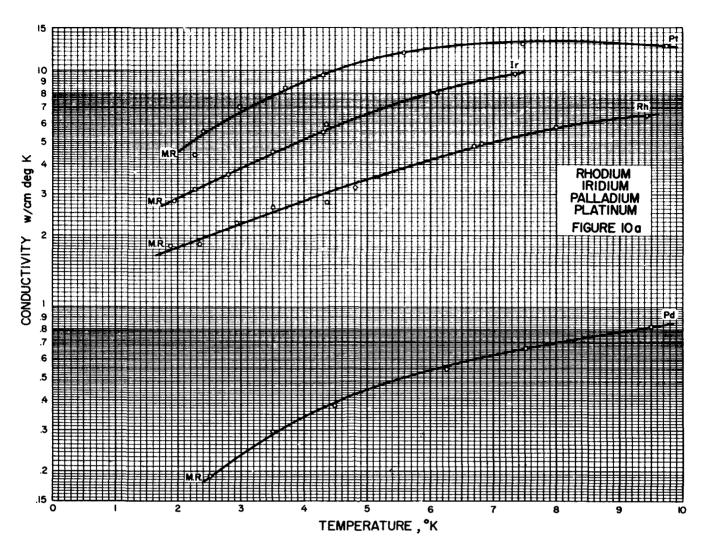


PALLADIUM

Curve Sample source and analysis		Remarks	Reference	
	"Chem. pure"	k=0.7 at 18°C; R, Cp, emf	W. Jacger and H. Diesselhors (1900).	
		k=0.42 at 17°C for commercial palladium, 0.60 at 17°C for "pure".	T. Barratt and R. M. Winter (1925).	
G. Re	Heraeus; "pure".	Unannealed; plotted with open circles; R.	E. Grüneisen and H. Reddemann (1934).	
G. Re	do	Cold-drawn; annealed at 360°C for two hours; plotted with dark- ened circles; R.	Do.	
M. R	Johnson, Mat- they; 99.95% pure.	Annealed; $\alpha = 64 \times 10^{-5}$, $\beta = 11$	K. Mendelssohn and H. M. Rosenberg (1952b).	
•••••		$\alpha=41\times10^{-6}, \beta=11.7$	H. M. Rosenber (1954a).	

RHODIUM

Curve Sample source and analysis		Remarks	Reference
• • • • • • • • • • • • • • • • • • • •		k=0.88 at 17°C	T. Barratt and R. M. Winter (1925).
• • • • • • • • • • • • • • • • • • • •	Heraeus; "pure".	Annealed; k=2.15 at 83°K, 23.8 at 21°K, R.	E. Grüneisen and E. Goens (1927).
M. R	Johnson Mat- they; 99.95% pure.	α=22×10 ⁻⁵ , β=1.4	K. Mendelssohn and H. M. Rosenberg (1952b).
		$\alpha = 10.7 \times 10^{-5}, \beta = 1.38$	H. M. Rosenberg (1954a).

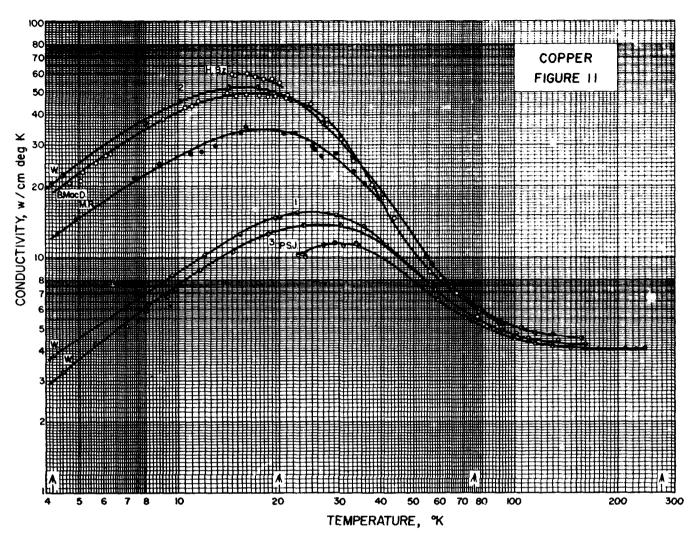


PLATINUM

Curve	Sample source and analysis	Remarks	Reference
	"Pure"	k=0.78 at 18°C	J. H. Gray (1895).
•••••	do	k=0.7 at 18°C; R, Cp, conf	W. Jacger and H. Diesselhorst (1900).
Ме	Heracus; "very pure".	Drawn; electrically annealed	W. Meissner (1915).
•••••	do	Drawn; electrically annealed; obtained same results as Meissner (1915) at 21° and 83°K; R.	E. Grüneisen and E. Goens (1927).
• • • • • • • • • • • • • • • • • • •	Heraeus; "less pure".	k=2.96 at 21°K	Do.
•••••	Heraeus	k=4.25 at 21°K; measured effect of magnetic field on k, R.	E. Grüneisen and H. Adenstedt (1938).
M. R	Johnson, Mat- they; 99.999% pure.	α=43×10 ⁻⁵ , β=0.40	K. Mendelmohn and H. M. Rosenberg (1952b).
• • • • • • • • • • • • • • • • • • • •		$\alpha = 43 \times 10^{-5}, \ \beta = 0.35$	H. M. Rosenberg (1954a).

RIDIUM

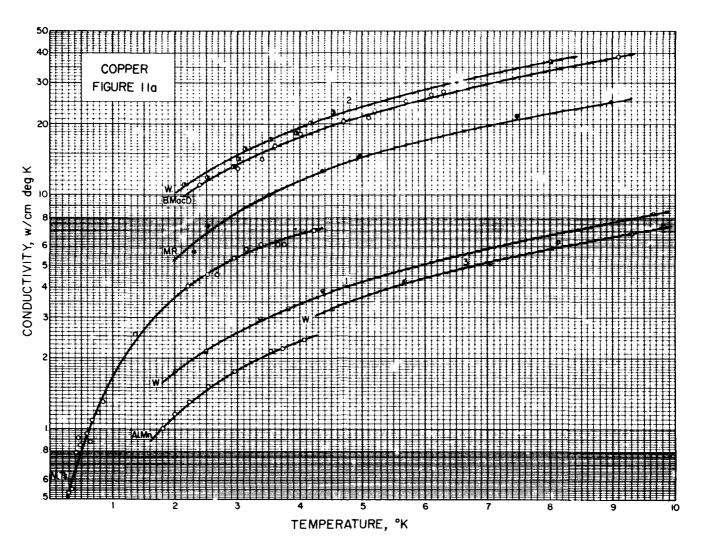
Curve	Sample source and analysis	Romarks	Reference	
		k=0.59 at 17°C	T. Barratt and R. M. Winter (1925).	
M. R	Johnson, Mat- they; 99.95% pure.	Annealed; $\alpha = 3.6 \times 10^{-4}$, $\beta = 0.77$.	K. Mendelssohn and H. M. Rosenberg (1952b).	
R		α=4.6×10 ⁻⁶ , β=0.75	H. M. Rosenberg (1954a).	



COPPER

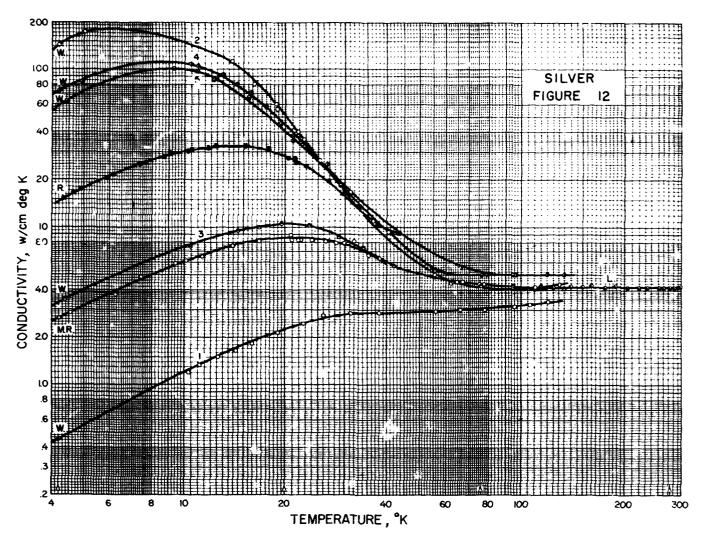
COPPER (Cont'd)

Carve	Sample source and analysis	Remarks	Reference	Curve	Sample source and analysis	Remarks	Reference
	"Pure"	k=3 at 0°C	L. Lorens (1881a).			Measured 18 samples of various crystal structure, purity, and annealing at 21° and 83°K: R.	E. Grüneisen and E. Goens (1927).
	"Pure"	One sample had k=3.6 at 10°C; the second, 1.3 at 10°C. k=3.9 at 18°C.	J. H. Gray (1895).		Gen. Elec	Single crystal; between 95° and 300°K, the results are close to	W. G. Kannaluik and T. H. Laby
	do	k=3.9 at 18°C; R, Co	E. Grüneisen (1900). W. Jaeger and			curve W1. Measured 14 different copper sam- ples at 20° and 90°K; R.	(1928). E. Grüneisen and H. Reddemann
		, , ,	H. Diemelhorst (1900).	R. Bs		pres as 20 and 50 k; 16.	(1934). W.J.de Hassand
•••••	do	k=3.95 at 20°C	W. Schaufel- berger (1902). C. H. Lees			Oundred offices of accounting cold and	T. Biermass (1936). E. Grüneisen and
		$k=3.8$ at 27° C; results at 100° K are close to the P.S.J. curve.	(1908).	•••••		Studied effect of magnetic field on k, R.	H. Adenstedt (1938).
		Electrolytic copper wires; values at 21°, 91° and 273° are close to the W1 curve.	W. Meissner (1915).	Al. Mn	Johnson, Mat- they; free of 02; .003% each Ag, Ni, and	Machined and annealed	J. F. Allen and E. Mendosa (1947).
	"Very pure"	Natural single crystal; results un- certain due to very small size of sample.	R. Schott (1916).	P. S. J	Pb. Am. Brass; "O. F. H. C."	Oxygen-free, high conductivity	R. W. Powers, D. Schwartz,
•••••	"Tech. pure"	Approximately on curve of W3 down to 22°K.	Do.				and H. L. Johnston (1951).



COPPER (Cont'd)

Curve	Sample source and analysis	Remarks	Reference	
B. McD	Johnson, Mat- they; .0005% Ag, .0003% Ni, .0004% Pb.	Cold-drawn, then annealed 6 hr at 450°C in helium gas.	R. Berman and D. K. C. Mac- Donald (1952).	
M. R	Johnson, Mat- they; 99.999% pure.	Annealed; $\alpha = 3.2 \times 10^{-5}$, $\beta = 0.35$.	K. Mendelssohn and H. M. Rosenberg (1952a).	
Ni. Ta	Gen. Elec.; "comm. high purity".	Polycrystalline wire	J. Nicol and T. P. Tseng (1953).	
W1	Johnson, Mat- they; 99.999% pure; same as B. MacD.	As cold-drawn; $\alpha = 2.55 \times 10^{-6}$, $\beta = 1.15$.	G. K. White (1953c).	
₩2	do	Annealed in vacuum at 550°C for 3 hours; $\alpha = 2.55 \times 10^{-3}$, $\beta = 0.21$.	Do.	
W3	do	As cold-drawn; same as W1	Do	
R		$\alpha = 2.5 \times 10^{-8}, \ \beta = 0.35 \dots$	H. M. Rosenberg (1954a).	

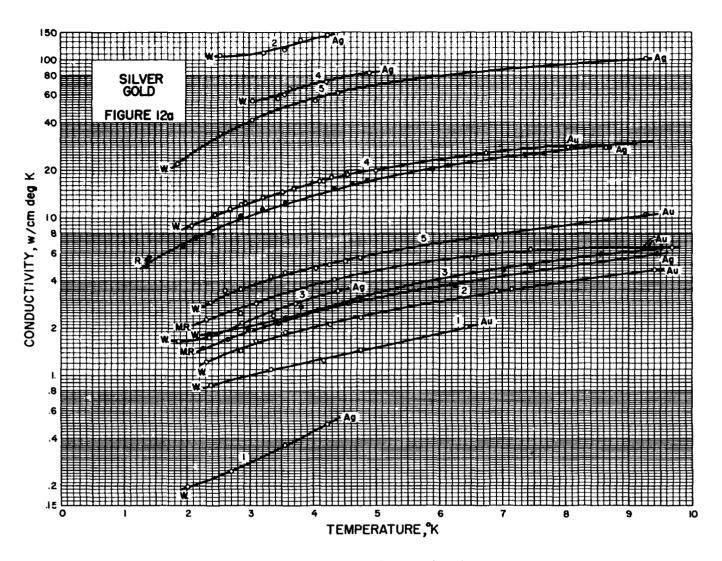


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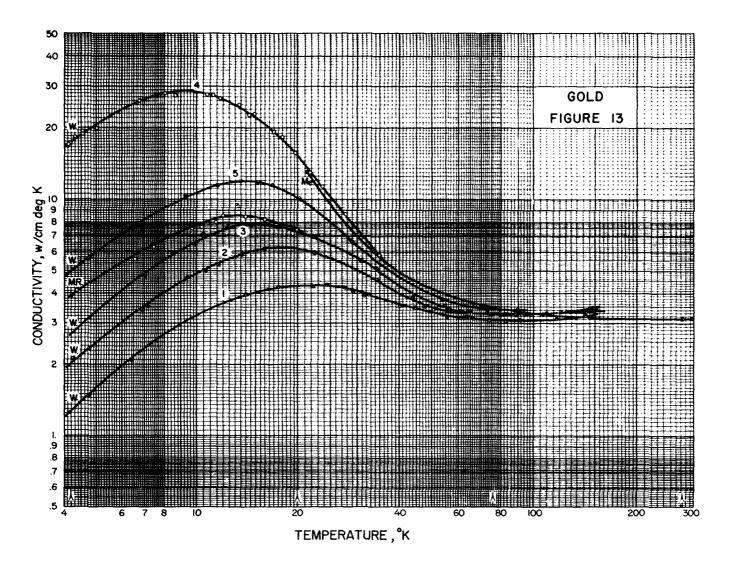
Curve	Sample source and analysis	Remarks	Reference
		Silver wire; k = 4.02 at 18°C	J. H. Gray (1895).
•••••	99.98% pare	k=4.21 at 18°C; also measured R, Cp, and emf.	W. Jaeger and H. Diesselhorst (1900).
L	Johnson, Mat- they; 99.9% pure.	Lathe-turned from a larger rod	C. H. Lees (1908).
•••••		Two silver wires had k=4.11 and 4.04 at 0°C.	W. G. Kannaluik (1931).
•••••	Hilger; trace of Cu, Pb, Bi, Mg, Ca, Na, Si.	At 90°, 195°, and 273°K values are somewhat higher than those of Lees (1908).	W. G. Kannaluik (1933).
•••••		Five rods of silver, varied in com- position, annealing, crystal structure. At 20° and 90°K pure rods had values close to curve W3; R.	E. Grüneisen and H. Reddemann (1934).

SILVER (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
	Hönig-schmid	Annealed; electrolytic; k=31.4 at 21°K; measured effect of magnetic field on k and R.	E. Grüneisen and H. Adenstedt (1938)
M. R	Johnson, Mat- they; 99.99% pure.	$\alpha = 9.0 \times 10^{-8}, \ \beta = 1.6.$	K. Mendelssohn and H. M. Rosenberg (1952a).
	do	Measured effect of magnetic field	K. Mendelssohn and H. M. Rosenberg (1953).
W. 1-5	Johnson, Mat- they; 99.999% pure,	No. 1 was unannealed; #2, annealed at 650°C, grain size 0.1 mm; #3, cold-drawn; #4, the previous one annealed; #5, a rerun of #4.	G. K. White (1953b).
R	Johnson, Mat- they.	$\alpha=5\times10^{-5},\ \beta=0.3$	H. M. Rosenberg (1954a).



(See next page for the table on GOLD.)

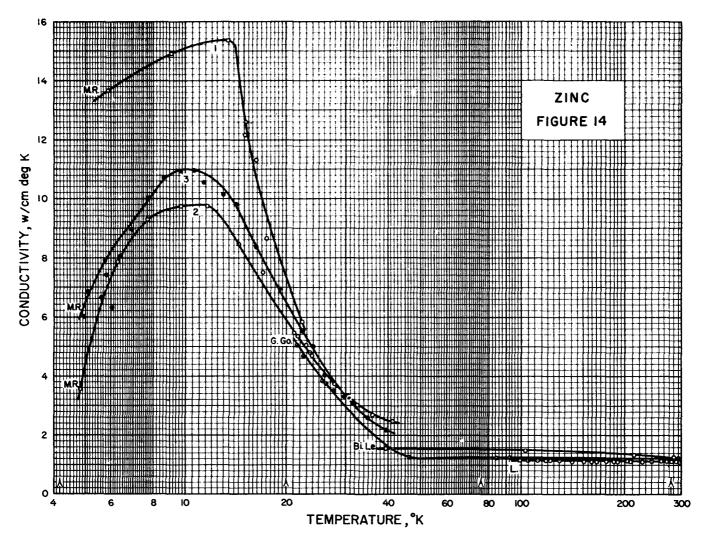


GOLD

Curve	Sample source and analysis	Remarks	Reference
		k=3.14 at 18°C	J. H. Gray (1895).
	"Pure"	k=2.93 at 18°C; a less pure sample had k=1.79 at 18°C; R, Cp, emf.	W. Jaeger and H. Diesselhorst (1900).
Me	Mylius; 99.999% pure.	Cold-drawn; annealed	W. Meissner (1915).
		k=2.95 at 17°C	T. Barratt and R. M. Winter (1925).
•••••		k=2.98 at 24°C	H. Masumoto (1927).
•••••		Six samples of various composi- tion, annealing; R. Results for "very pure" gold at 21° and 83° K fall close to curve W4.	E. Grüneisen and E. Goens (1927).
		k=3.06 at 0°C	W. G. Kannaluik (1931).

GOLD (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
M. R	Johnson, Mat- they;99.999% pure.	$\alpha \approx 18 \times 10^{-5}, \ \beta = 1.15$	K. Mendelssohn and H. M. Rosenberg (1952a).
W. 1, 2	Garrett, David- son, Matthey; 99.9% pure (comm.); Ag, trace of Pt, Fe, Pb, Cu, Sn.	No. 1 sample unannealed; #2, an- nealed.	G. K. White (1953a).
W. 3, 4, 5	Johnson, Mat- they; 99.999% pure; trace of Ag, Cu; faint trace of Cd, Fe, Mg, Na, Ca, Zn.	No. 3 sample cold-drawn; #4, annealed in vacuum at 700°C for 3 hours; #5 was the fourth redrawn.	Do.
		$\alpha = 19 \times 10^{-5}, \beta = 1.13$	H. M. Rosenberg (1954a).

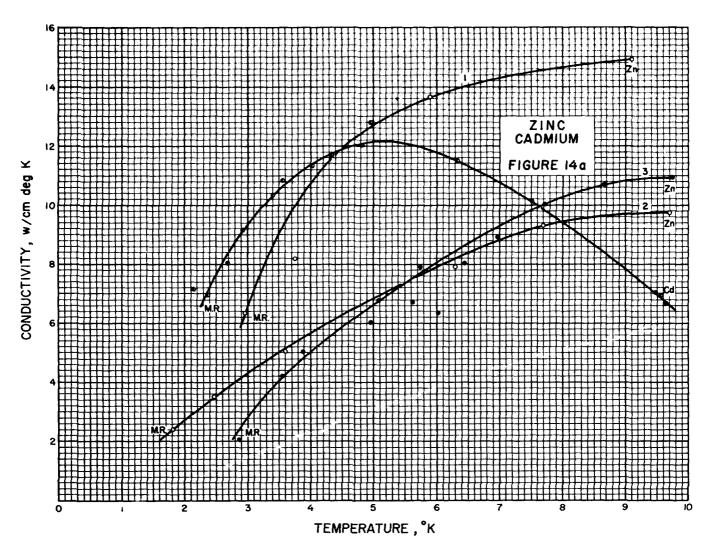


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Curve	Sample source and analysis	Remarks	Reference -
	"Pure"	k=1.1 at 18°C; R, Cp, emf	W. Jaeger and H. Diesselhorst (1900).
L	"Pure redis- tilled".	Lathe-turned from a cast stick	C. H. Lees (1908).
Bi. Le	99.993% pure	Single crystal; also measured poly- crystalline samples.	C. C. Bidwell and E. J. Lewis (1929); also E. J. Lewis and C. C. Bidwell (1928).
	Kahlbaum	k=1.26 at 83°K and 1.25 at 0°C	J. Staebler (1929).
G. Go	Kahlbaum; "pure".	Single crystals each with rod axis parallel to main crystal axis. An- other sample with axes perpen- dicular had a conductivity 10% lower.	E. Goens and E. Grüneisen (1932).

ZINC (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
M. R. 1	Hilger; 99.995% pure.	Polycrystalline; $\alpha = 21 \times 10^{-6}$, $\beta = 0.4$.	K. Mendelssoh and H. M. Rosenberg (1952a).
M. R. 2, 3.	Imperial Smelt- ing; 99.997%.	No. 2 had rod axis inclined 80° to hexagonal crystal axis, $\alpha = 34 \times 10^{-5}$, $\beta = 0.7$; #3, inclined 13°, $\alpha = 31 \times 10^{-5}$, $\beta = 0.6$.	Do.
• • • • • • • • • • • • • • • • • • • •	Same as M. R. 1, 2, 3.	Measured effect of magnetic field, .	K. Mendelsoob and H. M. Rosenberg (1953).
	•••••	$\alpha = 30 \times 10^{-5}, \ \beta = 0.6$	H. M. Rosenber (1954a).



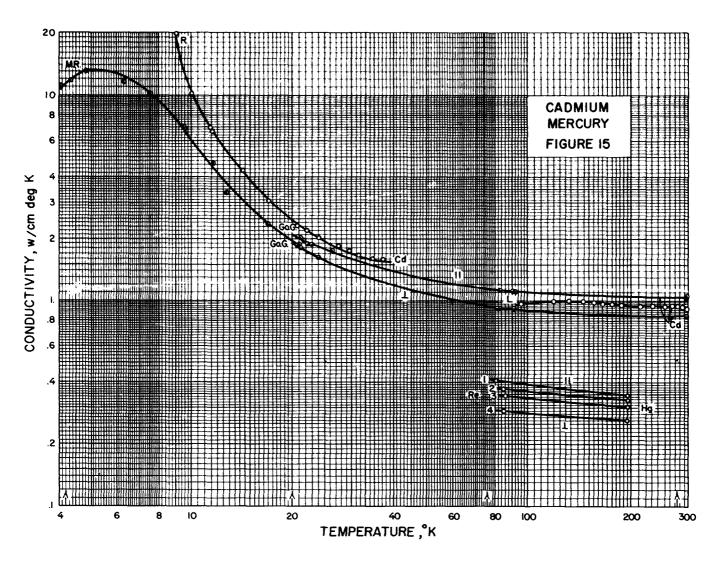
(See previous page for the table on ZINC.)

CADMIUM

Curve	Sample source and analysis	Remarks	Reference
	"Pure	k=0.92 at 0°C	L. Lorens (1881).
	do	k = 0.93 at 18°C; R, Cp, emf	W. Jacger and H. Diesselhorst (1900).
L	"Pure redis- tilled".	Lathe-turned from a cast stick	C. H. Lees (1908).
	Kahibaum; "pure".	k=1.02 at 273° and 194°K, 1.23 at 83°K.	A. Eucken and G. Gehlhoff (1912).
*********	Kahihaum; "chem. pure".	At 20° and 273°K values fall just below upper curve of Go. G.	R. Schott (1916).

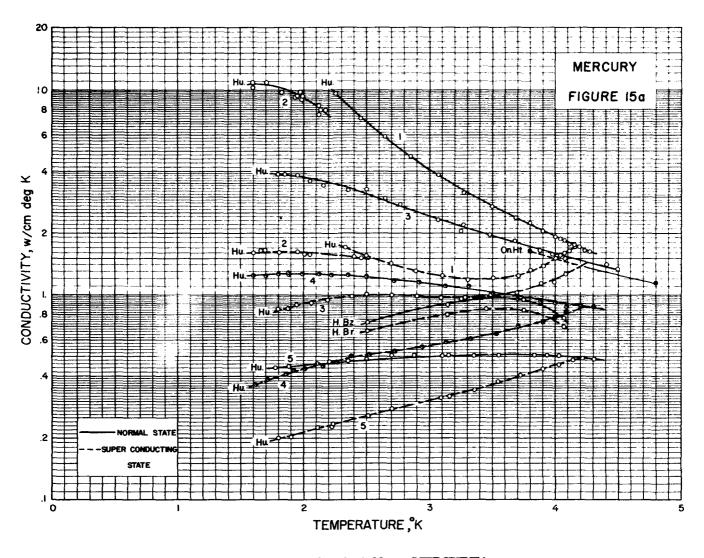
CADMIUM (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
Go. G	Kahlbaum; "pure".	Two single crystals, each with main crystal and rod axes par- allel.	E. Goens and E. Grüneisen (1932).
Go. G	do	Single crystal with main crystal and rod axes perpendicular.	Do.
•••••	Hilger; 99.999% pure.	Measured effect of magnetic field	K. Mendelssohn and H. M. Rosenberg (1951).
M. R	Hilger; 99.9999% pure.	Cast in glass; $\alpha = 140 \times 10^{-5}$, $\beta = 0.5$.	K. Mendelssohn and H. M. Rosenberg (1952a).
	do	Measured effect of magnetic field	K. Mendelssohn and H. M. Rosenberg (1953).
R		Maximum conductivity of 88 between 4° and 5°K; $\alpha = 122 \times 10^{-4}$, $\beta = 0.02$.	H. M. Rosenberg (1954a).

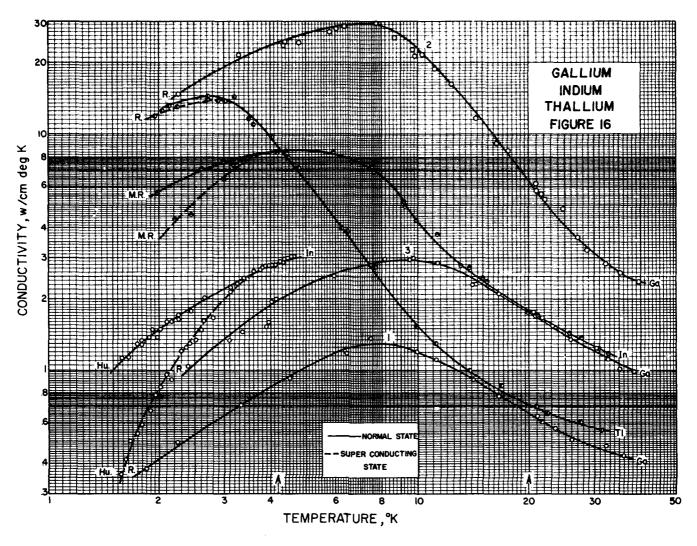


MERCURY

Curve	Sample source and analysis	Remarks	Reference
On. Ht		Ср	H. Kamerlingh Omnes (1914).
• • • • • • • • •		Measured in the liquid state and in solid state near melting.	G. Gehlholf and F. Neumeier (1919).
Re		Measured ten single crystal rods; fall into four groups. No. 1, rod axis parallel to crystal axis; #2, axes inclined 25°; #3, axes in- clined 45°; #4, axes perpen- dicular.	H. Reddemann (1932).
H. Ec		Measured in both normal and su- perconducting states.	W.J. de Haas and H. Bremmer (1936).
Ru. 1-5	Basic rod (#1) from Johnson, Matthey; #2, .002% Cd; #3, .007% Cd; #4, .10% In; #5, .39% In.	Homogeneous solid solutions; poly- crystalline, but large crystals; measured in both normal and superconducting state.	J. K. Hulm (1950).
•••••		Measured in the intermediate state near 4 K.	R. T. Webber and D. A. Spohr (1953).



(See previous page for the table on MERCURY.)



GALLIUM

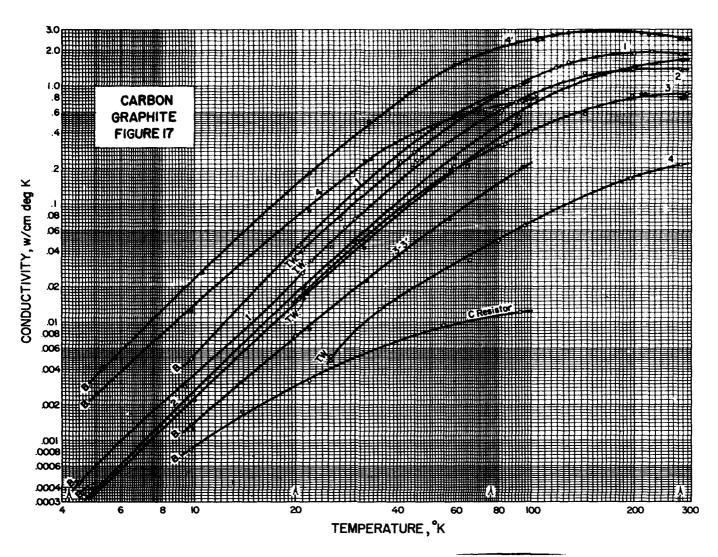
Curve	Sample source and analysis	Remarks	Reference
R		Three single crystals of different orientation; No. 1 $\alpha = 160 \times 10^{-5}$, $\beta = 4.7$; #2, $\alpha = 23 \times 10^{-5}$, $\beta = 0.165$; #3, $\alpha = 87 \times 10^{-5}$, $\beta = 2.22$.	H. M. Rosenberg (1954a).

THALLIUM

	Kahlbaum	Drawn; k=0.51 at 0°C and 0.64 at 80°K	A. Eucken and K. Dittrich (1927).
•••••	Johnson, Mat- they; 99.99% pure.	Annealed; polycrystalline; measured effect of magnetic field.	K. Mendelssohn and H. M. Rosenberg (1953).
	do	Annealed; polycrystalline; measured conductivity below 1°K; between 0.3° and 0.65°K, conductivity was of form in k=aT; k=0.2 at 0.62°K, 0.0015 at 0.3°K.	K. Mendelssohn and C. A. Renton (1953).
R	do	$\alpha = 537 \times 10^{-6}, \beta = 0.1$	H. M. Rosenberg (1954a).

INDIUM

Curve	Sample source and analysis	Remarks	Reference
	Hilger	Absolute values were not deter- mined.	W. J. de Haas and H. Bremmer (1932a).
Ни	Johnson, Mat- they; 0.1% impurity.	α=189×10 ⁻⁵ , β=1.38	J. K. Hulm (1950).
•••••	Johnson, Mat- they.	Single crystal; measured conduct- ivity in intermediate state.	D. P. Detwiler and H. A. Fairbank (1952a, b).
M. R	Johnson, Mat- they; 99.993% pure.	α=190×10 ⁻⁵ , β=0.4; measured in both normal and superconducting state.	K. Mendelssohn and H. M. Rosenberg (1952a).
•••••	do	Measured effect of magnetic field	K. Mendelssohn and H. M. Rosenberg (1953).
	do	Measured conductivity below 1°K; between 0.2° and 0.7°K, conductivity equation was $k=2.5 \times 10^{-2}$ Ts; $k=.019$ at 0.8°K, .003 at 0.46°K.	K. Mendelesohn and C. A. Renton (1953).
R		α =185×10 ⁻⁵ , β =0.35 for normal state conductivity.	H. M. Rosenberg (1954a).

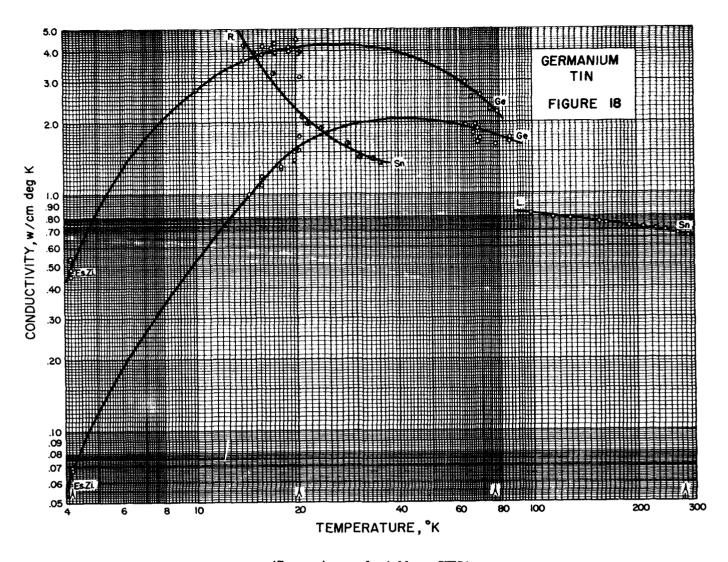


CARRON	(Graphite)
	(UII DEUILIND)

Curve	Sample source and analysis	Remarks	Reference
• • • • • • • •		Pencil lead; density of 2.11 g/cm ³ ; k=0.15 at 18°C.	T. Barratt and R. M. Winter (1925).
	Acheson graph- ite.	k=1.78 at 122°K, 1.72 at 300°K	A. P. Crary (1933).
•••••	Carbon; 80% petroleum coke, 20% hampblack.	k=0.016 at 0°C	R. W. Powell, F. H. Schofield (1939).
	Acheson graph- ite.	Two rods gave k=1.21 and 1.67 at 0°C.	Do.
	National Car- bon; Acheson graphite.	For a sample with rod axis parallel to extrusion direction, $k=1.76$ at 0°C, 2.5 at 82°K; for a second sample with rod axis perpendic- ular to extrusion, $k=1.13$ at 0°C; 1.76 at 82°K.	R. A. Buerschaper (1944).
•••••	National Car- bon; carbon electrode.	k=0.06 at 0°C, 0.01 at 82°K	Do.
•••••		Measured conductivities of graph- ite and amorphous carbon.	S. Misushima and J. Okada (1951).
		Measured effect of crystal size	S. Mrosowski (1952).

CARBON (Graphite) (Cont'd)

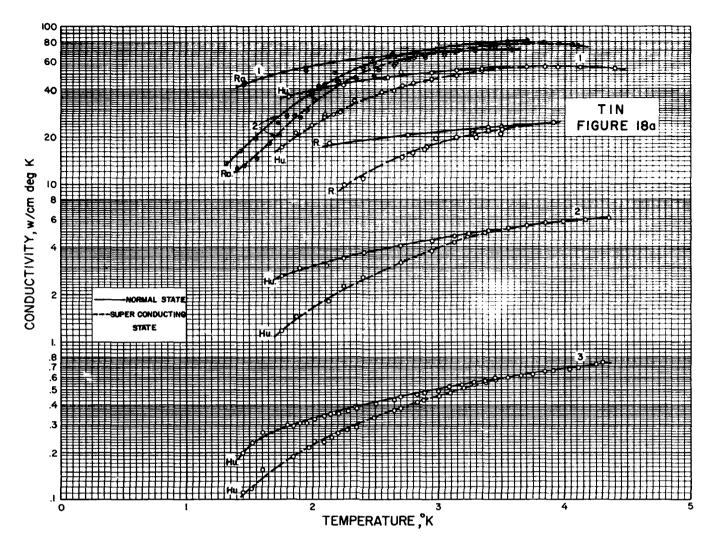
Curve	Sample source and analysis	Remarks	Reference
B. 1-3, 1'- 3'.	Atomic Energy Res. Estab- lish.	Artificial graphite rods; very anisotropie; unprimed numbers refer to rods with axes parallel to direction of extrusion; primed numbers refer to rods with axes perpendicular to the extrusion; densities were respectively 1.79, 1.60, and 1.77 g/cm²; crystal sizes 2000, 1000, and 300 A.	R. Berman (1952).
B. 4, 4'	Natural graph- ite.	Density 2.25 g/cm ² ; crystal sise, 2000 A; unprimed number re- fers to sample with its rod axis parallel to preferred c-axis; primed number, perpendicular.	Do.
T. W. 1, 2.	National Car- bon; graphite.	Densities were 1.70 g/cm ² ; rod axes were respectively perpendicular and parallel to the preferred c-axes.	W. W. Tyler and A. C. Wilson (1953).
T. W. 3	Natural graph- ite.	Molded; density of 1.80; rod axis perpendicular to preferred c-axis.	Do.
T. W. 4	Lampblack	Molded; density of 1.65; rod axis parallel to preferred c-axis.	Do.
		Abstract only; data not given	A. W. Smith (1954).
		SILICON	
	Impurities of 1 ×10 ⁻⁵ percent as shown by Hall effect.	"Filament cut from a crystal pulled in [100] direction;" k=1.48 at 0°C, 9 at 80°K, 19.5 at 30°K.	G. W. Hull and T. H. Gebalk (1954).



(See next page for table on TIN.)

GERMANIUM

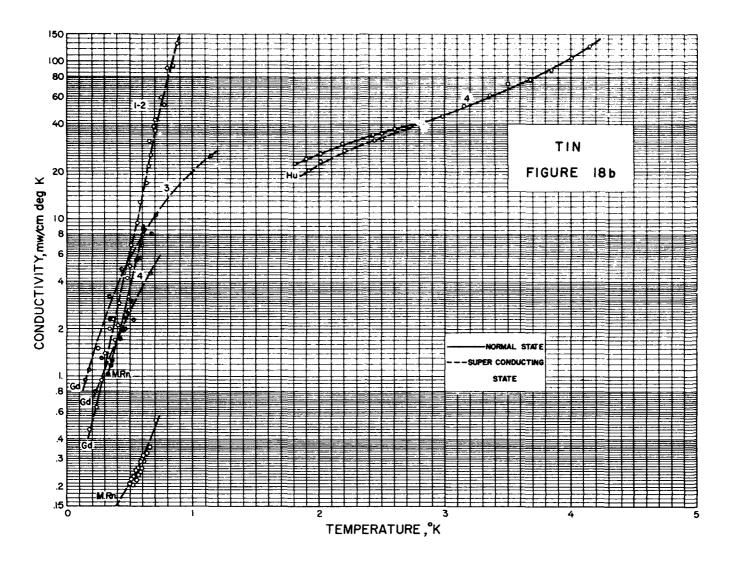
Curve	Sample source and analysis	Remarks	Reference
Es. Zi	"High purity"	Cast; higher of two Ge curves on figures 18 and 19a.	I. Estermann and J. E. Zimmer- man (1951).
Es. Zi	0.006 atomic % of Al.	Cast; lower of two Ge curves on figures 18 and 19s.	Do.

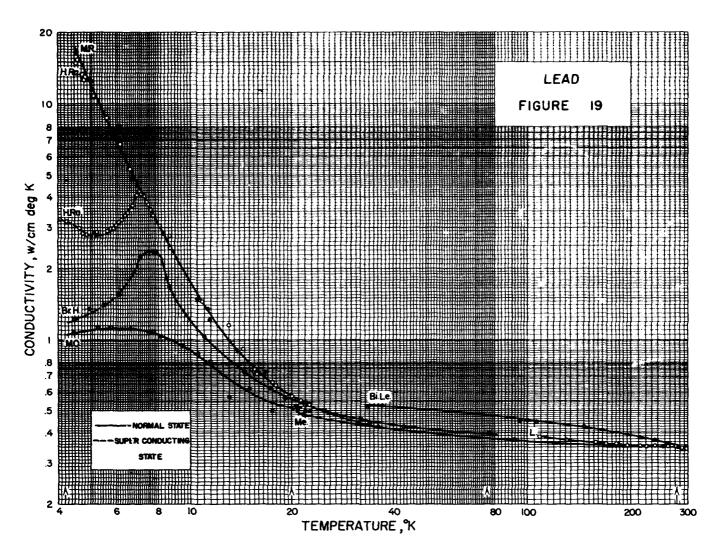


Curve	Sample source and analysis	Remarks	Reference
	"Pure"	k=0.64 at 0°C	L. Lorens (1881).
•••••	"Pure" .03% Pb.	k=0.61 at 18°C	W. Jaeger and H. Diesselhorst (1900).
L	Kahlbaum; "pure".	Lathe-turned from a rod	C. H. Lees (1908).
		Measured the relative change in conductivity at low tempera- tures, absolute values not given.	W. J. de Haas, S. Aoyama, and H. Brem- mer (1931) and W. J. de Haas and H. Brem- mer (1931a).
Hu. 1-4	Johnson, Mat- they; No. 1, 99.996% pure; #2, .03%, Hg added; #3, .3%, Hg ad- ded; #4,4.1%, Hg added.	Samples 1-3 were homogeneous solid solutions; #4 was two- phase; both normal and super- conducting state measured.	J. K. Hulm (1949, 1950).

TIN (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
		Measured relative change upon be- coming superconducting.	C. V. Heer and J. G. Daunt (1949).
Rs. 1, 2	Chempur; 99 992% pure.	Single crystals with tetragonal axis inclined 85° to rod axis.	A. Rademakers (1949).
• • • • • • • • •	Johnson, Mat- they;99.996% pure.	Measured conductivity in the in- termediate state.	D. P. Detwiler H. A. Fairban (1952a,b).
	Johnson, Mat- they; 99.997%.	Single crystal; measured the effect of a magnetic field; for sero field, k=25.1 at 4.4°K, 19.6 at 3.0°K, 18.0 at 2.4°K, in normal state.	K. Mendelssoh H. M. Rosen berg (1953).
M. Rn	do	Single crystal; upper curve in fig- ure 18b; superconducting state.	K. Mendelssoh C. A. Renton (1953).
M. Rn	do	Polycrystalline; lower curve on fig-	Do.
R	do	ure 18b; superconducting state. $\alpha = 60 \times 10^{-5}$, $\beta = 0.12$	H. M. Rosenber (1954a).
Gd. 1-5	No. 1 and 2, "spect. pure"; #3, 0.3% In; #4 and 5, 3%	Polycrystalline; crystal sizes about 1 to 3 mm; cast in glass.	B. B. Goodman (1953).



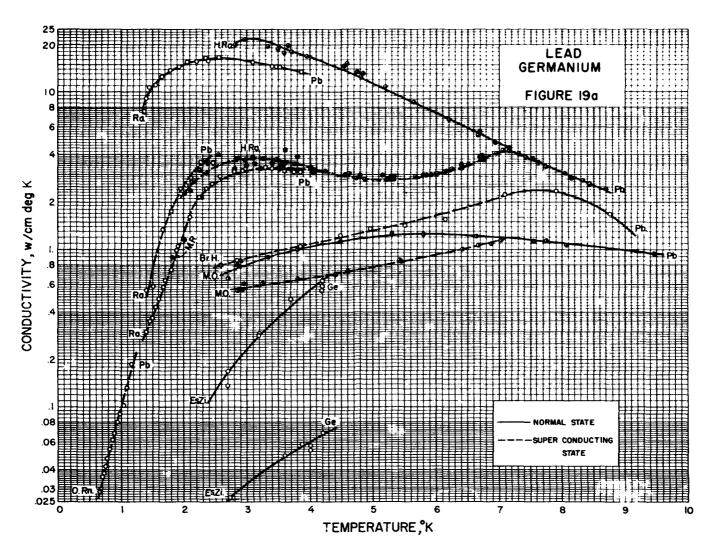


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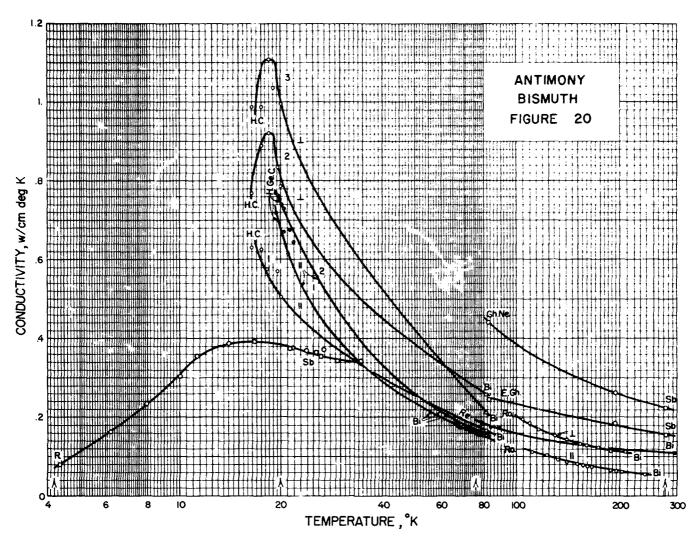
	and analysis		
	"Pure"	k=0.35 at 0°C	L. Lorens (1881).
	do	k=0.35 at 18°C	W. Jaeger and H. Diesselhorst (1900).
		k=0.35 at 25°C, 0.45 at 90°K	P. Macchia (1907).
L	Baxendale; "pure".	Lathe-turned from a bar	C. H. Lees (1908).
Me	Kahibaum; 99 998% pure.	Cold-drawn	W. Meissner (1915).
	Kahlbaum; "pure".	Results agree with Meissner (1915).	R. Schott (1916).
		k=0.38 at 12°C, 0.36 at 23°C	T. Pecsalski (1917).
Bi. Le			C. C. Bidwell and E. J. Lewis (1929).
		Measured relative temperature variation.	W. J. de Haas and H. Brem- mer (1931).
Br. H		Measured in both normal and su- perconducting states.	H. Bremmer and W. J. de Haas (1936).
••••	Hilger; 99.999% pure.	Measured changes in conductivity during superconducting transi- tion.	K. Mendelssohn and R. B. Pontius (1937)

LEAD (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
H. Ra	Hilger	Melted under vacuum; single crystal; in both normal and superconducting states.	W. J. de Hass and H. Rade makers (1940)
Ra	do	Two single crystals	A. Rademakers (1949).
м. о	0.02% Bi	In both normal and superconducting states.	K. Mendelmohi and J. L. Olner (1950c).
		Studied conductivity in intermediate state.	R. T. Webber and D. A. Spob (1951).
M. R	Tadenac, 99 998% pure.	Single crystal; measured in both normal and superconducting states; normal curve gives $\alpha = 325 \times 10^{-3}$, $\beta = 0.10$; same curve on graph as H. Ra.; experimental points marked by filled circles.	K. Mendelssoh and H. M. Rosenberg (1952b).
0. Rn		Single crystal	J. L. Oben and C. A. Rentor (1952).
	Tadenac	Measured effect of magnetic field	K. Mendelssohr and H. M. Rosenberg (1953).
	Same as M.R	$k=25$ at 3.1°K, 28 at 2.7°K; continuation of work of Mendelssohn, Rosenberg (1952b); $\alpha=290\times10^{-5}$, $\beta=0.10$.	H. M. Rosenberg (1954a).



(For GERMANIUM, see the table under figure 18, page 31.)

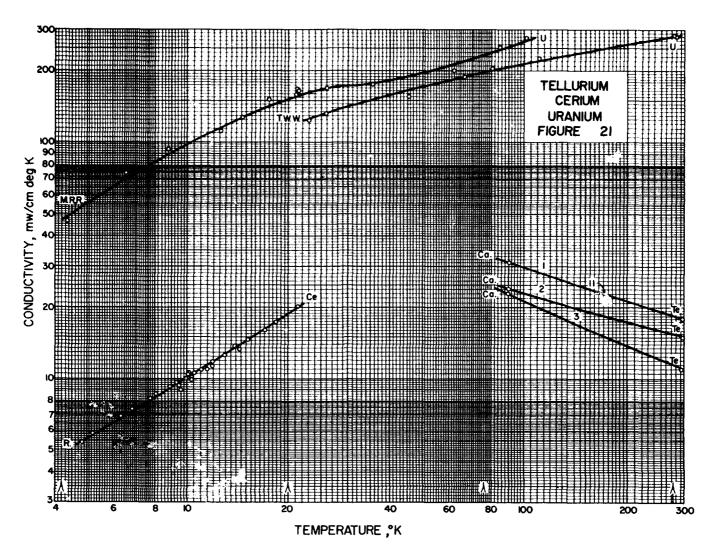


ANTIMONY

Curve	Sample source and analysis	Remarks	Reference
		k=0.19 at 0°C	L. Lorens (1881).
E. Gh	Kahlbaum	Cold-drawn	A. Eucken and G. Gehlhoff (1912).
Gh. Ne	Kahibaum; "pure".	R	G Gehlhoff and F. Neumeier (1913a).
•••••	Kahibaum; "chem. pure".	Measured effect of grain size; sample with largest grains had a conductivity close to the Gh. Ne. curve.	A. Eucken and O. Neumann (1924).
•••••	Kahlbaum; "pure".	Measured effect of magnetic field on k and R for single and poly- crystalline rods; at 80° and 90°K results agree with those of Gh. Ne. curve.	K. Rausch (1947).
R			H. M. Rosenberg (1954a).

BISMUTH

Curve	Sample source and analysis	Remarks	Reference
		k=0.074 at 0°C	L. Lorens (1881).
•••••	"Pure"	k=0.081 at 18°C; R, Cp, emf.	W. Jaeger and H. Diesselhors (1900).
		At 87°, 194°, and 291°K results are close to Rodine's upper curve.	E. Giebe (1903
	Kahlbaum; "pure".	Drawn; at 83°, 194°, and 273°K results are close to Rodine's upper curve.	G. Gehlhoff an F. Neumeie. (1913a).
• • • • • • • • •	do	Measured pressed powders; k = 0.21 at 83°K, 0.08 at 0°C.	G. Gehlhoff an F. Neumeier (1913b).
	0.02% Pb, trace Fe.	Single crystals, measured anistropy.	G. W. C. Kaye and J. K. Roberts (1923).
• • • • • • • • • • • • • • • • • • • •		Measured effect of grain size	A. Eucken and O. Neumann (1924).
		Measured effect of magnetic field on conductivity in single crystals.	G. W. C. Kayear W. F. Higgi (1929).



BISMUTH (Cont'd)

Curve	Sample source and analysis	Remarks	Reference
Re	Kahlbaum; "pure".	Rod axis inclined 80° to crystal axis; studied effect of magnetic field; R.	H. Reddemann (1934).
Ro		Measured two single crystals, one with rod axis parallel to trigonal crystal axis, one perpendicular.	M. T. Rodine (1934).
H. C. 1, 2, 3.	Hilger; 99.995% pure, trace of silver.	Single crystals; No. 1, rod axis par- allel to main crystal axis; #2, rod axis parallel to binary axis; #3, rod axis parallel to bissect- rix of binary axes.	W. J. de Haas and W. H. Capel (1934a, b).
		At 83° and 90°K results are close to the ones above; measured effect of magnetic field on k and R.	E. Grüneisen and J. Gielessen (1936).
H. Ge. C	Hilger; .002% milver, trace of Pb.	Single crystal; rod axis parallel to main crystal axis; measured ef- fect of magnetic field.	W. J. de Haas, A. N. Gerrit- sen, and W. H. Capel (1936).
•••••		Measured k between 2.3° and 77.4°K.	S. Shalyt (1947)
•••••		Measured effect of magnetic field	E. Grüneisen, K. Rausch, and K. Weis (1950).

TELLURIUM

Curve	Sample source and analysis	Remarks	Reference
Ca	99.999% pure	No. 1, a single crystal with rod axis parallel to main crystal axis; #2 and #3 are polycrystalline; also measured R, emf.	C. H. Cartwright (1933).
		CERIUM	
R		k=T/900 from 4° to 20°K	H. M. Rosenberg (1954a).
		URANIUM	
T. W. W		Quenched; also measured R, emf	W. W. Tyler, A. C. Wilson, and G. J. Wolga (1952).
M. R	Assoc. Elec. Ind Res. Lab.	α=750×10 ⁻⁶ , β=95	K. Mendelssohn and H. M. Rosenberg (1952b).
R	do	α=790×10 ⁻⁵ , β=93	H. M. Rosenberg (1954a).

2.3 Alloys and Commercial Metals

The values for conductivities of experimental and commercial alloys are given in figures 22 through 29a and in the following tables, arranged according to periodic group of the major component. In several instances a particularly large class of alloys has been separately presented, i. e., copper-nickel alloys. Many of the experimental results are for a limited temperature range, so more of the data are presented in tables than in sec' on 2.2 on metals. This section is also not as complete as section 2.2 because many of the data were published in now unavailable journals or institute reports. As for the preceding tables the following tables contain columns indicating the curve identifying marks, composition, conductivity, remarks, and reference. In addition, they occasionally contain information on trade designation or symbols and manufacturing tempers. The names or numbers and the arrangement within a group are based upon the corresponding arrangements in Metals Handbook.¹ The composition limits for many of the alloys are also taken from the Metals Handbook. The tables listed below, which quote "company or trade manuals", are all based on room-temperature values.

In pure metals the greater part of the energy transfer is by electrons, whereas in alloys the transfer by the lattice vibration is very significant and may be the predominant mode. For that reason the conductivity is not as sensitive to small differences in composition as it is in nearly pure metals. It will be noted in the following graphs that the conductivity curves of alloys with similar compositions are usually parallel to each other and seldom intersect.

1Metals Handbook, 1948 ed., Am. Soc. for Metals, Cleveland, Ohio.

ALKALI METAL ALLOYS

Nominal Composition (%)	Conductivity and remarks	Reference
Sodium-potassium; 50-50 by atomic percent.	w/cm deg K k=0.29 at -8.9°C, 0.30 at -10.6°C	J. W. Hornbeck (1913).

BERYLLIUM

Commercially pure; Beryllium Co. of America.	See figure 2, under "Metallic Elements"	E. J. Lewis (1929).
Copper-beryllium	See "Copper Alloys"	

MAGNESIUM ALLOYS

0.5 Mn	k=1.60 at 273°K, 1.34 at 87°K 1	J. Staebler (1929).
0.8 Mn	k=1.58 at 273°K, 1.22 at 87°K 1	Do.
2 Mn	k=1.18 at 273°K, 0.67 at 87°K 1	Do.
3.54 Mn	k=1.02 at 273°K, 0.57 at 87°K 1	Do.
6 Al	k=0.80 at 273°K, 0.59 at 87°K 1	Do.
8 Al	k=0.65 at 273°K, 0.42 at 87°K 1	Do.
12 Al	k=0.59 at 273°K, 0.33 at 87°K 1	Do.
0.7 Si	k=1.48 at 273°K, 1.10 at 87°K 1	Do.
1.5 Si	k=1.40 at 273°K, 0.95 at 87°K 1	Do.
8 Ce	k=1.25 at 273°K, 1.06 at 87°K 1	Do.
12 Ce	k=1.03 at 273°K, 0.81 at 87°K 1	Do.
8 Cu	k=1.25 at 273°K, 0.88 at 87°K 1	Do.
8 Zn	k=1.19 at 273°K, 0.89 at 87°K 1	Do.
8 Cd	k=1.42 at 273°K, 1.30 at 87°K 1	Do.
2 Si, 6 Al	k=0.69 at 273°K, 0.48 at 87°K 1	Do.
2 Si, 8 Al	k=0.61 at 273°K, 0.38 at 87°K 1	Do.
2 Si, 10 Al	k=0.55 at 273°K, 0.29 at 87°K 1	Do.
2 Si, 12 Al	k=0.54 at 273°K, 0.28 at 87°K 1	Do.
	l e	

MAGNESIUM ALLOYS (Cont'd)

Nominal Composition (%)	Conductivity and Remarks	Reference
	w/cm deg K	
15 Cu	k=1.54 at 273°K, 1.51 at 87°K; chill-cast	W. Mannchen (1931).
20 Cu, 3 Si	k=1.08 at 273°K, 0.89 at 87°K; chill-cast	Do.
2.2 Ag	k=1.31 at 25°C 2	R. Kikuchi (1932).
6.0 Ag	k=1.16 at 27°C 2	Do.
2.1 Al	k=0.88 at 27°C 2	Do.
4.2 Al	k=0.69 at 22°C 2	Do.
6.2 Al	k=0.56 at 22°C 2	Do.
8.2 Al	k=0.51 at 18°C 2	Do.
10.3 Al	k=0.45 at 19°C 2	Do.
12.2 Al	k=0.39 at 23°C 2	Do.
2.4 Cu	k=1.39 at 20°C° 3	Do.
8.3 Cu	k=1.31 at 24°C 2	Do.
1.9 Ni	k=1.36 at 20°C 2	Do.
5.8 Ni	k=1.26 at 24°C 2	Do.
2.2 Sn	k=1.06 at 21°C 2	Do.
6.4 Sn	k=0.74 at 22°C 2	Do.
2.1 Zn	k=1.26 at 26° 2	Do.
6.1 Zn	k=1.09 at 26°C 2	Do.
4 Zn, 0.5 Cu	"Elektron" k = 1.14 at 26° C°2	Do.
4 Al, 1 Zn, 1 Cd, 1 Sn.	k=0.56 at 22°C 2	Do.
8 Al, 3 Zn, 0.4 Cu	"Dow metal" k=0.61 at 29°C 2	Do.
Al, 0.5 Zn, 2 Cd, 1 Sn.	k=0.63 at 22°C 2	Do.
4 Al, 3 Cd, 1 Sn	k=0.69 at 22°C 2	Do.
4 Al, 2 Cd, 2 Sn	k=0.56 at 30°C° 2	Do.

² Vacuum-annealed.

¹ Chill-cast; also measured R.

MAGNESIUM ALLOYS (Cont'd) COMPANY AND TRADE MANUALS

ASTM designation	Trade designations	Nominal composition (%)	Conductivity
			w/cm deg K
A 8	Dowmetal A; Maslo AM 241; Brit- ish DTD 59A, DTD 289, Elek- tron A8, Elektron A8K.	8 Al, 0.2 Mn	0.75
A 10	Downetal G; Maslo AM 240; AM- C598; British DTD 259; Elek- tron VI.	10 Al, 0.1 Mn	.71
AM 80 A		See A 8	
AM 100 A		See A 10	
AZ 31X, A, B.	Downetal FS-1; Maslo AM-C52S; Whitelight FS-1; British DTD 120A; Elektron AZ 31.	3 Al, 1 Zn, 0.3 Mn.	.96
AZ 51 X	Dowmetal J8-1	5 Al, 1 Zn, 0.25 Mn.	.88
AZ 61 X, A, B.	Dowmetal J-1; Maslo AM-C578; Whitelight J-1; British DTD 88B, DTD 120A, DTD 259; Elektron AZM.	6 Al, 1 Zn, 0.2 Mn.	.80
AZ 63, A	Dowmetal H; Maslo AM 265; Brit- ish DTD 59A, DTD 289; Elek- tron AZG.	6 Al, 3 Zn, 0.2 Mn.	.75
AZ 80 X, A	Dowmetal 0-1; Maslo AM-C588; Whitelight 0-1; British DTD 88B; Elektron AZ 855.	8.5 Al, 0.5 Zn, .15 Mn.	.75
AZ 91 A, B, C.	Downetal R, RC;Maslo AM 263; British DTD 136A; Elektron AZ 91.	9 Al, 0.7 Zn, 0.2 Mn.	.71
AZ 92, A	Dowmetal C; Maslo AM 260	9 Al 2 Zn, 0.1 Mn.	.71
M 1 A, B	Downetal M; Maslo AM403, AM 38; Whitelight M; British DTD 142, 118, 140A; Elektron AM 503.	1.5 Mn	1.26
	Maslo AM 244	4 Al, 0.2 Mn	0.96
	Downetal EK 30A	3 rare earths, 0.35 Zr, 0.3 Zn.	.27

ALUMINUM ALLOYS

Composition (%)	Conductivity and remarks	Reference
	w/cm deg K	
0.5 Fe, 0.4 Cu	k=2.01 at 18° C	W. Jaeger and H. Diesselhorst (1900).
Commercial	k=1.93 at 0°C, 1.90 at 86°K, 1.59 at 21.4°K.	R. Schott (1916).
Composition (%) 1	Conductivity 1	State 1
	w/cm deg K	
12.2 Cu, 0.3 Si, 0.6 Fe 2	1.24 1.48	Cast. Annealed.
12.2 Cu, 0.2 Si, 0.6 Fe, 1 Mn ² .	0.93 1.33	Cast. Annealed.
10.5 Cu, 0.3 Si, 0.8 Fe, 1 Ni, 3 Sn ² .	1.35 1.59	Cast. Annealed.
8.4 Cu, 0.3 Si, 0.7 Fe, 0.7 Mn ²	1.02 1.35	Cast. Annealed.
8.1 Cu, 0.4 Si, 0.6 Fe 2	1.39 1.67	Cast. Annealed.
6.9 Ca, 0.3 Si, 0.7 Fe, 1.2 Sn ²	1.47. 1.66.	

ALUMINUM ALLOYS (Cont'd)

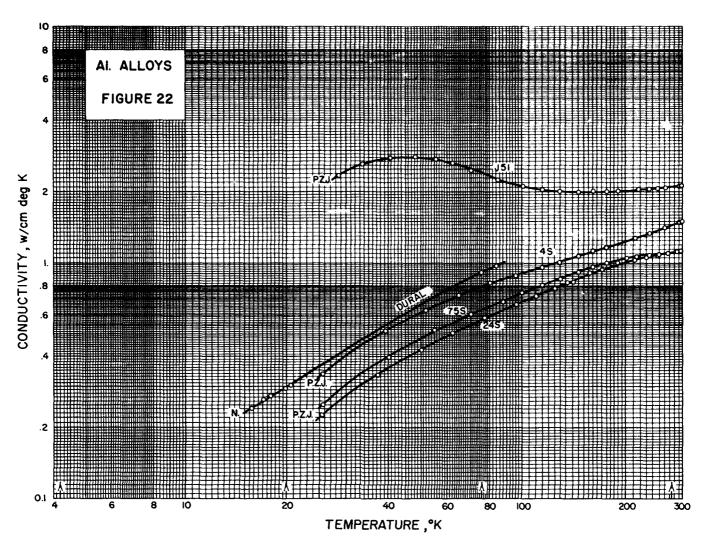
Composition (%) 1	Conductivity 1	State 1
	w/cm deg K	
.3 Cu, 0.5 Si, 0.8 Fe, 0.5 Mn, 1.2 Mg. ³	1.18 1.52	Cast. Annealed.
.3 Cu, 0.4 Si, 0.9 Fe, 0.6 Mn, 0.4 Mg. ²	1.22 1.52	Cast. Annealed.
.5 Cu, 0.4 Si, 0.9 Fe, 1.8 Ni, 0.9 Mg. ²	1.44 1.63	Cast. Annealed.
4 Cu, 0.5 Si, 0.7 Fe, 2.1 Ni, 0.9 Mg. ²	1.30	Cast. Annealed.
.8 Cu, 6.1 Si, 0.9 Fe, 0.6 Mn, 1.6 Mg. ²	1.00	Cast. Aunealed.
.7 Cu, 0.4 Si, 0.6 Fe, 12.0 Zn *.	1.32	Cast. Annealed.
.6 Cu, 0.4 Si, 0.6 Fe, 20.3 Zn 2	1.07.	Cast. Annealed.
.5 Cu, 0.3 Si, 0.8 Fe, 0.5 Mn, 2.6 Zn.2	1.26	Cast. Annealed.
.9 Cu, 0.1 Si, 1 Fe, 1.5 Mg 2.	1.57 1.65	Cast. Annealed.
.8 Cu, 0.4 Si, 0.9 Fe, 0.9 Cr *.	1.05	Cast. Annealed.
.8 Cu, 0.3 Si, 0.6 Fe, 1 Ni, 1.6 Mg. ²	1.48. 1.65.	Cast. Annealed.
1.9 Si, 0.8 Fe ²	1.31	Cast. Annealed.
.1 Si, 0.6 Fe ²	1.86. 2.00	Cast. Annealed.
.1 Cu, 0.4 Si, 0.6 Fe *	1.33 1.32	Quenched. Aged.
.3 Cu, 0.5 Si, 0.8 Fe, 0.5 Mn, 1.2 Mg. ³	1.23 1.23	Quenched. Aged.
.5 Cu, 0.4 Si, 0.9 Fe, 1.8 Ni, 0.9 Mg. ³	1.38	Quenched. Aged.
.8 Cu, 6.1 Si, 0.9 Fe, 0.6 Mn, 1.6 Mg. ³	1.16 1.14	Quenched. Aged.
.6 Cu, 0.4 Si, 0.6 Fe, 20.3 Zn *.	0.98 0.98	Quenched. Aged.
5 Ca, 0.3 Si, 0.9 Fe, 0.5 Mn, 2.6 Zn. ³	1.32	Quenched.
.9 Cu, 0.1 Si, 1 Fe, 1.5 Mg	1.59	Quenched.
.8 Cu, 0.3 Si, 0.6 Fe, 1.0 Ni, 1.6 Mg. ³	1.45	Quenched.
.3 Cu, 0.5 Si, 0.8 Fe, 0.5 Mn, 1.2 Mg.4	1.36	Drawn. Annealed.
.3 Cu, 0.4 Si, 0.9 Fe, 0.4 Mn, 0.6 Mg.4	1.48. 1.73.	Drawn. Annealed.
1.9 Si, 0.8 Fe 4	1.73 1.81	Drawn. Annealed.
.1 Si, 0.5 Fe 4	2.06 2.07	Drawn.

¹ Results by H. Masumoto (1925) at 27°C.

² The samples were chill cast in an iron mold, then annealed for 30 minutes at 450°C.

³ Chill-cast in an iron mold, annealed, then heated for 30 minutes at about 500°C, quenched in water, and later aged two weeks.

⁴ Chill-cast in an iron mold, fo.ged, then cold-drawn to 60% of original diameter, and later annealed for 30 minutes at 500°C.



ALUMINUM ALLOYS (Cont'd)

Nominal composition (%)	Conductivity and remarks	Reference
	w/cm deg K	
8 Cu	k=1.32 at 273°K; 0.88 at 87°K	W. Mannchen (1931).
D ₀	k=1.31 at 273°K; 0.90 at 87°K	Do.
15 Cu	k=1.48 at 273°K; .90 at 87°K	Do.
8 Mg	k=1.00 at 273°K; .73 at 87°K	Do.
D ₀	k=1.05 at 273°K; 77 at 87°K; thermally treated.	Do.
12 Mg	k=0.77 at 273°K; .56 at 87°K	Do.
14 Mg	k=0.69 at 273°K; .44 at 87°K; thermally treated.	Do.
20 Si	k=1.59 at 273°; 1.21 at 87°K; "Alusil".	Do.
4 Cu, 2 Ni, 1.5 Mg; "Y" Alloy".	k=1.62 at 273°K; 1.12 at 87°K	Do.
Do	k=1.53 at 273°K; 1.38 at 87°K; thermally treated.	Do.
Mg, Mn, Sb; K-S Alloy 245	k=1.07 at 273°K; 1.00 at 87°K	Do.
Mn, Mg, 8b; K-8 Alloy 280	k=1.00 at 273°K; 0.80 at 87°K	Do.
Mn, Mg, Sb; K-S Alloy Special.	k=1.39 at 273°K; 1.14 at 87°K	Do.
Cu, Mn, Mg; Nelson-Kolben 10.	k=1.60 at 273°K; 1.32 at 87°K	Do.
Cu; Neison-Kolben Vn 1	k=1.43 at 273°K; 1.18 at 87°K	Do.
Cu; Nelson-Kolben	k=1.59 at 273°K; 1.30 at 87°K	Do.
3-5 Cu, 0.5 Mg	k=1.50 at 273°K; 0.89 at 87°K	Do.

ALUMINUM ALLOYS (Cont'd)

Curve	Composition (%)	Remarks	Reference
N"Dural"	0.57 Mg, 0.42 Fe, 4.10 Cu, 94.0 Al.	As stamped; "Dur- aluminium".	J. de Nobel (1951).
P. Z. JJ51	0.29 Cu, 0.56 Mg, 0.02 Ma, 0.56 Fe, 0.30 Si, 0.01 Cr, 0.01 Ti.		R. W. Powers, J. B. Ziegler and H. L. Johnston (1951).
P. Z. J48	0.16 Cu, 1.02 Mg, 1.20 Mn, 0.52 Fe, 0.13 Si, 0.02 Cr, 0.02 Ti.		Do.
P. Z. J758	1.5 Cu, 5.5 Zn, 2.5 Mg, 0.2 Mn, 0.3 Cr.		Do.
P. Z. J248	4.49 Cu, 0.01 Zn, 1.47 Mg, 0.66 Mn, 0.34 Fe, 0.13 Si, 0.01 Cr, 0.02 Ti.		Do.

ALUMINUM ALLOYS (Cont'd)

ASTM designa- tion	Trade designation	Nominal compositions (%)	Conductivity	State
À2	EC. 2S; British BS 2L	99.45 Al99 Al	w/cm deg K 2.34 2.22 2.18	Annealed
	WR	OUGHT ALLOYS	 	
MI	38	1.2 Mn	1.93 1.63	O H 12
	4S	19W- 1W-	1.59 1.55	H 14 H 18
· ,		1.2 Mn, 1 Mg	1.63 1.63	O H 38
CP 21 CS 41	11S; British BS 6L1 14S; British DTD 364.	5.5 Cu, 0.5 Pb, 0.5 Bi. 4.4 Cu, 0.8 Si, 0.8 Mn, 0.4 Mg.	1.55 1.93 1.55	T 3 O T 6
CM 21	17S; British BS 6L1	4 Cu, 0.5 Mg, 0.5 Mn.	1.21 1.72 1.21	T4 O T 4
	A 17S. 18S; British BS 4L25, BS 2L42.	2.5 Cu, 0.3 Mg 4 Cu, 2 Ni, 0.5 Mg	1. 55 1.93	Ť 4 O T 61
	B18S	4 Cu, 1.5 Mg, 2.0 Ni	1.55 1.93	0
○G 21	24S; British BS2L40, DTD 273.	4.5 Cu, 1.5 Mg, 0.6 Mn.	1.72 1.88 1.21	T 72 O T 4
	258	4.5 Cu, 0.8 Mn, 0.8 Si.	1.55 1.93	T 6
	328	12.5 Si, 1.0 Mg, 0.9 Cu, 0.9 Ni.	1. 55 1.38	O T 6
	50S	1.2 Mg	1.93 1.93	O H 38
	C50S	1.3 Mg 1.0 Si, 0.6 Mg, 0.25 Cr.	1.55 2.09 1.72	0 T 4
3R 1	52S	2.5 Mg, 0.25 Cr	1.38 1.38	O H 38
	538	1.3 Mg, 0.7 Si, 0.25 Cr.	1.72 1.55	O T 4
`		5.2 Mg, 0.1 Mn, 0.1 Cr.	1.17 1.09	O H 18
S 21		1 Mg, 0.6 Si, 0.25 Cu, 0.25 Cr.	1.72 1.55	O T 4
	628	0.25 Cu, 0.6 Si, 1 Mg.	1.72 1.55	O T 4
	638	0.4 Cu, 0.7 Mg	1.93 2.09	T 42 T 5
G 42	758	5.5 Zn, 2.5 Mg, 1.5 Cu, 0.3 Cr, 0.2 Mn.	1.21	Τ̈́δ
	R. 301	1 Mg, 0.7 Si, 0.5 Mn.	1.93 1.21	O T 4
	R 317	4 Cu, 0.5 Mn, 0.5 Mg, Pb, 0.5 Bi.	1.55 1.72 1.21	T 6 O T 4
· · · · ·	CA	STING ALLOYS	<u></u> '	
5	13	12 Si	1.55 to 1.21	
	43	5 Si	1.47 1.67	Annealed.
C 2	85 108	5 Si, 4 Cu	1.17 1.21 1.47	Cast. Annealed.
C 8	Allcast	5 Si, 3 Cu	1.05 1.17 1.13 1.38	Cast. Relieved. Aged. T 7
C 1	A108112	5.5 Si, 4.5 Cu 7 Cu, 1.7 Zn	1.42 1.17	
S 22	113	7 Cu, 2 Si, 1.7 Zn	1.47 1.17	Annealed.
- 1	C113122	7 Cu, 3.5 Si	1.47 1.09 1.59 1.30	Annealed. T 2 T 61
		i		

ALUMINUM ALLOYS (Cont'd) COMPANY AND TRADE MANUALS COMPANY AND TRADE MANUALS COMPANY AND TRADE MANUALS

ASTM designa- tion	Trade designation	Nominal compositions (%)	Conductivity	State
			w/cm deg K	
SC 41	A 132	12 Si, 2.5 Ni, 1.2 Mg, 0.8 Cu.	1.17	T 551
• • • • • • • • •		9 Si, 3.5 Cu, 0.8 Mg, 0.8 Ni.	1.09	T 5
CN 21	138142	10 Cu, 4 Si, 0.3 Mg. 4 Cu, 2 Ni, 1.5 Mg.	1.05 1.67 1.34 1.51	T 21 T 571 T 77
C 1	195	4.5 Cu	1.30 1.38 1.47	T 61 T 4 T 62
CS 4	В 195	4.5 Cu, 2.5 Si	1.38 1.42 to 1.88	T 4
G 1	212 214; British DTD 165.	8 Cu, 1.2 Si	1.17 1.38	
• • • • • • • • • • • • • • • • • • • •	A 214. B 214. F 214. 218.	8 Mg		Annealed
3C 8	319. 333.	10 Mg	.88 1.13 1.05 1.21 1.17	T 4 F T 5 T 6
SC 21	355	5 Si, 1.3 Cu, 0.5 Mg	1.42 1.67 1.42 1.47 1.63	T7 T51 T6 T61
3G 1	356	7 Si, 0.3 Mg	1.51 1.67 1.55 1.59 1.63	Chill T 6 T 51 T 6 T 7 Chill T6.
	360, A360 380, A380 384 A612	9.5 Si 9 Si, 3.5 Cu 12 Si, 3.8 Cu 6.5 Zn, 0.7 Mg, 0.5	1.13 to 1.47 0.96 to 1.09 0.96 .96	
	C 612	Cu.	1.59	
	750	Mg. 6.5 Sn, 1 Cu, 1.0 Ni	1.80	

TITANIUM ALLOYS

Curve	Composition (%)	Conductivity and Remarks	Reference
	2.8 Cr, 1 Fe	w/cm deg K Abstract only; k=0.13 at 273°K, 0.10 at 195°K, 0.06 at 80°K.	C. J. Rigney and L. I. Bockstahler (1951).
Fig. 29; T.W Ti.	Rem-Cru Titanium, RC 130-B; 4.7 Mn, 3.99 Al, 0.14 C.	R, emf	W. W. Tyler and A. C. Wilson (1952).

TUNGSTEN

Composition	Conductivity and remarks	steference
"Impure"	w/cm deg K Single crystal; k≈1.83 at 83°K, 1.80 at 21°K	E. Grüneisen and E. Goens (1927).

CHROMIUM COMPANY AND TRADE MANUALS

Composition	Conductivity		
Commercial	k=0.67 at 20°C.	w/cm deg K	

IRON

See figures 8 and 9 under "METALLIC ELEMENTS"

STEELS

The tables for steels are arranged into groups where the principal alloying metals are as follows: carbon; silicon; copper, chromium, cobalt, manganese, molybdenum, nickel, tungsten, vanadium; and aluminum.

CARBON STEELS

Composition (%)	Conductivity and remarks	Reference
	w/cm deg K	
0.1 C	k=0.67 at 18°C; wrought iron	W. Jaeger and H. Diesselhors (1900).
1 C	k=0.45 at 18°C, wrought iron	Do.
0.1 C, 0.06 Mn, 0.05 Cu, 0.02 Si, S, 0.03 P.	k=0.72 at 18°C	E. Grüneisen (1900).
0.57 C, 0.2 Si, 0.1 Mn, 0.04 S, 0.03 Cu, 0.01 P.	k=0.52 at 18°C	Do.
0.99 C, 0.1 Mn, 0.06 Si, 0.03 S, Cu.	k=0.51 at 18°C	Do.
1.5 C, 0.2 Mn, 0.05 Si, 0.03 Cu, S, 0.01 P.	k=0.50 at 18°C	Da.
1 C; "silver steel"	See figure 23, curve with initial L	C. H. Lees (1908).

CARBON STEELS (Cont'd)

Composition (%)1	Conductivity 1	State 1
	w/cm deg K	
0.1 C, 0.4 Mn, 0.02 P, 0.02 S	.45 .42 .42 .40	As cast. Annealed 1,000°C, 2 hr. 6 hr. 8 hr.

CARBON STEELS (Cont'd)

Composition (%) 1	Conductivity 1	State 1
	w/cm deg K	
.53 C, 0.05 Si, 0.02 Mn, 0.01 P, 0.03 S	.31	As cast.
.67 C, 0.11 Si, 0.02 Mn, 0.03 P, 0.05 S	30	Do.
• • • •	.32	1,000°C annealed, 2 h
	.32	6 hr.
	.32	8 hr.
3.12 C, 0.06 Si, 0.05 Mn, 0.02 P, 0.06 S	26	As cast.
.14 C, 0.01 Si, 0.03 Mn, 0.02 P, 0.03 S	26	Do.
1.17 C, 0.21 Si, 0.08 Mn, 0.04 P, 0.06 S		Do.
	.26	Annealed 1,000°C, 2 h
	.26	6 hr.
	.27	8 hr.
.53 C, 0.04 Si, 0.05 Mn, 0.01 P, 0.05 S	23	As cast.
.64 C, 0.16 Si, 0.04 Mn, 0.02 P, 0.02 S		Do.
	.23	Annealed 1,000°C, 2 h
	.23	6 hr.
02 C 0.15 C: 0.04 M. 0.00 D 0.05 O	.23	8 hr.
1.93 C, 0.15 Si, 0.04 Mn, 0.02 P, 0.05 S		As cast.
3.96 C, 0.2 Si, 0.06 Mn, 0.01 P, 0.02 S	. 19	As cast.
	.21	Annealed 1,000°C, 2 h
	.26 .50	6 hr. 8 hr.
.13 C, 0.10 Si, 0.03 Mn, 0.02 P, 0.02 S	18	
6, 0.10 St, 0.03 MtH, 0.02 P, 0.02 S	26	As cast.
.26 C, 0.10 Si, 0.03 Mn, 0.02 P, 0.02 S	17	Annealed 1,000°C, 2 h
C, 0.10 DI, 0.00 MIII, 0.02 I , 0.02 D	.19	Annealed 1,000°C, 2 h
.35 C, 0.35 Si, 0.08 Mn, 0.02 P, 0.02 S		As cast.
O, 0.00 DI, 0.00 MII, 0.02 I, 0.02 D	.57	Annealed 1,000°C, 2 h
.40 C, 0.34 Si, 0.03 Mn, 0.02 P, 0.08 S		As cast.
0, 0.01 21, 0.00 1112, 0.02 1, 0.00 2	.17	Annealed 1,000°C, 2 h
.61 C, 0.37 Sì, 0.03 Mn, 0.02 P, 0.04 S	.]	As cast.
0, 0.0, 0., 0.00 1.12, 0.02 1, 0.01 0;;;	.15	Annealed 1,000°C, 2 h
.63 C, 0.54 Si, 0.08 Mn, 0.02 P, 0.07 S		As cast.
	.56	Annealed 1,000°C, 2 h
.82 C, 1.24 Si, 0.09 Mn, 0.01 P, 0.06 S	.13	As cast.
, , ,	.20	Annealed 800°C, 1 hr.
.81 C, 1.96 Si, 0.05 Mn, 0.05 S		As cast.
	.35	Annealed 800°C, 1 hr.
	.40	Add. annealed 1,000°
.84 C, 1.98 Si, 0.06 Mn, 0.01 S	43	As cast.
01 2.00 000 0.00 4.11, 0.01 0	.52	Annealed 800°C, 1 hr.

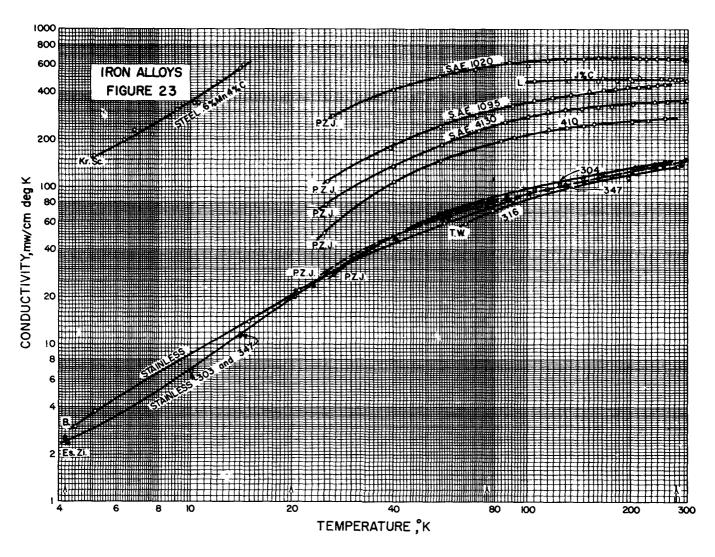
¹ Results by H. Masumoto (1927) at 25°C.

CARBON STEELS (Cont'd)

Curve	Composition (%)	Remarks	Reference
Fig. 24; Mild.	0.14 C, 0.08 Si, 0.07 Mn	"Mild steel"; heated to 800°C and fur- nace-cooled.	J. de Nobel (1951).
Fig. 23; P. Z. J SAE 1020.	0.33 Mn, 0.18 C, 0.014 Si	•••••••	R. W. Powers, J. B. Ziegler, H. L. Johnston (1951a).
Fig. 23; P. Z. J SAE 1095.	0.93 C, 0.34 Mn, 0.26 Si, 0.1 Ni, Cr, 0.05 Mo.	******************	Do.

CARBON STEELS (Cont'd) COMPANY AND TRADE MANUALS

Composition (%)	Conductivity
	w/cm deg K
0.08 C, 0.045 Cr, 0.07 Ni, 0.31 Mn, 0.02 Mo 0.23 C, trace Cr, 0.074 Ni, 0.635 Mn, 0.13 Cu 0.415 C, trace Cr, 0.063 Ni, 0.643 Mn, 0.12 Cu 0.80 C, 0.11 Cr, 0.13 Ni, 0.32 Mn, 0.07 Cu, 0.01 Mo 1.22 C, 0.11 Cr, 0.13 Ni, 0.35 Mn, 0.01 Mo, 0.08 Cu	.52 .49



Specific references can be found under the type of steel.

SILICON STEELS

Composition (%)	Conductivity and remarks	Reference
0.2 Si, 0.1 C, 0.1 Mn, trace of P, S, and Cu.	w/cm deg K k=0.60 at 18°C	W. Jaeger and H. Diesselhorst (1900).

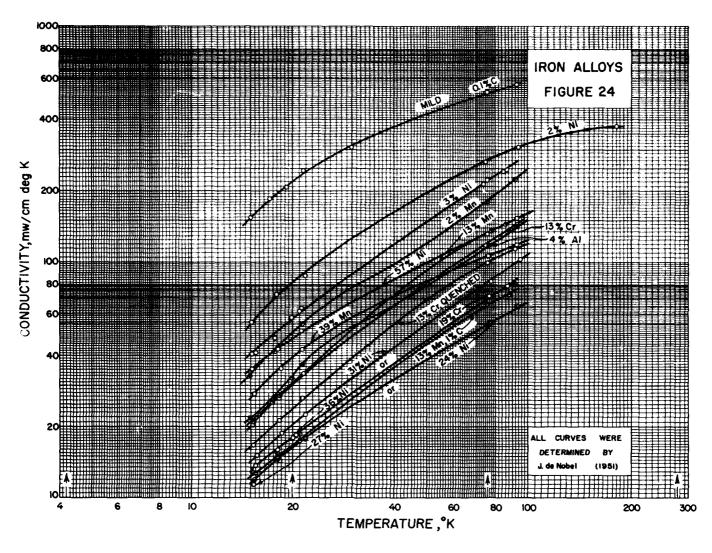
CORROSION RESISTING STEELS

(Copper, chromium, cobalt, manganese, molybed-num, nickel, tungsten, and vanadium)

Curve	Composition (%)	Remarks	Reference
Fig. 23; Kr. Sc.	0.6 Mn, 0.4 C, 0.3 Si, 0.03 S, 0.3 P.		J. Karweil and K. Schäfer (1939).
	"Stainless"	k=7 mw/cm deg K at 10°K, 11 at 15° K, 15 at 20°K.	K. R. Wilkinson and J. Wilks (1949).

CORROSION RESISTING STEELS (Cont'd)

Curve	Composition (%)	Remarks	Reference
. 24; 2% li.	1.92 Ni, 0.72 Mn, 0.21 Si, 0.14 C.	Heated to 800°C and furnace-cooled.	J. de Nobel (1951).
% Ni	. 24.30 Ni, 6.05 Mn, 1.18 C	Heated to 1,050°C and water-quenched.	Do.
6 Ni	27.30 Ni, 14.6 Cr, 3.5 W, 1.62 Si, 1.34 Mn, 0.44 C.	Heated to 1,000°C and water-quenched, "era/ATV".	Do.
6 Ni	31.4 Ni, 0.82 Mn, 0.7 C	Heated to 800°C and furnace-cooled.	Do. Do.
, Ni	36.17 Ni, 0.92 Mn, 0.16 C, 0.09 Si.	Heated to 1,050°C and water-quenched.	Do.
, Ni	57.5 Ni, 1.31 Mn, 0.34 C, 0.14 Si.	As forged; "A.M.F.".	Do.
Mn	2.23 Mn, 0.41 C, 0.07 Si	Heated to 800°C and furnace-cooled.	Do.
Mn, 1%	12.69 Mn, 1.27 C, 0.12 Si	Heated to 2,000°C and water-quenched, "manganese steel".	Do. Do.
, Mn	12.95 Mn, 0.10 S, 0.12 Si, 0.09 C, 0.05 P.	Heated to 1,000°C and water-quenched.	Do.
, Mn	38.9 Mp, 0.7 Si, 0.2 C, 0.06 S. 0.04 P.	do	Do.



Specific references can be found under the type of steel.

CORROSION RESISTING STEELS (Cont'd)

Curve	Composition (%)	Remarks	Reference
13% Cr	13.5 Cr, 0.36 C, 0.22 Si, 0.13 Mn.	Heated to 800°C and furnace-cooled.	J. de Nobel (1951).
13% Cr, quenched.	do	Heated to 950°C, oil quenched, reheated to 450°C, air- cooled.	Do. Do.
19% Cr	18.8 Cr. 8.1 Ni, 0.43 Si, 0.24 Mn, 9.12 C.	Heated to 1,150°C and water-quenched.	Do.
3% Ni	2.61 Ni, 0.75 Mo, 0.49 Cr, 0.45 Mn, 0.27 C, 0.11 Si, 0.03 P, 0.01 S.	Heated to 850°C, oil- quenched, reheated to 650°C, water- quenched.	Do.
Fig. 23; P.Z.JSAE 4130.	0.99 Cr, 0.52 Mn, 0.33 C, 0.2 Si, Ni, and Mo each.		R. W. Powers, J. B. Ziegler and H. L. Johnston (1951a).
P. Z. J410	12.6 Cr, 0.36 Si, 0.32 Mn, 0.12 Ni, 0.09 C, 0.06 Cu, 0.03 N, 0.01 P.		Do.

CORROSION RESISTING STEELS (Cont'd)

Curve	Composition (%)	Remarks	Reference
P.Z.J347	17.88 Cr, 10.28 Ni, 1.24 Mn, 0.85 Nb, 0.57 Si, 0.26 Cu, 0.06 C, 0.03 N, 0.02 P.		R. W. Powers, J. B. Ziegler, and H. L. Johnston (1951a).
P. Z. J304	18.68 Cr, 8.84 Ni, 1.12 Mn, 0.43 Si, 0.06 Cu, 0.05 C, 0.03 N, 0.02 P.		Da.
Fig. 23; B Stainless.	7.9 Ni, 18.9 Cr, 1 Ti, 0.7 Si, 0.1 C.	Austenite grains about 0.01 mm across.	R. Berman (1951b).
Fig. 23; Es. Zi303.	18 Cr, 9 Ni, 0.15 C		I. Estermann and J. E. Zimmer- man (1952).
Es. Zi347	18 Cr, 10 Ni, 0.5 Nb, 0.08 C.		Do.
T.W316	17 Cr, 12 Ni, 2.5 Mo, 0.1 C.	25% cold reduction	W. W. Tyler and A. C. Wilson (1952).

CORROSION RESISTING STEELS (Cont'd) COMPANY AND TRADE MANUALS

AISI No.	Nominal composition (%)	Conductivity
		w/cm deg K
	0.08 C, .045 Cr, .07 Ni, .31 Mn, .02 Mo	0.59
	0.23 C, trace Cr074 Ni, .635 Mn, .13 Cu	.52
		.52
		.37
		.33
	0.315 C. 1.09 Cr. 0.073 Ni. 69 Mn012 Mo07 Cu	.48
	0.35 C, .88 Cr, .26 Ni, .59 Mn, .2 Mo, .12 Cu	.43
	5 Cr. 0.5 Mo	.37
		.13
		.13
		.16
		.27
		.26
	1	.25
302	0.14 C, 18 Cr. 9 Ni. 2 Mn	.22
303	0.15 C, 18 Cr, 9 Ni, 0.07 P, S, Se each, 6 Zr, Mo each.	.22
309	0.20 C, 23 Cr, 13 Ni, 2 Mn	.19
410	0.15 C, 12.5 Cr	.40
416	0.15 C, 13 Cr, 0.07 P, S, Se each, .6 Zr, Mo each	.40
420	0.15 C or more, 13 Cr	.33
430	0.12 C, 16 Cr	.30
440	0.7 C, 17 Cr. 0.75 Mo.	.25
3.10	15 Cr. 35 Ni	.13

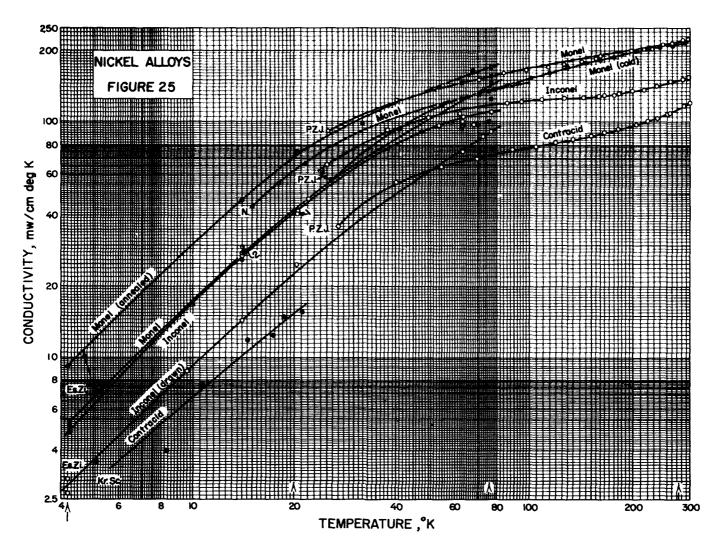
DEOXIDIZED STEELS

(Aluminum)

Curve	Composition (%)	Remarks	Reference
Fig. 24; 4% Al.	4.11 Al, 0.13 Si, 0.08 Mn,	Heated to 800°C and	J. de Nobel
	0.03 C, 0.01 S.	furnace-cooled.	(1951).

NICKEL ALLOYS COMPANY AND TRADE MANUALS

Trade Designation	Nominal composition (%)	Conductivity
		w/cm deg K
A Nickel	99.4 Ni+Co, 0.2 Mn, .15 Fe, .1 Cu, .1 C.	0.61
D Nickel		.48
Monel		.26
K Monel		.19
Hastellov A	57 Ni. 20 Mo. 20 Fe	.17
Hastelloy B	62 Ni, 30 Mo, 5 Fe	.11
Hastellov C		.13
Hastelloy D		.21
Inconel		.15
Illium G		.12
		.56
		.14
		.13
Constantan	45 Ni. 55 Cu	.23

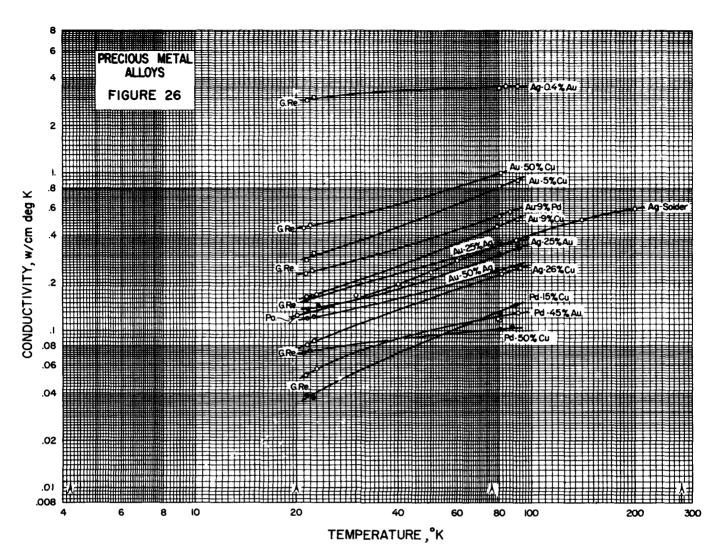


NICKEL ALLOYS (Cont'd)

Curve	Composition (%)	Conductivity and remarks	Reference
		w/cm deg K	
	97,0 Ni, 1.4 Co, 1 Mn, 0.4 Fe.	k=0.59 at 18°C	W. Jaeger and H. Diesselhorst (1900).
•••••	80 Ni, 20 Cr; "nichrome"	k=0.31 above room temperature.	R. Kikuchi (1932).
	70 Ni, 18 Cr, 12 Fe	k=0.28 above room temperature.	Do.
Fig. 25; Kr. ScContr- acid.	60 Ni, 15 Cr, 16 Fe, 7 Mo		J. Karweil and K. Schäfer (1939).
P.Z.JInco- nei.	80 Ni, 14 Cr, 6 Fe		R. W. Powers, J. B. Ziegler, and H. L. Johnston (1951c).
P.Z.JContracid.	60.05 Ni, 14.74 Cr, 15.82 Fe, 7.2 Mo, 2.14 Mn, 0.05 C.		Do.
P. Z. J. Monel.	67 Ni, 30 Cu, 1.4 Fe, 1.0 Mn, 0.15 C, .1 Si, .01 S.	Hot-rolled	Do.
P.Z.JMonel, cold.	do	Cold-rolled	Do.

NICKEL ALLOYS (Cont'd)

Curve	Composition (%)	Conductivity and remarks	Reference
		w/cm deg K	
	Commercial; 99.4 Ni	See Fig. 8 and Nickel Table under "Me- tallic Elements".	J. de Nobel (1951).
NMonel	67 Ni, 30.2 Cu	As forged	Do.
Fig. 24; 57% Ni.	57.5 Ni, 1.31 Mn, 0.34 C, .14 Si; remainder Fe, ap- prox. 40.	As forged	Do.
Fig. 25; Es. ZiInconel (drawn).	Inconel	Hard-drawn	I. Estermann and J. E. Zimmer- man (1952).
Es. Zi Inco- nel, #1.	do	Annealed	Do.
Es. ZiInco- nel, #2.	do	Hot-rolled	Do.
Es.ZiMonel.	Monel	Hard-drawn	Do.
Es. ZiMonel, (annealed).	Monel	Annealed	Do.



PRECIOUS METAL ALLOYS See also the tables given under "SILVER ALLOYS" and "GOLD ALLOYS".

PALLADIUM ALLOYS

Composition (%)	Conductivity and remarks	Reference
	w/cm deg K	
90 Pd, 10 Ag	k=0.48 at 25°C	F. A. Schulze (1911).
80 Pd, 20 Ag	k=0.37 at 25°C	Do.
70 Pd, 30 Ag	k=0.32 at 25°C	Do.
60 Pd, 40 Ag	k=0.27 at 25°C	Do.
50 Pd, 50 Ag	k=0.32 at 25°C	Do.
90 Pd, 10 Au	k=0.52 at 25°C	Do.
90 Pd, 20 Au	k=0.42 at 25°C	Do.
70 Pd, 30 Au	k=0.40 at 25°C	Do.
60 Pd, 40 Au	k=0.36 at 25°C	Do.
50 Pd, 50 Au	k=0.36 at 25°C	Do.
90 Pd, 10 Pt	k=0.56 at 25°C	Do.
80 Pd, 20 Pt	k=0.44 at 25°C	D ₀ .
70 Pd, 30 Pt	k=0.40 at 25°C	Do.
80 Pd, 40 Pt	k=0.38 at 25°C	Do.
50 Pd, 50 Pt	k=0.37 at 25°C	Do.

PALLADIUM ALLOYS (Cont'd)

Composition (%)	Conductivity and remarks	Reference
	w/cm deg K	
Commercial	k=0.42 at 17°C	T. Barratt and R. M. Winter (1925).
85.5 Pd, 14.5 Cu	Polycrystalline; see Fig. 26, "Pd-15% Cu"	E. Grüneisen and H. Reddemann (1934).
50 Pd, 50 Cu	Annealed; see Fig. 26, "Pd-50% Cu"	Do.
55 Pd, 45 Au	Annealed 2 hr. at 800°C; see Fig. 26, "Pd-45% Au".	Do.

PLATINUM ALLOYS

"Impure"	k=0.516 at 18°C	W. Jacger and H. Diesselhorst (1900).
90 Pt, 10 Pd	k=0.43 at 25°C	F. A. Schulse (1911).
80 Pt, 20 Pd	k=0.42 at 25°C	Do.
70 Pt, 30 Pd	k=0.36 at 25°C	Do.
60 Pt, 40 Pd	k=0.34 at 25°C	Do.
50 Pt, 50 Pd	k=0.37 at 25°C	Do.

PLATINUM ALLOYS (Cont'd)

COPPER ALLOYS (Contd)

Composition (%)	Conductivity and remarks	Reference
	w/cm deg K	
90 Pt, 10 Ir	k=0.31 at 17°C	T. Barratt and R. M. Winter (1925).
85 Pt, 15 Ir	k=0.23 at 17°C	Do.
80 Pt, 20 Ir	k=0.18 at 17°C	Do.
90 Pt, 10 Rh	k=0.30 at 17°C	Do.
96 atomic % Pt, 4 atomic % Au.	k=0.46 at 18°C	C. H. Johansson and J. O. Linds (1930).
90 atomic % Pt, 10 atomic % Au.	k=0.35 at 18°C	Do.
75 atomic % Pt, 25 atomic % Au.	k=0.24 at 18°C	Do.
55 atomic % Pt, 45 atomic % Au.	k=0.21 at 18°C	Do.

COPPER ALLOYS

See also the "COPPER-NICKEL ALLOY" graph and and tables.

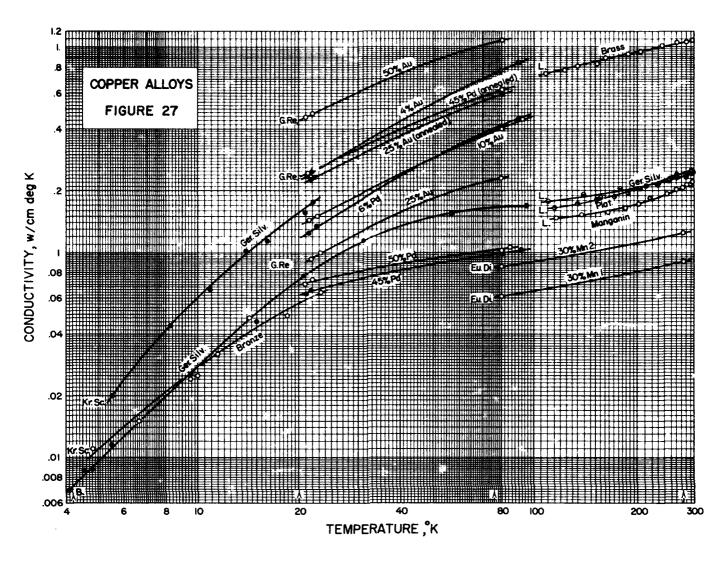
Curve	Composition (%)	Conductivity and remarks	Reference
		w/cm deg K	
•••••	About 62 Cu, 15 Ni, 22 Zn.	"Neusilber"; k=0.29 at 0°C.	L. Lorens (1881).
	About 82 Cu, 18 Zn	"Red brass"; k=1.03 at 0°C.	Do.
	About 65 Cu, 35 Zn	"Yellow brass"; $k = 0.85$ at 0°C.	Do.
	0.34 P	k=0.95 at 15°C	A. Rietzsch (1900).
	0.87 P	k=0.61 at 15°C	Do.
	1.79 P	k=0.53 at 15°C	Do.
	2.08 P	k=0.34 at 15°C	Do.
	2.35 P	k=0.27 at 15°C	Do.
	5.25 P	k=0.15 at 15°C	Do.
	1.04 As	k=1.14 at 15°C	Do.
	1.80 As	k=0.82 at 15°C	Do.
	2.66 As	k=0.54 at 15°C	Do.
	3.00 As	k=0.54 at 15°C	Do.
·····	5.02 As	k=0.20 at 15°C	Do.
•••••	85.7 Cu, 7.15 Zn, 6.39 Sn, 0.6 Ni.	"Red brass"; k=0.60 at 18°C.	W. Jaeger and H. Diesselhorst (1900).
	84 Cu, 12 Mn, 4 Ni	k=0.22 at 18°C	Do.
Fig. 27; L Brass.	70 Cu, 30 Zn	• • • • • • • • • • • • • • • • • • • •	C. H. Lees (1908).
Fig. 27; L Ger. silv.	62 Cu, 22 Zn, 15 Ni	"German silver"	Do.
LPlat	Approx. same as above.	"Platinoid"	Do.
LManganin.	84 Cu, 12 Mn, 4 Ni	"Manganine"	Do.
	82 Cu, 18 Zm	"Red brass"; "fine" crystals; k=1.27 at 273"K, 0.65 at 90"K.	A. Eucken and O. Neumann (1924).
•••••	do	"Red bram"; "large" crystals; k=1.30 at 273"K, 0.65 at 90"K.	Da.

Curve	Composition (%)	Conductivity and remarks	Reference
		w/cm deg K	
Eu. DiCu- 50% Mn, 1.	70 Cu, 30 Mn	About 48 crystals per centimeter.	A. Eucken and K. Dittrich (1927).
Eu. DiCu- 50% Mn, 2.	do	About 112 crystals per centimeter.	Do.
	3 Ag	Unannealed; k = 3.57 at 83° and 21°K.	E. Griineisen and E. Goens (1927).
	do	Annealed 3 hr at 390°C; k=4.06 at 83°K, 6.19 at 21°K.	Do.
Fig. 27; G. Re 4% Au.	95.5 Cu, 4.5 Au	Polycrystalline; unan- nealed.	E. Grüneisen and H. Reddemann (1934).
G. Re10% Au.	90.3 Cu, 9.7 Au	Polycrystalline; unan- nealed.	Do.
	75.1 Cu, 24.9 Au	Quenched from 800°C; k=0.34 at 83°K.	Do.
	do	Annealed 20 hr at 400°C; k=0.61 at 83°K.	Do.
	do	Annealed 32 hr at 360°C; k=0.63 at 83°K.	Do.
	do	Same as above except later annealed 2 hr at 820°C, then quenched; k=0.23 at 83°K.	Do.
G. Re25% Au.	do	Same as above except later annealed 5 months at room temperature.	Do.
G. Re25% Au, an- nealed.	do	Same as above except additionally annealed 30 hr at 320°C.	Do.
G. ReCu- 50% Au.	49.9 Cu, 50.1 Au	Annealed 30 hr at 320°C	Do.
G. ReCu- 50% Pd.	49.9 Cu, 50.1 Pd	Annealed	Do.
G. Re6% Pd.	93.6 Cu, 6.4 Pd	Polycrystalline; unan- nealed.	Do.
	55 Cu, 45 Pd	Annealed; $k=0.67$ at 83°K.	Do.
G. Re45% Pd.	do	Annealed 2 hr at 800°C	Do.
G. Re45% Pd, (an- nealed).	do	Further annealed 30 hr at 320°C.	Do.

COPPER ALLOYS (Cont'd)

Composition (%) ¹	Conductivity 1
	w/cm deg K
99.986 Cu, 0.0016 Fe, .02 Oz. 99.80 Cu, 0.19 Si, .02 Fe. 99.78 Cu, 0.23 Si, .02 Fe.	3.93 2
99.80 Cu. 0.19 Si02 Fe	2.13 3
99.78 Cu. 0.23 Si02 Fe	1.92 2
99.65 Cu. U.32 St. J332 Fe	1.65 *
99.53 Cu. 0.45 Si03 Fe.	1.29 *
99.06 Cu, 1.00 Si, 0.03 Fe	0.82 3
98.09 Cu, 1.98 Si, 0.05 Fe	0.51 2
96.00 Cu, 3.91 Si, 0.02 Fe.	0.34 1

 $^{^1}$ These values were determined by C. S. Smith (1935) at 20°C. Sometimes the composition percentages add up to more than 100. 2 The copper-silicon alloys were hot-rolled, cold-drawn and annealed at 700°C and were in the homogeneous α solid solution.



COPPER ALLOYS (Cont'd)

Composition (%) 1	Conductivity 1
	w/cm deg K
99.95 Cu, 0.07 Al, .01 Fe	3.52 2
99.77 Cu, 0.22 Al, .01 Fe	2.91 2
99.47 Cu, 0.47 Al, .02 Fe	2.35 2
99.20 Cu, 0.71 Al, .09 Fe	1.75 2
98.08 Cu, 1.89 Al, 0.03 Fe	1.23 *
95.25 Cu, 4.61 Al, 0.14 Fe.	0.83 2
00 15 CL 7 70 Al D 19 E	0.72 2
92.15 Cu, 7.72 Al, 0.13 Fe	0.72 2
90.56 Cu, 9.37 Al, 0.07 Fe	
89.88 Cu, 9.90 Al, 0.22 Fe	0.66 2
87.76 Cu, 12.15 Al, 0.09 Fe	0.54 2
99.94 Cu, 0.07 Mn, .01 Fe, .02 Mg	3.62 *
99.88 Cu, 0.14 Mn, .01 Fe, .01 Mg	3.28 3
99.55 Cu, 0.43 Mn, .01 Fe, .01 Mg	
99.05 Cu, 1.05 Mn, 0.01 Fe, .01 Mg	1.50 *
98.27 Cu, 1.77 Mn, 0.03 Fe, .01 Mg	1.02 3
95.34 Cu, 4.55 Mn, 0.06 Fc, .02 Mg.	0.49
90.25 Cu, 9.53 Mn, 0.18 Fe, .02 Mg, .021 C.	
80.03 Cu, 19.82 Mn, 0.09 Fe, .02 Mg, .035 C	0.15 *

¹ These values were determined by C. S. Smith (1935) at 20°C. Sometimes the composition percentages add up to more than 100.

COPPER ALLOYS (Cont'd)

Composition (%) 1	Conductivity 1	State 1
	w/cm deg K	
99.986 Cu, 0.002 Fe, .02 O2	3.94 *	
66.24 Cu, 33.72 Zn, 0.03 Pb, .01 Fe, .001 S	1.20 2	
96.94 Cu, 3.04 Zn, 0.02 Fe	2.68 2	
95.21 Cu, 4.77 Zn, 0.02 Fe		
97.49 Cu, 0.06 Fe, 27 Ni, 2.24 Be	0.86 2	Quenched.
97.49 Cu, 0.06 Fe, .27 Ni, 2.24 Be	1.03 *	Reheated.
97.49 Cu, 0.06 Fe, .27 Ni, 2.24 Be	0.74 2	Quenched, cold- drawn.
97.49 Cu, 0.06 Fe, .27 Ni, 2.24 Be	0.82 2	Reheated.
85.10 Cu, 12.97 Zn, 1.88 Pb, 0.05 Fe		
61.85 Cu, 34.79 Zn, 3.29 Pb, 0.07 Fe	1.08 2	1
65.99 Cu, 29.18 Zn, 4.02 Pb, 0.01 Fe	1.11 2	1
88.07 Cu, 3.70 Zn, 3.77 Sn, 3.83 Pb, 0.03 Fe	0.90 2	
88.08 Cu, 4.09 Zn, 3.76 Sn, 3.80 Pb, 0.02 Fc, .25 P.	0.56 2	
60.41 Cu, 37.09 Zn, 1.03 Sn, 1.12 Pb, 0.02 Fe, .18 Al, .21 Si.	1.00 2	Chill-cast.
56.01 Cu, 25.93 Zn, 0.18 mn, 17.95 Ni, 0.08 Fe, .02 C.	0.30 2	1
63.76 Cu, 19.79 Zn, 0.18 Mn, 16.29 Ni, 0.14 Fe.	.34 2	
65.51 Cu, 23.86 Zn, 0.18 Mn, 10.36 Ni, 0.08 Fe, .01 C.	.46 *	
59.76 Cu, 29.88 Zn, 0.15 Mn, 10.13 Ni, 0.04 Fe, .04 Mg.	.42 2	
64.04 Cu, 30.50 Zn, 5.41 Ni, 0.05 Fe	.59 2	1

¹ These values were determined by C. S. Smith (1935) at 20°C. Sometimes the composition percentages add up to more than 100.

² The copper-aluminum alloys were rolled and annealed at 700°C and were in the α solid solution (except the 12% Al, which was δ).

² The copper-manganese alloys were deoxidized with magnesium, hot-rolled, and annealed at 700°C.

² The miscellaneous alloys were extensively worked, annealed, and slov y cooled except where noted.

COPPER ALLOYS (Cont'd)

Composition (%) 1	Conductivity 1	State 1
	w/cm deg K	
6.57 Cu, 17.65 Zn, 13.24 Ni, 0.10 Fe, 2.23 Sn,	.31 *	Sand-cast.
10.44 Pb.		
9.08 Cu, 4.98 Ni, 0.08 Fe, 5.11 Al, 0.74 Si	.45 2	Quenched.
9.08 Cu, 4.98 Ni, 0.08 Fe, 5.11 Al, 0.74 Si	.57 2	Reheated.
9.08 Cu, 4.98 Ni, 0.08 Fe, 5.11 Al, 0.74 Si	.66 3	Furnace-cooled.
6.13 Cu, 42.34 Zn, 1.02 Ni, 0.49 Fe.	1.14 2	
9.38 Cu, 0.31 Ni, 0.52 Fe, .38 Sn, 9.41 Al	0.60 2	
5.79 Cu, 22.22 Zn, 0.01 Fe, 1.98 Al	1.00 *	
9.35 Cu, 38.36 Zn, 0.12 Mn, 1.06 Fe, 0.98 Sn,	1.01 *	
.13 Pb.		
9.654 Cu. 0.03 Fe32 Si	1.65 2	
5.61 Cu, 4.51 Mn, 0.11 Fe	0.46 2	
9.21 Cu, 0.01 Fe, .01 Si, .85 Cd	3.45 2	
8.41 Cu, 0.02 Fe, .59 Sn, .02 Si, 1.07 Cd	2.33 1	ſ
2.49 Cu, 17.76 Zn, 3.34 Mn, 1.78 Fe, 4,44 Al	0.50 *	
4.00 Cu, 1.03 Mn, 0.08 Fe, 4.68 Si	0.25 2	Sand-cast.
5.69 Cu, 0.99 Mn, 0.16 Fe, 3.23 Si	0.33 *	
8.10 Cu, 0.30 Mn, 0.06 Fe, 1.50 Si	0.54 2	ł
1.55 Cu, 14.21 Zn, 0.20 Mn, 0.04 Fe, 4.00 Si	0.28 2	Chill-cast.
5.83 Cu, 1.12 Zn, 0.02 Fe, 3.11 Si	0.37 *	
8.30 Cu, 0.20 Ni, 8.04 Sn, 13.32 Pb, 0.1 P	0.42 *	Sand-cast.
7.86 Ca, 3.05 Za, 0.03 Fe, 8.87 Sa	0.54 ±	Sand-cast.
8.36 Cu, 1.90 Zn, 0.07 Fe, 9.55 Sn	0.50 1	Sand-cast.
0.54 Cu. 36.46 Zn. 0.21 Mn. 0.73 Fe. 1.48 Sn.		Sand-cast.
0.04 AL]
9.04 Cu, 0.07 Fe, .9 Cd	2.76 2	
0.75 Cu, 0.47 Mn, 48.69 Fe, 0.05 Si, .02 C	0.99 *	1

¹ These values were determined by C. S. Smith (1935) at 20°C. Sometimes the composition percentages add up to more than 190.

² The miscellaneous alloys were extensively worked, annealed, and slowly cooled except where noted.

COPPER ALLOYS (Cont'd)

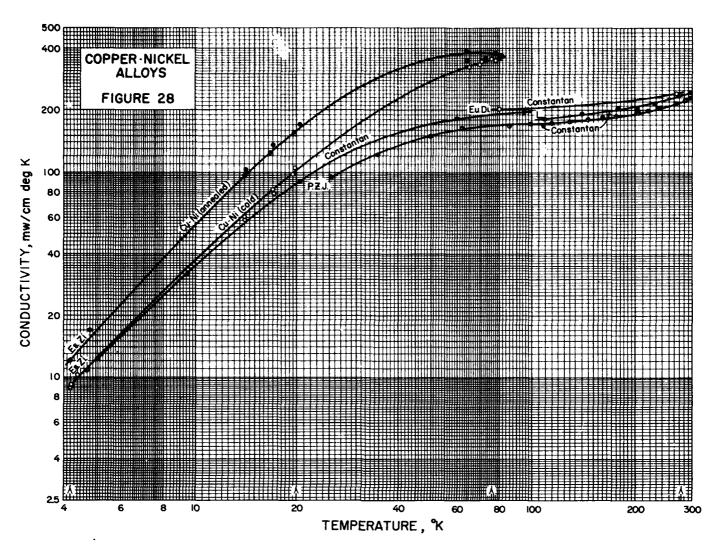
Curve	Composition (%)	Remarks	Reference
Fig. 27; Kr. 8Ger. silv.	64 Cu, 20 Zn, 16 Ni	"Neusilber"	J. Karweil and K. Schäfer (1939).
Fig. 27; Kr. Scbronse	46 Cu, 41 Zn, 13 Ni	"Silberbronse"	Do.
Fig. 29a; Al. MnGer. silv.	45.9 Cu, 42.1 Zn, 9.8 Ni, 2.0 Pb, 0.15 Fe, 0.06 Mn.	"German silver"; data fits equation $k=5.3\times10^{-4}$ T ³ .	J. F. Allen and E. Mendosa (1948).
Fig. 27, Fig. 29a; B Ger. nilv.	47 Cu, 41 Zn, 9 Ni, 2 Pb.	Mean diameter of crystals was 0.02 mm.	R. Berman (1951b).

COPPER ALLOYS (Cont'd) COMPANY AND TRADE MANUALS

Name	Nominal Composition (%)	Conductivity
		w/cm deg K
Electrolytic Tough Pitch	99.92 Cu, 0.04 Oz	3.91
		3.39
Description Oxygen-free high cond	. 99.99 CQ, U.U2 F	3.93
OXAF-it-se priku como	. 99.92 Ca	
Silver bearing	. 99.9 Cu, trace Ag	3.93
Arsenical phosphorised		1.76
Free cutting	. 99.4 Cu, 0.6 Te	3.55
Boron deoxidised		3.88
Selenium copper	.] 99.4 Cu, 0.6 Se	3.84
Leaded copper		3.84
Chromium copper		3.24
Cadmium copper		3.44
Bracece:		
Gilding	. 95 Cu, 5 Zn	2.34
Commercial bronse	. 90 Cu. 10 Zn	1.88
Bearing bronse	. 90 Cu, 9.5 Zn, 0.5 Sn	1.73
Commercial bronse		1.73
Red brass		
Low brans		1.38
Cartridge brass		
Yellow brass		1.17
Munts metal		1.21

COPPER ALLOYS (Cont'd) COMPANY AND TRADE MANUALS

Name	Nominal Composition (%)	Conductivity
		w/cm deg K
Leaded Brasses: Leaded commercial bronse	90 Cu, 9.5 Zn, 0.5 Pb	1.80
Leaded commercial bronse	89 Cu, 0.25 Zn, 1.75 Pb	1.80
Commercial bronse	90.25 Cu, 6.9 Zn, 1.75 Pb, 1 Ni	1.40
Low leaded brass	64.5 Cu, 35 Zn, 0.5 Pb	1.17 1.16
Medium leaded brass	89 Cu, 0.25 Zn, 1.75 Pb 90.25 Cu, 6.9 Zn, 1.75 Pb, 1 Ni 64.5 Cu, 35 Zn, 0.5 Pb 67 Cu, 32.5 Zn, 0.5 Pb 64.5 Cu, 34.5 Zn, 1.0 Pb 64.5 Cu, 34.5 Zn, 1.0 Pb	1.17
High leaded brass		
High leaded brass	64 Cu, 34 Zn, 2.0 Pb	1.15
Extra high leaded brass	61.5 Cu 35.5 Zn 3 Pb	1.17 1.17
Leaded Munts metal	60 Cu, 39.5 Zn, 0.5 Pb	1.21
Free cuting Munts metal	60.5 Cu, 38.4 Zn, 1.1 Pb	1.17
Forging brase	57 Cu 40 Zn 3 Ph	1.17 1.21
Leaded naval brass	62.5 Cu, 35 Zn, 2.5 Pb 61.5 Cu, 35.5 Zn, 3 Pb 60 Cu, 39.5 Zn, 0.5 Pb 60 5 Cu, 38.4 Zn, 1.1 Pb 60 Cu, 38 Zn, 2 Pb 57 Cu, 40 Zn, 3 Pb 60 Cu, 37 5 , 0.7 to 1.75 Pb,	1.17
Leaded tin bearing bronse High leaded tin bronse (bushing).	80 Ct. 10 Sp. 10 Ph	0.47
Dairy bronse	64 Cu, 8 Zn, 20 Ni, 4 Pb, 4 Sn	0.23
Leaded nickel brass	37 Cu, 4 Zn, 8 Sn, 1 Pb 80 Cu, 10 Sn, 10 Pb 64 Cu, 8 Zn, 20 Ni, 4 Pb, 4 Sn 60 Cu, 16 Zn, 16 Ni, 5 Pb, 3 Sn	0.27
Special Brasses:	1	
Admiralty metal	71 Cu, 28 Zn, 1 Sn	1.09
Naval brass		1.17
Manganese brouse	58.5 Cu, 39 Zn, 1.4 Fe, 1 Sn, 0.1 Mn.	1.09
Aluminum brass	76 Cu, 22 Zn, 2 Al	1 100
Aluminum brass. "Ambronse-474". "Ambronse-421".	76 Cu, 22 Zn, 2 Al	1.64
"Ambronse-421"	88.00 Cu, 10.0 Zn, 2.0 Sn 85.0 Cu, 14.0 Zn, 1.0 Mn	1.19 0.99
Silicon red brass	900 C. 170 7. 10 C.	0.67
Trumpet brass	81.0 Cu, 18.0 Zn, 1.0 Sn.	1.21
Arsenical admiralty	1 71.0 Cu, 28.0 Zn, 1.0 Sn, 0.04 As.	1.11 0.74
Nickel silver 18%-B	55 Cu. 27 Zn. 18 Ni	0.29
Manganese brass. Nickel silver 18%-B. Nickel silver 15%.	66 Cu, 19 Zn, 15 Ni	0.35
Leaded nickel silver 12%	81.0 Cu, 18.0 Zn, 1.0 Sn. 71.0 Cu, 28.0 Zn, 1.0 Sn, 0.04 As. 70.0 Cu, 29.0 Zn, 1.0 Sn, 0.04 As. 55 Cu, 27 Zn, 18 Ni. 66 Cu, 19 Zn, 15 Ni. 65 Cu, 20.7 Zn, 12 Ni, 2 Pb, 0.3 Mn.	0.40
Phosphor bronse 5%-A. Phosphor bronse 8%-C. Phosphor bronse 10%-D. Phosphor bronse 1.25%-E.	95 Cu, 5 Sn, trace P	1
Phosphor bronze 8%-C	95 Cu, 5 Sn, trace P	0.63
Phosphor bronse 1959 E	90 Cu, 10 Sn, trace P	0.50 2.06
radional productions	. 90.1 Cu, 1.20 Du, v.vo f	2.18
<u>D</u> o	. 1 98.24 Cu, 1.75 Sp. U.U. P	1 1.47
Do	A 2 2 C. 4 A C. A 9 E D	0.81 0.81
Do	95.17 Cu, 4.0 Sn, 0.08 P, 0.5 Fe 94.17 Cu, 5.0 Sn, 0.25 P. 93.9 Cu, 5.0 Sn, 0.1 P, 1 Pb. 93.7 Cu, 6.0 Sn, 0.3 P. 91.75 Cu, 8.0 Sn, 0.25 P. 275 Cu, 8.0 Sn, 0.25 P.	0.76
Do	94.75 Cu, 5.0 Sn, 0.25 P	0.81
Do	93.9 Cu, 5.0 Sn, 0.1 P, 1 Pb	0.83 0.57
Do	91.75 Cu. 8.0 Sn. 0.25 P.	0.62
Do		0.50
Do	87.90 Cu, 4.0 Sn, 4 Zn, 4 Pb, 0.1 P.	0.55
	0.1 F.	ļ
Special Brenzee:		
Silicon bronse A	. 96 Cu, 3 St	0.38 0.59
"Everdur-1010"	95.8 Cu. 3.1 Si. 1.1 Mn.	0.33
Silicon bronse B "Everdur-1010" Everdur-1012" "Everdur-1015" "Everdur-1014"	96 Cu, 3 Si. 97 Cu, 1.5 Si. 95.8 Cu, 3.1 Si, 1.1 Ma. 95.6 Cu, 3.0 Si, 1.0 Mn, 0.4 Pb. 98.25 Cu, 1.5 Si, 0.25 Mn. 90.75 Cu, 2.0 Si, 7.25 Al.	0.33
"Everdur-1015"	1 98.25 Cu, 1.5 Si, 0.25 Ma	0.54 0.45
5% Aluminum bronse		
	. 92 Cu. 8 Al	0.71
10% Aluminum bronze	. 88 Cu, 10 Al	0.60
Aluminum silicon bronse	. 02.0 Cu, 10 Al, 2 Ni, 2.0 Fe 91 Cu. 7 Al. 2 Si	0.38 0.38
"Calsun"		Į 0.0 <i>i</i>
Chromium copper	. 3 99.06 Cu. 0.85 Cr	1 3.20
"Hitenso-961" "Hitenso-965"	99.0 Cu, 1.0 Cd	3.44 2.33
Aluminum bronse 89-1-10	89 Cu. 10 Al. 1 Fe	0.55
Aluminum bronse 86-4-10	86 Cu, 10 Al, 4 Fe	0.59
Aluminum bronse	. 1 87.5 Cu. 9 Al. 3.5 Fe	0.59



COPPER ALLOYS (Cont'd)
COMPANY AND TRADE MANUALS

Nominal Composition (%) Conductivity State Name w/cm deg K Beryllium Coppers: Beryllium copper... 97 Cu, 2 Be, 0.25 Co.... 0.84 Solution treated, quenched. As above 1.05 plus chemically hardened. 0.84 As above plus cold-rolled. Solution treated, chemically quenched, 0.75 cold-rolled. 2 Be, 0.3 Co, balance Cu... 1.7 Be, 0.3 Co, balance Cu... 0.5 Be, 2.4 Co, balance Cu... 0.4 Be, 1.55 Co, 1.0 Ag, bal-ance Cu... 2.1 Be, 0.5 Co, balance Cu... 2.7 Be, 0.5 Co, balance Cu... 0.6 Be, 2.5 Co, balance Cu... Beryllium alloy 25... Beryllium alloy 165... Beryllium alloy 10... Beryllium alloy 50... 1.21 1.21 2.26 2.22 1.05 0.96 2.13 Beryllium alloy 20C. Beryllium alloy 275C. Beryllium alloy 10C.

COPPER-NICKEL ALLOYS
See also the "COPPER ALLOY" graph and tables.

Curve	Composition (%)	Conductivity and remarks	Reference
		w/cm deg K	
	About 62 Cv, 22 Zn, 15 Ni.	"Neusilber"; k=0.29 at 0°C.	L. Lorens (1881).
	60 Cu, 40 Ni	"Constantan"; k=0.23 at 18°C.	W. Jaeger and H. Diesselhorst (1900).
	54 Cu, 46 Ni	k=0.21 at 18°C	E. Grüneisen (1900).
Fig. 27; L Ger. silv.	62 Cu, 22 Zn, 15 Ni	"German silver"	C. H. Lees (1908).
LPlat	Approx. same as above.	"Platinoid"	Do.
	10 Cu, 40 Ni	"Eureka" or constantan; $k=0.21$ at 17°C.	T. Barratt and R. M. Winter (1925).
Fig. 28; Eu. Dicon- stantan.	60 Cu, 40 Ni	51 crystals per cm; also measured samples with other crystal size.	A. Eucken and K. Dittrich (1927).
	1 Ni	k=1.50 at 83°K, 0.62 at 21°K.	E. Grüneisen and E. Goens (1927).
Fig. 27; Kr. ScGer. silv.	64 Cu, 16 Ni, 20 Zn	"Neusilber"	J. Karweil and K. Schäfer (1939).

COPPER-NICKEL ALLOYS (Cont'd)

Curve	Composition (%)	Conductivity and remarks	Reference
		w/cm deg K	
Fig. 29a; Al, MnGer. silv.	45.9 Cu, 42.1 Zn, 9.8 Ni, 2.0 Pb, 0.15 Fe, 0.05 Mn.	"German silver"; data fits equation $k = 5.3 \times 10^{-4}$ T ³ .	J. F. Allen and E. Mendosa (1948).
:	63 Cu, 20 Ni, 17 Zn	"Nickel-silver"; k = 25.5 mw/cm deg at 10°K, 48.5 at 15°K, 71.1 at 20°K.	K. R. Wilkinson and J. Wilks (1949).
	70 Cu, 30 Ni	"Cupro-nickel"; k=20.9 mw/cm deg at 10°K, 35.6 at 15°K, 50.2 at 20°K.	Do.
Fig. 28; P.Z.J constan- tan.	55 Cu, 45 Ni		R. W. Powers, J. B. Ziegler, and H. L. Johnston (1951c).
Fig. 29a; Hu Cu- 20% Ni.	80 Cu, 20 Ni	Also obtained k = 127 mw/ cm deg at 21.9°K and 79.9 at 16.3°K.	J. K. Hulm (1951).
Fig. 28; B Constan- tan.	60 Cu, 40 Ni		R. Berman (1951b).
Figs. 27, 29a; B Ger. silv.	47 Cu, 41 Zn, 9 Ni, 2 Pb.	Mean diameter of crystals was 0.02 mm.	Do.
Fig. 27; Es. Zi Cu - 10% Ni, annealed.	90 Cu, 10 Ni	Two samples which were annealed, one a single crystal.	I. Estermann and J. E. Zimmermann (1952).
Es.ZiCu- 10% Ni, cold.	do	Two samples which were cold-worked.	Do.

COPPER-NICKEL ALLOYS (Cont'd)

Composition (%)	Conductivity
99.73 Cu, 0.28 Ni, .01 Fe, .03 Mg. 99.47 Cu, 0.54 Ni, .02 Fe, .04 Mg. 97.94 Cu, 1.97 Ni, 0.02 Fe, .04 Mg. 94.92 Cu, 5.09 Ni, 0.01 Fe, .03 Mg. 89.90 Cu, 10.07 Ni, 0.02 Fe, .03 Mg, .02 C.	2.92 1, 2 1.72 1, 2 1.00 1, 2
89.50 Cu, 15.07 Ni, 0.05 Fe, .01 Mg, .03 Mn 79.68 Cu, 19.79 Ni, 0.23 Fe, .30 Mg. 69.54 Cu, 30.23 Ni, 0.05 Fe, .05 Mg, .13 Mn	0.47 1, 2

Composition (%)	Conductivity	State
	w/cm deg K	
64.14 Cu, 18.38 Ni, 0.19 Fe, 17,06 Zn, 0.3 Mn, .02 C.	0.33 1, 8	
63.37 Cu, 19.89 Ni, 0.14 Fe, 8.22 Zu, 3.31 Sn, 5.4 Pb. 0.23 Mn.	0.28 1, 1	Sand-cast.
96,05 Cu, 3.01 Ni, 0.004 Fe, .88 Si	0.76 1, 3 1.58 1, 3 1.69 1, 3	Quenched. Rehested. Furnace-cooled.
74.07 Cu, 19.96 Ni, 0.09 Fe, 5.31 Zn	0.39 1, 8 0.28 1, 8	

COPPER-NICKEL ALLOYS (Cont'd) COMPANY AND TRADE MANUALS

Name	Nominal composition (%)	Conductivity
		w/cm deg K
Cupro-nicke! 30%	70 Cu, 30 Ni	0.29
Cupro-nickel 10%	88.5 Cu, 10 Ni, 1.5 Fe	0.47
Nickel silver 18%-A	65 Cu, 18 Ni, 17 Zn	0.33
Nickel silver 18%-B	55 Cu, 18 Ni, 27 Zu	0.25
Nickel silver 15%	66 Cu, 15 Ni, 19 Zn	0.35
Constantan	55 Cu, 45 Ni	0.23
Dairy bronze	1 64 Cu. 20 Ni. 8 Zn. 4 Pb. 4 Sn	0.23
Leaded nickel brass.	60 Cu, 16 Ni, 16 Zn, 5 Pb, 3 Sn	0.27
Leaded nickel silver 12%.	65 Cu, 12 Ni, 20.7 Zn, 2 Pb, 0.3 Mn	0.40

SILVER ALLOYS

Curve	Composition (%)	Conductivity and remarks	Reference
		w/cm deg K	
• • • • • • • • • • • • • • • • • • • •	90 Ag, 10 Pd	k=1.41 at 25°C	F. A. Schulse (1911).
	80 Ag, 20 Pd	k=0.84 at 25°C	Do.
	70 Ag, 30 Pd	k=0.57 at 25°C	Do.
	60 Ag, 40 Pd	k=0.45 at 25°C	Do.
	50 Ag, 50 Pd	k=0.32 at 25°C	Do.
	90 Ag, 10 Pt	k=0.98 at 25°C	Do.
	75 Ag, 25 Pt	k=0.38 at 25°C	Do.
	70 Ag, 30 Pt	k=0.31 at 25°C	Do.
	67 Ag, 33 Pt	k=0.30 at 25°C	Do.
Fig. 26; G. Re Ag-0.4% Au.	99.63 Ag, 0.37 Au		E. Grüneisen and H. Reddemann (1934).
G. Re-Ag- 25% Au.	75 Ag, 25 Au	Single crystal	Do.
G. ReAu 50% Ag.	50 Ag, 50 Au	Single crystal	Do.
PoAg Solder.	50 Ag, 15.5 Cu, 16.5 Zn, 18 Cd.	"Easy-fio"; flame annealed :	R. L. Powell (1953).

GOLD ALLOYS

	90 Au, 10 Pd	k=0.98 at 25°C	F. A. Schulse (1911).
	80 Au, 20 Pd	k=0.59 at 25°C	Do.
	70 Au, 30 Pd	k=0.44 at 25°C	Do.
,	60 Au, 40 Pd	k=0.40 at 25°C	Do.
	50 Au, 50 Pd	k=0.36 at 25°C	Do.
	90 Au, 10 Pt	k=0.76 at 25°C	Do.
	80 Au, 20 Pt	k=0.41 at 25°C	Do.
	70 Au, 30 Pt	k=0.30 at 25°C	Do.
	60 Au, 40 Pt	k=0.26 at 25°C	Do.

¹ The values were determined by C. S. Smith, E. W. Palmer (1935) at 20°C. Sometimes the composition percentages add up to more than 100.

² The copper-nickel alloys were deoxidised with magnesium, cold-rolled, and annealed at 800°C.

³ The Miscellaneous alloys were extensively worked, annealed, and slowly cooled except where noted.

GOLD ALLOYS (Cont'd)

Curve	Composition (%)	Conductivity and remarks	Reference
		₩/cm deg K	
	92 atomic % Au, 8 atomic % Pt.	k=0.80 at 18°C	C. H. Johansson and J. O. Linde (1930).
	84 Au, 16 Pt	k = 0.48 at 18°C	Do.
	68 Au, 32 Pt	k=0.23 at 18°C	Do.
,	55 Au, 45 Pt	k=0.21 at 18°C	Do.
Fig. 26; G. Re Au-50% Ag.	50 Au, 50 Ag	Single crystal	E. Grüneisen and 11. Reddemann (1934).
G. ReAu- 25% Ag.	75 Au, 25 Ag	Single crystal	Do.
G. ReAu- 26% Cu.	73.7 Au, 26.4 Cu	Polycrystalline	Do.
G. Re,-Au- 9% Cu.	91 Au, 9 Cu	Polycrystalline	Do.
************	50.1 Au, 49.9 Cu	Quenched from 800°C; k=0.193 at 86°K.	Do.
	do	Same as above except an- nealed 22 hr at 360°C; k=1.28 at 85°K.	Do.
	do	Requenched from 800°C; k=0.23 at 83°K.	Do.
G. ReAu- 50% Cu.	do	Annealed 30 hr at 320°C	Do.
G. ReAu- 9% Pd.	91.2 Au, 8.8 Pd	Tempered at 800°C for 2 hr.	Do.
	83 Au, 17 Pd	Annealed 2 hr at 800°C; approx. same curve as "Ag-25% Au".	Do.
	69 Au, 25 Ag, 6 Pt	k = 0.53 at room temperature.	Trade Manual.

INDIUM, THALLIUM ALLOYS

	indion, intimion amous					
,	66 ТІ, 34 РЬ	For a sample with "large" crystals, k=0.22 at 273° K, 0.13 at 80°K; for a sample with "small" crystals, k=0.23 at 273° K, 0.14 at 80°K.	A. Eucken and K. Dittrich (1927).			
	67 Tl, 34 Pb by atomic percent.	Measured relative change of thermal conductivity when the alloy became superconductive.	W. J. de Haas and H. Brem- mer (1932).			
Fig. 29, 29a; Br. HIn- 9% Pb.	91.4 In, 8.6 Pb by atomic percent.	Became superconducting at 4.2°K.	H. Bremmer and W. J. de Haas (1936).			
Br. HPb- 50% In.	50 In, 50 Pb by atomic percent.	Became superconducting at 6.54°K.	Do.			
Fig. 29a; Hu-In- 10% Tl.	90 In, 10 Tl by atomic percent.	Single crystal; measured both in the normal and superconducting state; transition temperature about 3.4°K.	J. K. Hulm (1952b).			

ZINC ALLOYS

Trade Designation	Nominal composition (%)	Conductivity
		w/cm deg K
"Zamak-3"		1.13
"Zamak-5" "Zamak-2"		1.09 1.05
Comm. rolled	99.8 Zn, 0.08 Pb	1.09
Do	99.8 Zu, 0.06 Pb, 0.06 Cd	1.09
Rolled zine alloy, "Zilloy-15".	98.7 Zn, 1 Cu, 0.01 Mg	1.05

CADMIUM ALLOYS

Curve	Composition (%)	Remarks	Reference
Fig. 29; Eu. Ge Cd-33% Sb.	66.7 Cd, 33.3 Sb		A. Eucken and G. Gehlhoff (1912).
Eu. GeCd- 50% Sb.	50 Cd, 50 Sb		Do.

MERCURY ALLOYS

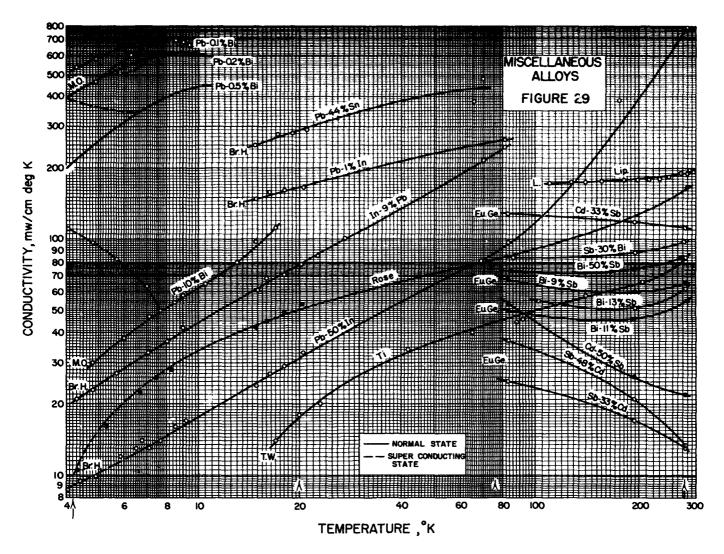
Composition (%)	Remarks	Reference	
98.8 Hg, 1.19 In	Measured ratio of conductivities in normal and superconducting states.	J. K. Hulm (1950).	
	See also the graph and table under "Metal- lic Elements".		

TIN ALLOYS

Curve	Composition	Remarks	Reference
Figs. 18a, b.	Up to 4% mercury	See table under Figs. 18a, b, "Metallic Elements".	J. K. Hulm (1950).
D ₀	Up to 3% indium	do	B. B. Goodman (1953).

TIN ALLOYS (Cont'd) COMPANY AND TRADE MANUALS

Name	Nominal Composition (%)	Conductivity
		w/cm deg K
Eutectic soft solder Tin foil	63 Sn, 37 Pb	0.50 0.59

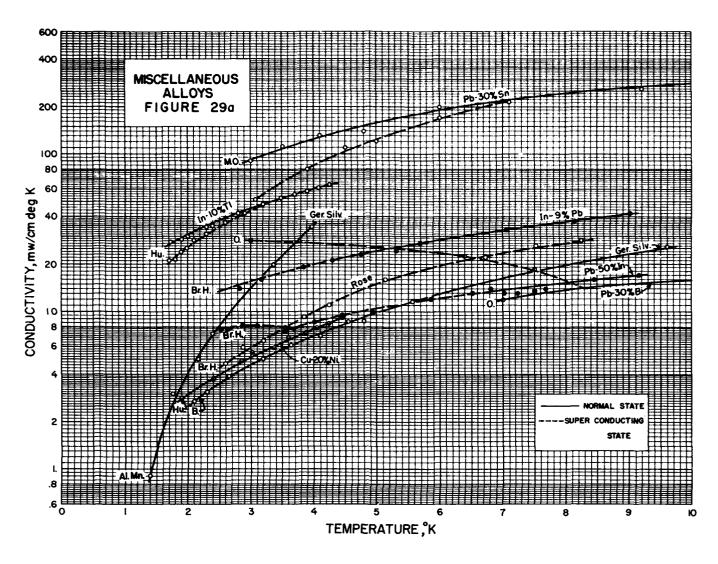


LEAD ALLOYS

Curve	Composition (%)	Remarks	Reference
Fig. 29; Br. HPb- 44% Sn.	56 Pb, 44 Sn	Not in solid solution	H. Bremmer and W. J. de Haas (1936).
	90 Pb, 10 Bi	Measured conductivity in intermediate state and as a function of mag- netic field.	K. Mendelssohn and R. B. Pontius (1937).
Fig. 29a; M. OPb- 30% Sn.	70 Pb, 30 Sn	Measured in normal and superconductive states.	K. Mendelssohn and J. L. Olsen (1950a).
Fig. 29; M. OPb- 0.1% Bi.	99.9 Pb, 0.1 Bi		Do.
М.ОРЬ- 10% Ві.	90 Pb, 10 Bi	Note that the thermal con- ductivity in the super- conductive state was higher than in the nor- mal state.	Do.
M. OPb- 0.2% Bi.	99.8 Pb, 0.2 Bi	Measured in normal and superconductive states.	K. Mendelssohn and J. L. Olsen (1950c).
М. ОРЬ- 0.5% Ві.	99.5 Pb, 0.5 Bi	do	Do.
Fig. 29a; O Pb-30% Bi.	70 Pb, 30 Bi	do	J. L. Olsen (1952).

LEAD ALLOYS (Cont'd)

Name	Nominal composition (%)	Conductivity	
		w/cm deg K	
Corroding lead	99.73 Pb	0.35	
1% antimonial lead		0.33	
Hard lead	96 Pb, 4 Sb	0.31	
Do		0.29	
% antimonial lead	92 Pb, 8 Sb	0.27	
Grid metal		0.27	
5-95 soft solder		0.36	
20-80 soft solder		0.37	
50-50 soft solder		0.46	
Lead base babbitt		0.24	
Do		0.24	



BISMUTH ALLOYS

Curve	Composition (%)	Remarks	Reference
Fig. 29; L Lip.	50 Bi, 25 Pb, 14 Sn, 11 Cd.	"Lipowits alloy"	C. H. Lees (1908).
Ge. NeBi- 50% Sb.	50, Bi, 50 Sb		G. Gehlhoff and F. Neumeier (1913).
Ge. NeBi- 20% Sb.	80 Bi, 20 Sb		Do.
Ge. NeBi- 13% Sb.	87 Bi, 13 Sb		Do.
Ge. NeBi- 11% Sb.	89 Bi, 11 Sb		Do.
Ge. NeBi- 9% Sb.	91 Bi, 9 Sb		Do.
Fig. 29, 29a; Br. H Rose.	50 Ві, 25 Рь, 25 Sn	"Rose's metal"	H. Bremmer and W. J. de Haas (1936).

ANTIMONY ALLOYS

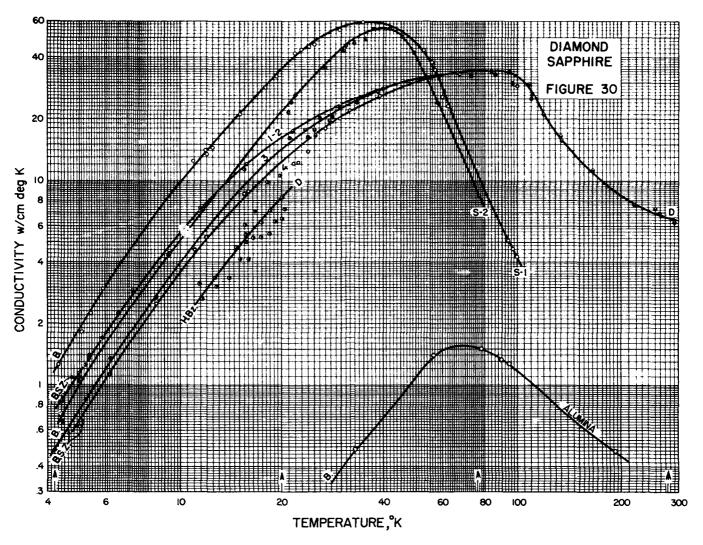
Curve	Composition (%)	Remarks	Re erence
Fig. 29; Eu. GeCd- 50% Sb.	50 Sb, 50 Cd		A. Eucken and G. Gehlhoff (1912).
Eu. GeSb- 48% Cd.	51.7 Sb, 48.3 Cd		Do.
Eu. GeSb- 33% Cd.	66.7 Sb, 33.3 Cd	"Very hard"	Do.
Ge. NeSb- 30% Bi.	70 Sb, 30 Bi		G. Gehlhoff and F. Neumeier (1913a).
Ge. NeBi- 50% Sb.	50 Sb, 50 Bi		Do.

2.4. Dielectric Crystals

This section is not comprehensive but is representative of the dielectrics. There have been few measurements on the conductivity of dielectrics at low temperatures. However, three series of experiments especially worth noting in this short summary are A. Eucken and G. Kuhn (1928), W. J. de Haas and T. Biermasz (in late 1930's), and R. Berman and others of the Clarendon Laboratory at Oxford (1950's).

The following miscellaneous dielectrics were measured by A. Eucken and G. Kuhn (1928) (all percentages are mole percent):

Name	Remarks		activity n deg K	Name	Remarks		ctivity n deg K
	Technal as	83° K	273° K			83° K	273° K
Marble	Small crystals, 99.9% CaCO1	42	33	25% KBr,	Pressed at 8,000 atm	46	33
Do	99.99% CaCO2	54	38	75% KCl.			ł
Do	Large crystals	50	33	10% KBr.	[do	80	50
Calcite	Main crystal axis perpendicular	180	46	90% KCl.	1		1
	to rod axis.	-00		50% KCl,	ldo	188	71
Do		293	54	50% NaCl.]
20	axis.	450	"	KNO	do	17	21
vlvite	Natural crystal	159	75	Mercuric	do	17	13
Ci.	Pressed at 8,000 atm	314	88	chloride.			1
= = =	From a melt.	402	92		do	109	25
			92		do	67	25
	do	343				33	13
vaCl	Pressed at 8,000 atm	251	71		do		
Rock salt	do	180	63	Copper		29	21
ylvite	do	343	84	Sulfate.			
⟨C]	Pressed at 1,250 atm	243	75	Magnesium-		25	25
	Pressed at 2,500 atm	368	92	sulfate.	[i
	Pressed at 8,900 atm	402	96	K4(FeCNs)		17	17
(Βr	Pressed at 8,000 atm	92	38	Chrom alum		13	21
laBr	do	50	25	Potassium	l	13	21
	do	121	29	alum.	1		1
CF.	do	234	71	Potassium	Main crystal axis perpendicular	17	21
VaF	do	519	105	bichromate.	to rod axis.		
	do	59	33	Do	Main crystal axis parallel to rod	17	17
	do	29	21	20	axis.		
0% KBr.	do	50	29	Topaz	Mineral		234
10% KCl.		90	45	Zincblend		63	264
10% KCi. 5% KBr.	do	29	21		do	88	84
	αο	29	21		dodo	38	46
25% KCl.	_	~-	1	Tourmaline		90	40
50% KBr, 50% KCl.	do	25	25]

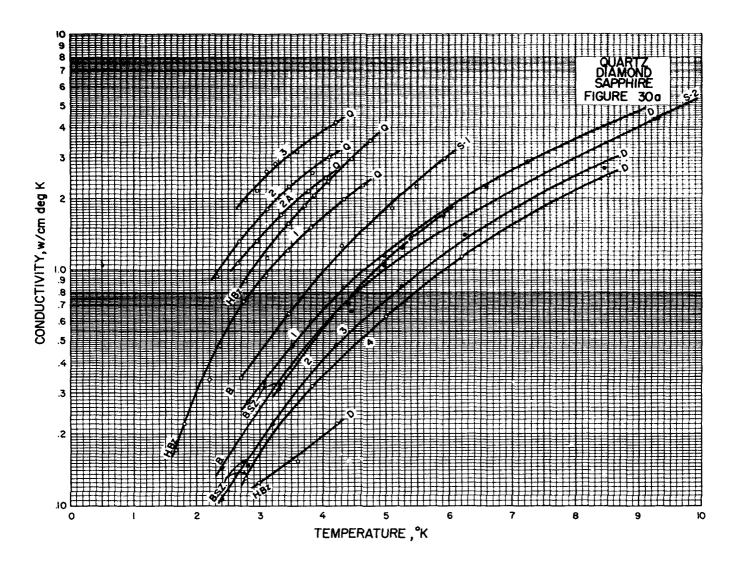


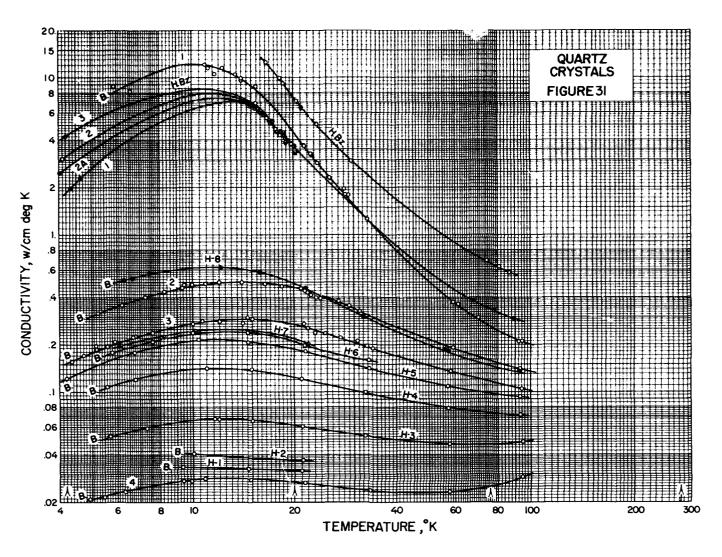
DIAMOND

Curve	Remarks	Reference
Fig. 30 and 30a; H.BzD.	In addition to the two curves, they obtained $k=14.3$ at 89° K.	W. J. de Haas and T. Biermasz (1938).
B. S. Z D1	Measured the "size effect" in diamond crystals of square cross-section. #1 was 3.9 mm wide, #2 was 3.1 mm wide, #3 was 1.7 mm wide, #4 was 1.1 mm wide. They used a type I stone. All were several centimeters long.	R. Berman, F. E. Simon, and J. M. Ziman (1953).

SAPPHIRE

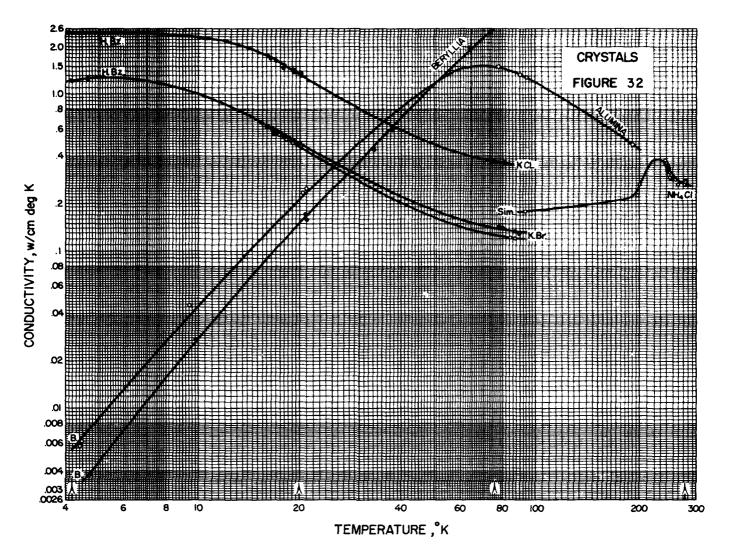
Curve	Remarks	Reference
Figs. 30, 30a; B S-1.	Curve S-1; artificial single crystal sapphire (corundum); 6 mm long, diameter of 3 mm; at lowest temperature, $k=2.7 \times 10^{-2} \mathrm{T}^2$; main crystal axis inclined 36° to rod axis.	R. Berman (1951).
Figs. 30, 30a; B S-2.	Same crystal as above except 1.5 mm diameter.	R. Berman (1952).
Figs. 30, 32; "Alumina".	Sintered alumina; density 3.70 g/cm³ (95% of single crystal); grain sizes about 5 to 30 microns.	Do.





QUARTZ

401212		
Curve	Remarks	Reference
Figs. 31, 30a; H. Bz.	A single crystal with principal axis parallel to rod axis; 5 cm long, 0.3 cm diameter.	W. J. de Haas and T. Biermasz (1935).
H. B.z-1	A single crystal with rod axis per- pendicular to principal crystal axis and parallel to bisector of two binary axes; diameter 0.216 cm.	W. J. de Haas and T. Biermasz (1937).
H. Bz2	Same as above, except diameter 0.454 cm.	W. J. de Haas and T. Biermasz (1938a)
H. Bz2A	Same rod as 2, except diameter ground down to 0.359 cm.	W. J. de Haas and T. Biermasz (1938b).
Fig 31; B1, 2, 3, 4; H 1, 2, 3, 4, 5, 6, 7, 8.	Single crystal 5 cm long; square cross-section, 5 mm on a side; rod length perpendicular to principal axis; #1 was without neutron irradiation; #2 was with 1 unit irradiation; #2 was with 1 unit irradiation of 1.4 units; #4, third additional irradiation of 16.5 units. The "H" curves were after heating as follows: #1, 300°C for 8 hrs.; #2, 400°C for 6 hrs.; #3, 500°C for 6 hrs.; #4, 565°C for 6 hrs.; #5, 540°C for 60 hrs.; #6, 540°C for 67 hrs.; #7, 600°C for 1 hr.; #8, 700°C for 6 hrs.	R. Berman (1951).

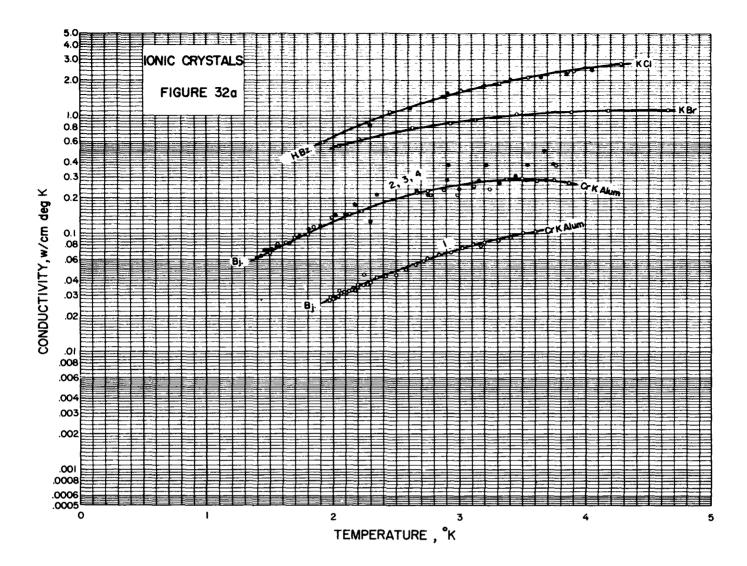


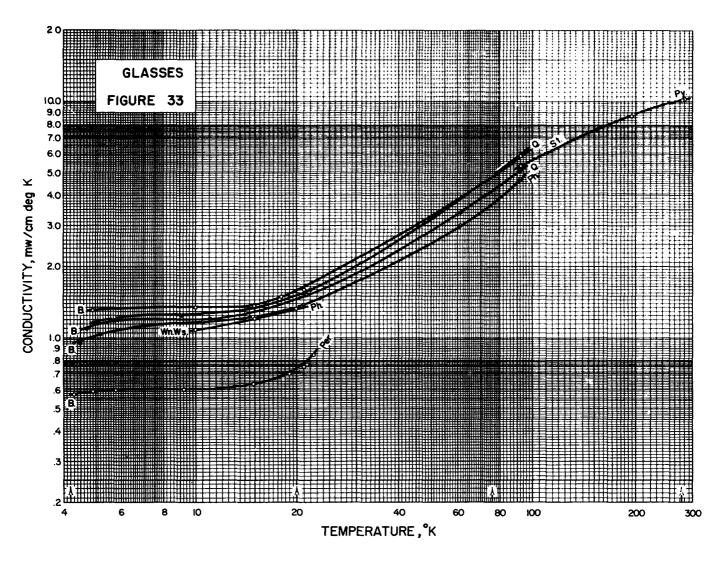
IONIC CRYSTALS

Curve	Remarks	Reference
Fig. 32; H. Bz KBr.	Measured a long potassium bro- mide crystal of approx. 3 cm di- ameter; the lower branch of the curve near 80°K is for a sample with soldered contacts; the up- per, amalgam contacts.	W. J. de Haas and T. Biermaaz (1937).
H. BzKCl	Measured along potassium chlor- ide crystal of square cross-sec- tion with a side of 0.252 cm.	Do.
•••••	Measured the change in conductivity with change in crystal cross-section for several KCl crystals.	W. J. de Haas and T. Biermasz (1938a).
Fig. 32a; Bj CrK Alum.	Measured the conductivity of chromium potassium alum used in magnetic thermometry. Found that the conductivity depended on the rate of cooling of the alum. Curve 1 was for the sam- ple cooled most rapidly.	D. Bijl (1949).
Fig. 32; Si NH ₄ Cl.		C. V. Simson (1951).

BERYLLIA

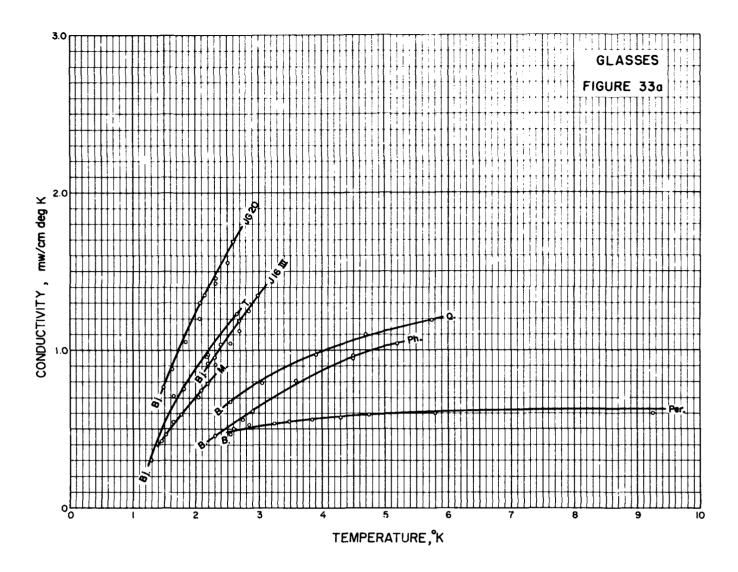
Curve	Remarks	Reference
Fig. 32-B. "Beryllia".	Sintered; density 2.94 g/cm² (97% of single crystal); crystallites with dimensions between 10 and 40 microns; k=3.8 at 90°K.	R. Berman (1952).
		<u> </u>





DISORDERED DIELECTRICS

Curve	Composition	Reference
Fig. 33; StPy.	Pyrex glass	R. W. B. Stephens (1932).
Fig. 33; Wn.Ws Ph.	"Phoenix"; boro-silicate glass	K. R. Wilkinson and J. Wilks (1948).
Fig. 33a; BjJ.G.20.	Jena Gerate 20 glass	D. Bijl (1949).
BjT	Thuringian glass	Do.
BjJ16 III	Jena 16 III glass	Do.
BjM	Monax glass	Do.
Figs. 33, 33a; BPh.	"Phoenix" glass	R. Berman (1951).
BQ	Quartz glass; upper curve is for a sample rod with approx. 7.5 mm diameter; lower curve, 6 mm	Do.
BPer	diameter.	Do.



3. Bibliography*

- J. F. Allen and E. Mendoza, Thermal conductivity of copper and German silver at liquid helium temperatures, Proc. Camb. Phil. Soc. 44, 280-288 (1948).
- F. A. Andrews, R. T Webber, and D A. Spohr, Thermal conductivities of pure metals at low temperatures, I. Alumium, Phys. Rev. 84, 994-996 (1951).
- S. S. Ballard, L. S. Combes, and K. A. McCarthy, A comparison of the physical properties of barium fluoride and calcium fluoride, J. Opt. Soc. Am. 42, 684-685 (1952) L.
- T. Barratt and R. M. Winter, Das thermische Leitvermögen von Drähten und Stäben, Ann. Physik 77, 1-15 (1925).
- R. Berman, Thermal conductivity of glasses at low temperatures, Phys. Rev. 76, 315-316 (1949) L.
- R. Berman, The thermal conductivities of some dielectric solids at low temperatures, Proc. Roy. Soc. (London) A208, 90-108 (1951a).
- R. Berman, The thermal conductivity of some alloys at low temperatures, Phil. Mag. 42, 642-650 (1951b).
- R. Berman, The thermal conductivity of some polycrystalline solids at low temperatures, Proc. Phys. Soc. (London) A65, 1029-1040 (1952a).
- R Berman, The thermal conductivity of disordered solids at low temperatures, Bull. Inst. Int. du Froid, Annexe 1952-1 (1952b).
- R. Berman, The thermal conductivity of dielectric solids at low temperatures, Adv. in Physics (Phil. Mag. Supp.) 2, 103-140 (1953).
- R. Berman and D. K. C. MacDonald. The thermal and electrical conductivity of sodium at low temperatures, Proc. Roy. Soc. (London) A209, 368-375 (1951).
- R. Berman and D. K. C. MacDonald, The thermal and electrical conductivity of copper at low temperatures, Proc. Roy. Soc. (London) A211, 122-128 (1952).
- R. Berman, F. E. Simon, and J. Wilks, Thermal conductivity of dielectric crystals; the "Umklapp" process, Nature 168, 277-280 (1951).
- R. Berman, F. E. Simon, and J. M. Ziman, The thermal conductivity of diamond at low temperatures, Proc. Roy. Soc. (London) A220, 171-183 (1953).
- C. E. Berry, The effect of an electric field on the thermal conductivity of glass, J. Chem. Phys. 1355-1356 (1949).
- C. C. Bidwell, Thermal conductivity and specific heat of lithium, Phys. Rev. 25, 896 (1925A).
- C. C. Bidwell, Therm conductivity of lithium, sodium, and lead to —250° C, Phys. Rev. 27, 819 (1926a) A.
- C. C. Bidwell, Thermal conductivity of Li and Na by a modification of the Forbes bar method, Phys. Rev. 28, 584-597 (1926b).
- C. C. Bidwell, A simple relation between thermal conductivity, specific heat and absolute temperature, Phys. Rev. 32, 311-314 (1928).
- C. C. Bidwell, Thermal conductivity of lead and of single and poly crystal zinc, Phys. Rev. 33, 249-251 (1929).
- C. C. Bidwell, Thermal conductivity of metals, Phys. Rev. 58, 561-564 (1940).
- D. Bijl, Some experiments on magnetic thermometry and heat conductivity of chromium potassium alum and glass at low temperatures, Physica 4, 684-693 (1949).
- *References followed by A are abstracts; by L, Letters to the Editor.

- H. Bremmer and W. J. de Haas, On the conduction of heat by some metals at low temperatures, Physica 3, 672-686 (1936).
- H. Bremmer and W. J. de Haas, On the heat conductivity of superconductivity alloys, Physica 3, 692-704 (1936).
- P. W. Bridgman, Thermal conductivity and thermoelectromotive force of single metal crystals, Proc. Nat. Acad. Sci. 11, 608-612 (1925).
- R. A. Buerschaper, Thermal and electrical conductivity of graphite and carbon at low temperatures, J. Appl. Phys. 15, 452-454 (1944).
- J. E. Calthrop, The effects of torsion on the thermal and electrical conductivities of metals, Proc. Phys. Soc. (London) 36, 168-175 (1924).
- J. E. Calthrop, The effects of torsion upon the thermal and electrical conductivities of aluminum, with special reference to single crystals, Proc. Phys. Soc. (London) 38, 207-214 (1926).
- C. H. Cartwright, Wiedemann-Franzsche Zahl, Wärmeleitfahigkeit und thermoelektrische Kraft von Tellur, Ann. Physik 18, 656-678 (1933).
- C. S. Chow, The thermal conductivity of some insulating materials at low temperatures, Proc. Phys. Soc. (London) 61, 206-216 (1948).
- W. F. Chubb, The thermal and electrical conductivities of metals and alloys, Phil. Mag. 30, 323-330 (1940).
- J. R. Clarke, On the thermal conductivity of some solid insulators, Phil. Mag. 40, 502-504 (1920).
- M. Cox, Thermal and electrical conductivities of tungsten and tantalum, $F_{-1.5}$. Rev. 64, 241-7 (1943).
- A. P. Crary, Thermal conductivity of Acheson graphite, Physics 4, 332-333 (1933).
- T. M. Dauphinee, D. G. Ivey, and H. D. Smith, The thermal conductivity of elastomers under stretch and at low temperatures, Can. J. Research A28, 596-615 (1950).
- D. P. Detwiler and H. A. Fairbank, Thermal conductivity in the intermediate state of pure superconductors, Phys. Rev. 86, 574 (1952a).
- D. P. Detwiler and H. A. Fairbank, The thermal resistivity of superconductors, Phys. Rev. 88, 1049-1052 (1952b).
- J. W. Donaldson, Thermal conductivities of industrial non-ferrous alloys, J. Inst. Metals 34, 43-56 (1925).
- H.-D. Erfling and E. Grüneisen, Weitere Untersuchungen an Berylliumkristallen im transversalem und longitudinalen Magnetfeld, Ann. Physik 41, 89-99 (1942).
- S. Erk, A. Keller and H. Poltz, Über die Wärmeleitfähigkeit von Kunstoffen, Physik. Z. 38, 394-402 (1937).
- Estermann and J. E. Zimmerman, Heat conduction in alloys and semi-conductors at low temperatures, Tech. Report No. 6 (Carnegie Institute of Technology) (June 1951).
- I. Estermann and J. E. Zimmerman, Heat conduction in alloys at low temperatures, J. Appl. Phys. 23, 578-588 (1952).
- A. Eucken, Über die Temperaturabhängigkeit der Wärmeleitfähigkeit fester Nichtmetalle, Ann. Physik 34, 185-221 (1911a).
- A. Eucken, Die Wärmeleitfähigkeit einiger Kristalle bei tiefen Temperaturen, Verhand. Deutsch. Physik. Gesell. 13, 829-835 (1911b).
- A. Eucken, Die Wärmeleitfähigkeit einiger Kristalle bei tiefen Temperaturen, Physik. Z. 12, 1005-1008 (1911c).

- A. Eucken, Zur Kenntnis des Wiedemann-Franzschen Gesetzes, III, Z. physik. Chem. 134, 220-229 (1928).
- A. Eucken and K. Dittrich, Zur Kenntnis des Wiedemann-Franzschen Gesetzes, II, Z. physik. Chem. 125, 211-228 (1927).
- A. Eucken and G. Gehlhoff, Elektrisches, thermisches Leitvermögen und Wiedemann-Franzsche Zahl der Antimon cadmiumlegierungen zwischen 0° und —190° C, Verhand. Deutsch. Physik. Gesell 14, 169-182 (1912).
- A. Eucken and G. Kuhn, Ergebnisse neuer Messungen der Wärmeleitfähigkeit fester krystallisierter Stoffe bei 0° und —190° C, Z. physik. Chem. 134, 193-219 (1928).
- A. Eucken and O. Neumann, Zur Kenntnis des Wiedemann-Franzschen Gesetzes, I, Z. physik. Chem. 111, 431-446 (1924).
- A. Eucken and E. Schröder, Das Wärmeleitvermögen einiger verfestigter Flüssigkeiten und Gase (Benzol, Bromwasserstoff, Stickoxydul), Ann. Physik. 36, 609-620 (1939).
- A. Eucken and H. Warrentrup, Eine Apparatur zur Messung der Wärmeleitfähigkeit von Metalblechen, Z. tech. Phys. 16, 99-104 (1935).
- C. G. B. Garrett, The thermal conductivity of potassium chrome alum at temperatures below one degree absolute, Phil. Mag. 41, 621-630 (1950).
- G. Gehlhoff and F. Neumeier, Uber die thermischen und elektrischen Eigenschaften der Wismut-Antimom-Legierungen zwischen —190° und +100° C, Verhand. Deutsch. Physik. Gesell. 15, 876-896 (1913a).
- G. Gehlhoff and F. Neumeier, Beiträge zur Kenntnis der thermischen und elektrischen Eigenschaften von gepressten Pulvern aus Antimon, Wismut und Bleiglanz, Verhand. Deutsch. Physik. Gesell. 15, 1069-1081 (1913b).
- G. Gehlhoff and F. Neumeier, Wärmeleitvermögen, elektrisches Leitvermögen, Thermokraft und Wiedemann-Franzsche Zahl des Quecksilbers zwischen —190° und +150° C . . .", Verhand. Deutsch. Physik. Gesell. 21, 201-217 (1919).
- A. N. Gerritsen, A review of the magnetic properties of Bi, Nederland. Tijdschr. Naturkunde 10, 160-170 (1943).
- E. Giebe, Über die Bestimmung des Wärmeleitvermögens bei tiefen Temperaturen, Verhand. Deutsch. Physik. Gesell. 5, 60-66 (1903).
- E. Goens and E. Grüneisen, Elektrizitäts und Wärmeleitung in Zink- und Cadmiumkristallen, Ann. Physik 14, 164-180 (1932).
- B. B. Goodman, The thermal conductivity of superconducting tin below 1° K, Proc. Phys. Soc. (London) A66, 217-227 (1953).
- E. Griffiths and G. W. C. Kaye, The measurement of thermal conductivity, Proc. Roy. Soc. (London) A104, 71-98 (1923).
- J. H. Gray, On a method of determining the thermal conductivity of metals, with application to copper, silver, gold, and platinum, Phil. Trans. Roy. Soc. (London) A186, 165-186 (1895).
- G. Groetzinger and R. Frey, Versuche über die Änderung des Wärmedurchganges durch gase ohne elektrisches Moment, durch Flussigkeiten und feste Körper infolge eines elektricitätischen Feldes, Physik. Z. 36, 292-297 (1935).
- E. Grüneisen, Ueber die Bestimmung des metallischen Wärmeleitvermögens und über sein Verhältnis zur elektrischen Leitfähigkeit, Ann. Physik. 3, 43-74 (1900).
- E. Grüneisen, Temperaturges tz des Wärmewiderstandes regulärer Metalle, Z. Physik. 46, 151-159 (1927).

- E. Grüneisen and H. Adenstedt, Anistropie der Wärmeleitung und Thermokraft regulärer Metalle (Wolfram) im transversalen Magnetfeld bei 20° K, Ann. Physik. 29, 597-604 (1937).
- E. Grüneisen and H. Adenstedt. Einfluss transversaler Magnetfelder auf Elektrizitäts- und Wärmeleitung reiner Metalle bei tiefer Temperatur, Ann. Physik. 31, 714-744 (1938).
- E. Grüneisen and H.-D. Erfling, Elektrischer und thermischer Widerstand von Berylliumkristallen im transversalen Magnetfeld, Ann. Physik. 38, 399-420 (1940).
- E. Grüneisen and J. Gielessen, Untersuchungen an Wismutkristallen. I. Wärme- und Elektrizitätsleitung in transversalen Magnetfeldern, Ann. Physik. 26, 449-464 (1936).
- E. Grüneisen and E. Goens, Elektrizitäts- und Wärmeleitung von ein- und viel- kristallinen Metallen des regulären Systems, Z. Physik 44, 615-642 (1927).
- E. Grüneisen, K. Rausch, and K. Weiss, Zur Elektrizitäts- und Wärmeleitung von Wismut- Einkristallen im Transversalen Magnetfeld, Ann. Physik. 7, 1-17 (1950).
- E. Grüneisen and H. Reddemann, Electronen- und Gitterleitung beim Wärmefluss in Metallen, Ann. Physik 20, 843-877 (1934).
- W. J. de Haas, S. Aoyama, and H. Bremmer, Thermal conductivity of tin at low temperatures, Comm. Kamerlingh Onnes Lab., Univ. Leiden 19, #214a (1931).
- W. J. de Haas and T. Biermasz, The thermal conductivity of quartz at low temperatures, Physica 2, 673-682 (1935).
- W. J. de Haas and T. Biermasz, Sur la conductibilité thermique aux basses temperatures, Leiden Communications 22, Supplement No. 82b, 1 13 (1936-1938), reprinted from: Rapports et Communications, No. 24, 7e Congres International du Froid, la Hague-Amsterdam, juin 1936.
- W. J. de Haas and T. Biermasz, The thermal conductivity of KBr, KCl and SiO₂ at low temperatures, Physica 4, 752-756 (1937).
- W. J. de Haas and T. Biermasz, The thermal conductivity of diamond and potassiumchloride, Physica 5, 47-53 (1938a).
- W. J. de Haas and T. Biermasz, Die Wärmleitfähigkeit von Kirstallen bei tiefen Temperaturen, Physica 5, 320-324 (1938b).
- W. J. de Haas and T. Biermasz, The dependence on thickness of the thermal resistance of crystals at low temperatures, Physica 5, 619-624 (1938c).
- W. J. de Haas and H. Bremmer, Thermal conductivity of lead and tin at low temperatures, Comm. Kamerlingh Onnes Lab., Univ. Leiden, 20, #214 d (1931).
- W. J. de Haas and H. Bremmer, Thermal conductivity of indium at low temperatures, Comm. Kamerlingh Onnes Lab., Univ. of Leiden, 20, #220 b (1932).
- W. J. de Haas and H. Bremmer, The conduction of heat of lead-thallium at low temperatures, Comm. Kamerlingh Onnes Lab., Univ. Leiden, 20, #220 c (1932).
- W. J. de Haas and H. Bremmer, Determination of the heat resistance of mercury at the temperatures obtainable with liquid helium, Physica 3, 687-691 (1936).
- W. J. de Haas and W. H. Capel, A method for the determination of the thermal resistance of metal single crystals at low temperatures, Physica 1, 725-736 (1934a).

- W. J. de Haas and W. H. Capel, The thermal resistance of bismuth single-crystals at low temperatures, Physica 1, 929-934 (1934b).
- W. J. de Haas, A. N. Gerritsen, and W. H. Capel, The thermal resistance of bismuth single-crystals at low temperatures and in a magnetic field, Physica 3, 1143-1158 (1936).
- W. J. de Haas and J. de Nobel, The thermal and the electrical resistance of a tungsten single-crystal at low temperatures and in magnetic fields, Physica 5, 449-463 (1938)
- W. J. de Haas and A. Rademakers, The thermal conductivity of lead in the superconducting and normal state, Physica 7, 992-1002 (1940).
- C. V. Heer and J. G. Daunt, Heat flow in metals below 1° K and a new method for magnetic cooling, Phys. Rev. 76, 854-855 (1949).
- J. W. Hornbeck, Thermal and electrical conductivities of the alkali metals, Phys. Rev. 2, 217-240 (1913).
- G. W. Hull and T. H. Geballe, Thermal conductivity of single crystalline silicon, Bull Am. Phys. Soc. 29, 11 (1954) A.
- J. K. Hulm, Thermal conductivity of superconductors, Nature 163, 368-369 (1949) L.
- J. K. Hulm, The thermal conductivity of tin, mercury, indium, and tantalum at liquid helium temperatures, Proc. Roy. Soc. (London) A204, 98-123 (1950).
- J. K. Hulm, The thermal conductivity of a coppernickel alloy at low temperature, Proc. Phys. Soc. (London) A64, 207-211 (1951).
- J. K. Hulm, Anomalous thermal conductivity of pure metals at low temperatures, Proc. Phys. Soc. (London) A65, 227-228 (1952a) L.
- J. K. Hulm, Heat transfer in superconducting alloys, N.B.S. Circular 519, 37-42 (1952b).
- J. K. Hulm, Thermal restivity of mercury in the intermediate state, Phys. Rev. 90, 1116 (1953) L.
- R. B. Jacobs and C. Starr, Thermal conductance of metallic contacts, Rev. Sci. Inst. 10, 140-141 (1939).
- W. Jaeger and H. Diesselhorst, Wärmeleitung, Elektricitätsleitung, Wärmecapacität und Thermokraft einiger Metalle, Wiss. Abhandlung Phys. Techn. Reichanstalt 3, 269-424 (1900). (Cataloged under Charlottenburg).
- C. H. Johansson and J. O. Linde, Kristallstruktur, . . . Wärmeletifähigkeit, . . . des Systems AuPt in Verbindung, Ann. Physik 5, 762-792 (1930).
- W. G. Kannaluik, On the thermal conductivity of some metal wires, Proc. Roy. Soc. (London) A131, 320-335 (1951).
- W. G. Kannaluik, The thermal and electrical conductivities of several metals between —183° C and 100° C, Proc. Roy. Soc. (London) A141, 159-168 (1933).
- W. G. Kannaluik and T. H. Laby, The thermal and the electrical conductivity of a copper crystal at various temperatures, Proc. Roy. Soc. (London) A121, 640-653 (1928).
- J. Karweil and K. Schäfer, Die Wärmeleitfähigkeit einiger schlecht leitender Legierungen zwischen 3 und 30° K, Ann. Physik 36, 567-577 (1939).
- G. W. C. Kaye and W. F. Higgins, The thermal conductivity of a single crystal of bismuth in a transverse magnetic field, Phil. Mag. 8, 1056-1059 (1929).
- G. W. C. Kaye and J. K. Roberts, The thermal conductivities of metal crystals, I. Bismuth, Proc. Roy. Soc. (London) A104, 98-114 (1923).
- P. H. Keesom, Heat conductivity of glass at 1.3° K, Physica 11, 339-342 (1945).

- W. R. G. Kemp, A. K. Sreedhar, and G. K. White, The thermal conductivity of magnesium at low temperatures, Proc. Phys. Soc. (London) A66, 1077-1078 (1953).
- R. Kikuchi, Wärmeleitvermögen und elektrisches Leitvermögen einer Anzahl von Magnesium-Legierungen und ihr Verhalten zum Wiedemann-Franzschen Gesetz, Sci. Rep. Tôhoku Imperial Univ. (1st Series) 21, 585-593 (1932) (Cataloged under Sendai).
- C. Kittel, Interpretation of the thermal conductivity of glasses, Phys. Rev. 75, 972-974 (1949).
- M. Kohler, Wärmeleitung der Metalle im starken Magnetfeld, Ann. Physik 5, 181-189 (1949).
- S. Konno, On the variation of thermal conductivity during the fusion of metals, Sci. Rep. Tôhoku Imperial Univ. (1st Series) 8, 169-179 (1919) (Cataloged under Sendai).
- S. Konno, On the variation of thermal conductivity during the fusion of metals, Phil. Mag. 40, 542-552 (1920).
- N. Kurti, B. V. Rollin, and F. Simon, Preliminary experiments on temperature equilibria at very low temperatures, Physica 3, 266-274 (1936).
- Landolt-Börnstein Physikalisch-chemische Tabellen, Edited by W. A. Roth and K. Scheel (Julius Springer, Berlin) 5th ed., vol. 2, 1923; 5th ed., 1st supplement, vol. 1, 1927; 5th ed., 2d supplement, vol. 2, 1981; 5th ed., 3d supplement, vol. 3, 1936.
- I. Langmuir and J. B. Taylor, The heat conductivity of tungsten and the cooling effects of leads upon filaments at low temperatures, Phys. Rev. 50, 68-87 (1936).
- C. H. Lees, Effects of temperature and pressure on the thermal conductivities of solids—Part 1. The effect of temperature on the thermal conductivities of some electrical insulators, Phil. Trans. Roy. Soc. (London) A204, 433-466 (1905).
- C. H. Lees, The effects of temperature and pressure on the thermal conductivities of solids—Part 2. The effects of low temperatures on the thermal and electrical conductivities of certain approximately pure metals and alloys, Phil. Trans. Roy. Soc. (London) A208, 381-443 (1908).
- C. H. Lees and J. E. Calthrop, The effect of torsion on the thermal and electrical conductivities of metals, Proc. Phys. Soc. (London) 35, 225-234 (1923).
- E. J. Lewis, Some thermal and electrical properties of beryllium, Phys. Rev. 34, 1575-1587 (1929).
- J. O. Linde, An investigation of the validity of the Wiedemann-Franz-Lorenz law, Arkiv Fysik 4, 541-554 (1952).
- L. Lorenz, Ueber das Leitungsvermögen der Metalle für Wärme und Electricität, Ann. Physik 13, 422-447 (1881a).
- L. Lorenz, Ueber das Leitungsvermögen der metalle für Wärme und Electricität, Ann. Physik 13, 582-606 (1881b).
- P. Macchia, Thermal conductivity at low temperatures, Accad. Lincei. Atti. 16, 507-517 (1907).
- W. Mannchen, Wärmeleitvermögen, elektrisches Leitvermögen und Lorenzsche Zahl einiger Leichtmetall-Legierungen, Z. Metalkunde 23, 193-196 (1931).
- Massachusetts Institute of Technology, Quarterly Progress Reports (October 15, 1952, April 15, 1953, July 15, 1953).
- W. Meissner, Über die thermische und electrische Leitfähigkeit von Kupfer zwischen 20 und 373° abs., Verhand. Deutsch. Physik. Gesell. 16, 262-272 (1914).
- W. Meissner, Thermische und elektrische Leitfähigkeit von Platin zwischen 20 und 373° abs., Ann. Physik 47, 1001-1058 (1915).

- W. Meissner, Thermische und elektrische Leitfähigkeit von Lithium zwischen 20 und 373° abs., Z. Physik 2, 373-379 (1920).
- H. Masumoto, On the thermal and electrical conductivities of some aluminum alloys, Sci. Rep. Tôhoku Imperial Univ. (1st Series) 13, 229-242 (1925) (Cataloged under Sendai).
- H. Masumoto, On the electrical and thermal conductivities of carbon steel and cast iron, Sci. Rep. Tôhoku Imperial Univ. (1st Series) 16, 417-435 (1927) (Cataloged under Sendai).
- K. Mendelssohn, Heat conductivity of metals at low temperatures, Bull. Inst. Int. du Froid, Annexe 1952-1, 69-79 (1952).
- K. Mendelssohn, Thermal conductivity of superconductors, Physica 19, 775-787 (1953).
- K. Mendelssohn and J. L. Olsen, Heat transport in superconductors, Proc. Phys. Soc. (London) A63, 2-13 (1950a).
- K. Mendelssohn and J. L. Olsen, Heat flow in superconductive alloys, Proc. Phys. Soc. (London) A63, 1182-1183 (1950b).
- K. Mendelssohn and J. L. Olsen, Anomalous heat flow in superconductors, Phys. Rev. 80, 859-862 (1950c).
- K. Mendelssohn and R. B. Pontius, Thermal conductivity of supraconductors in a magnetic field, Phil. Mag. 24, 777-787 (1937).
- K. Mendelssohn and C. A. Renton, Heat conductivities of superconductive Sn, In, Tl, Ta, Cb, and Al below 1° Kelvin, Phil. Mag (7) 44, 776-781 (1953).
- K. Mendelssohn and H. M. Rosenberg, The thermal conductivity of metals at low temperatures I: The elements of Groups 1, 2 and 3, Proc. Phys. Soc. (London) A65, 385-388 (1952a).
- K. Mendelssohn and H. M. Rosenberg, The thermal conductivity of metals at low temperatures II: The transition elements, Proc. Phys. Soc. (London) A65, 388-394 (1952b).
- K. Mendelssohn and H. M. Rosenberg. The thermal conductivity of metals in high magnetic fields at low temperatures, Proc. Roy. Soc. (London) A218, 190-205 (1953).
- W. C. Michels and M. Cox, The thermal conductivity of tungsten, Physics 7, 1953-155 (1936).
- S. Mizushima and J. Okada, Notes on the electrical and thermal conductivities of graphite and amorphous carbon, Phys. Rev. 82, 94-95 (1951).
- S. Mrozowski, Thermal conductivity of carbons and graphite, Phys. Rev. 86, 251-252 (1952) L.
- J. Nicol and T. P. Tseng, Thermal conductivity of copper between 0.25° K and 4.2° K, Phys. Rev. 92, 1062-1063 (1953) L.
- J. de Nobel, Thermal and electrical resistance of a tungsten single crystal at low temperatures and in high magnetic fields, Physica 15, 532-540 (1949).
- J. de Nobel, Heat conductivity of steels and a few other metals at low temperatures, 17, 551-562 (1951).
- J. L. Olsen, Heat transport in lead-bismuth alloys, Proc. Phys. Soc. A65, 518-532 (1952).
- J. L. Olsen and C. A. Renton, Heat conductivity of superconductive lead below 1° K, Phil Mag. 43, 946-948 (1952).
- J. L. Olsen and H. M. Rosenberg, The thermal conductivity of metals at low temperatures, Adv. Physics 2, 28-65 (1953).
- H. K. Onnes and G. Holst, Preliminary determination of the specific heat and of the thermal conductivity of mercury, Comm. Kamerlingh Onnes Lab., Univ. of Leiden, No. 142c, 24-33 (1914).

- J. R. Partington, An advanced treatise on physical chemistry (Longmans, Green & Co., Ltd., London, 1952), first edition, vol. 3, pp. 458-561.
- T. Peczalski, Contribution à l'étude de la conductibilité calorifique des solides, Ann. physique 7, 185-224 (1917).
- R. L. Powell, The thermal conductivity of "Easy-Flo" silver solder from 20° to 200° K, N.B.S. Report 2609 (1953, 3 pp. (unpublished).
- R. W. Powell, The electrical resistivity of gallium and some other anisotropic properties of this metal, Proc. Roy. Soc. (London) A209, 525-541 (1951).
- R. W. Powers, D. Schwartz, and H. L. Johnston, The thermal conductivity of metals and alloys at low temperatures I. Apparatus for measurements between 25° and 300° K. Data on pure aluminum, OFHC copper and "L" nickel, TR 264-5, Cyrogenics Laboratory, Ohio State University (1951) 22 pp.
- R. W. Powers, J. B. Ziegler, and H. L. Johnston, The thermal conductivity of metals and alloys at low temperatures. II. Data on iron and several steels between 25° and 300° K Influence of alloying constituents, TR 264-6, Cyrogenics Laboratory, Ohio State University (1951a), 17 pp.
- R. W. Powers, J. B. Ziegler, and H. L. Johnston, The thermal conductivity of metals and alloys at low temperatures. III. Data for aluminum alloys between 25° and 300° K, TR 264-7, Cyrogenics Laboratory, Ohio State University (1951b), 10 pp.
- R. W. Powers, J. B. Ziegler, and H. L Johnston, The thermal conductivity of metals and alloys at low temperatures. IV Data on constantan, monel and contracid between 25° and 300° K, TR 264-8, Cyrogenics Laboratory, Ohio State University (1951c), 11 pp.
- R. W. Quick, C. D. Child, and B. S. Lamphear, Thermal conductivity of copper, Phys. Rev. 3, 1-20 (1895)
- A. Rademakers, The thermal conductivity of lead and tin in the superconducting and in the normal state, Physica 15, 849-859 (1949).
- K. Rausch, Untersuchungen an Antimon-Einkristallen im transversalen Magnetfeld, Ann. Physik 1, 190-206 (1947).
- H. Reddemann, Wärmleitvermögen, Wiedemann-Franz-Lorenzsche Zahl und Thermokraft von Quecksilbereinkristallen, Ann. Physik. 14, 139-163 (1932).
- H. Reddemann, Änderung der thermischen und elektrischen, Leitfähigkeit eines Bi-Einkristallen im Magnetfeld, Ann. Physik 20, 441-448 (1934).
- H. Reddemann, Wiedemann-Franzche Zahl von β -Manganan bei —190° C, Ann. Physik 22, 28-30 (1935).
- A. Rietzsch, Über die thermische und elektrische Lietfähigkeit von Kupfer-Phosphor und Kupfer-Arsen, Ann. Physik 3, 403-427 (1900).
- C. J. Rigney and L. I. Bockstahler, The thermal conductivity of titanium between 20 and 273° K, Phys. Rev. 83, 220 (1951) A.
- M. T. Rodine, Thermal conductivities of bismuth single crystals as influenced by a magnetic field, Phys. Rev. 46, 910-916 (1934).
- H. M. Rosenberg, Private Communication (January 1954a).
- H. M. Rosenberg, The thermal and electrical conductivity of magnesium at low temperatures, Phil. Mag. 45, 73-79 (1954b).
- A. Schallmach, Heat conductivity of rubber at low temperatures, Nature 145, 67 (1940) L.
- A. Schallmach, Heat conductivity of rubber at low temperatures, Proc. Phys. Soc. (London) 53, 214-218 (1941).

- W. Schaufelberger, Wärmeleitungsfähigkeit des Kupfers, aus dem stationaren und varibeln Temperaturzustand bestimmt, und Wärmeleiters, Ann. Physik 7, 589-630 (1902).
- F. H. Schofield, The thermal and electrical conductivities of some pure metals, Proc. Roy. Soc. (London) A107 (1925).
- R. Schott, Über das Wärmleitvermögen einigen Metalle bei tiefen Temperaturen, Verhand. Deutsch. Physik. Gesell. 18, 27-34 (1916).
- F. A. Schulze, Die Wärmeleitfähigkeit einiger Reihen von Edelmetallegierungen, Verhand. Deutsch. Physik. Gesell. 13, 856-856 (1911).
- S. Shalyt, The thermal conductivity of bismuth at low temperatures, J. Phys. (U.S.S.R.) 8, 315-316 (1944) L.
- E. G. Sharkoff, Thermal conductivity of magnesium, Quarterly Progress Report (of Research Lab. of Electronics, M.I.T.) (October 15, 1952).
- E. G. Sharkoff, Thermal conductivity of magnesium, Quarterly Progress Report (of Research Lab. of Electronics, M.I.T.) (April 15, 1953a).
- E. G. Sharkoff, Electrical and thermal conductivity of magnesium, Quarterly Progress Report (of Research Lab. of Electronics, M.I.T.) (July 15, 1953b).
- C. V. Simson, Über die Wärmeleitfähigkeit des Ammoniumchlorid im Bereiche seiner II-III-Umwandlung, Naturwiss. 38, 559 (1951).
- A. W. Smith, Thermal conductivity of graphite, North American Aviation, Inc., Downey, California (1954) (Unpublished).
- C. S. Smith and E. W. Palmer, Thermal and electrical conductivities of copper alloys, Am. Inst. Mining Metal. Eng. Tech. Publ. No. 648 (1935).
- E. H. Sondheimer, The thermal conductivity of metals at low temperatures, Proc. Phys. Soc. (London) A65, 562-564 (1952) L.
- J. Staebler, Electriches und thermisches Leitvermögen und Wiedemann-Franzche Zahl von Leichtmetallen und Magnesium-legierungen. Dissertation, April, 1929, Technische Hochschule of Breslau. Published by Doktardruck-Graphiches Institut Paul Funk, Berlin.

- R. W. B. Stephens, The temperature variation of the thermal conductivity of Pyrex glass, Phil. Mag. 14, 897-914 (1932).
- W. W. Tyler and A. C. Wilson, Jr., Thermal conductivity, electrical resistivity, and thermoelectric power of titanium alloy RC-130-B, Knolls Atomic Power Laboratory Report 803 (1952) 41 pp.
- W. W. Tyler and A. C. Wilson, Jr., Thermal conductivity, electrical resistivity, and thermoelectric power of graphite, Phys. Rev. 89, 870-875 (1953).
- W. W. Tyler, A. C. Wilson, Jr., and G. J. Wolga, Thermal conductivity, electrical resistivity, and thermoelectric power of uranium, Knolls Atomic Power Laboratory Report 802 (1952) 25 pp.
- F. J. Webb, K. R. Wilkinson, and J. Wilks, The thermal conductivity of solid helium, Proc. Roy. Soc. (London) A214, 546-563 (1952).
- F. J. Webb and J. Wilks, The thermal conductivity of solid helium at high densities, Phil. Mag. 44, 663-674 (1953).
- R. T. Webber and D. A. Spohr, Thermal resistivity of superconducting mercury in the intermediate state, Phys. Rev. 91, 414-415 (1953) L.
- S. Weber, . . . des Verhältnisses von Wärmeleitung zur Elektrizitätsleitung . . . des Wolframs, Ann. Physik 54, 165-181 (1917).
- G. K. White, The thermal conductivity of gold at low temperatures, Proc. Phys. Soc. (London) A66, 559-564 (1953a).
- G. K. White, The thermal conductivity of silver at low temperatures, Proc. Phys. Soc. (London) A66, 844-845 (1953b) L.
- G. K. White, The thermal and electrical conductivity of copper at low temperatures, Aust. J. Physics 6, 397-404 (1953c).
- K. R. Wilkinson and J. Wilks, Some measurements of heat flow along technical materials in the region 4° to 20° K, Phys. in Ind. 26, 19-20 (1949).
- J. Wilks, The thermal conductivity of ideal dielectric crystals, Bull. Inst. Int. du Froid, annexe 1952-1 (1952).

BOULDER, COLORADO, July 23, 1954